

# Coherent Hard X-ray Scattering and Imaging

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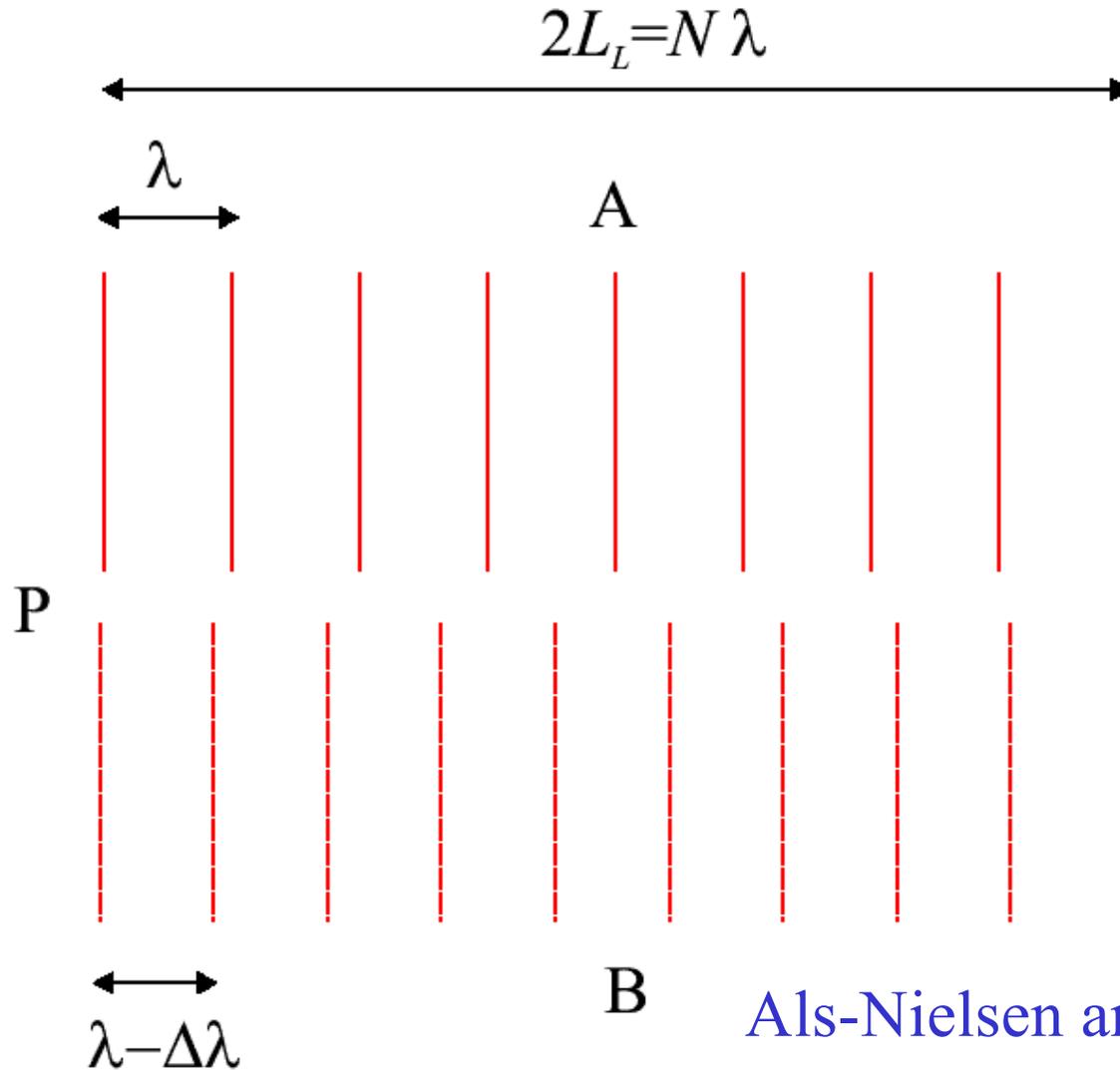
Workshop on emerging new  
directions of synchrotron  
research with soft X-  
rays VUV and Infrared

SRC, Madison, September 2004

# Outline

- Coherence in Diffraction
- How to Solve the Phase Problem
- Nanocrystal Shapes
- Use of Optics in CXD
- Diffraction from **Phase** Objects

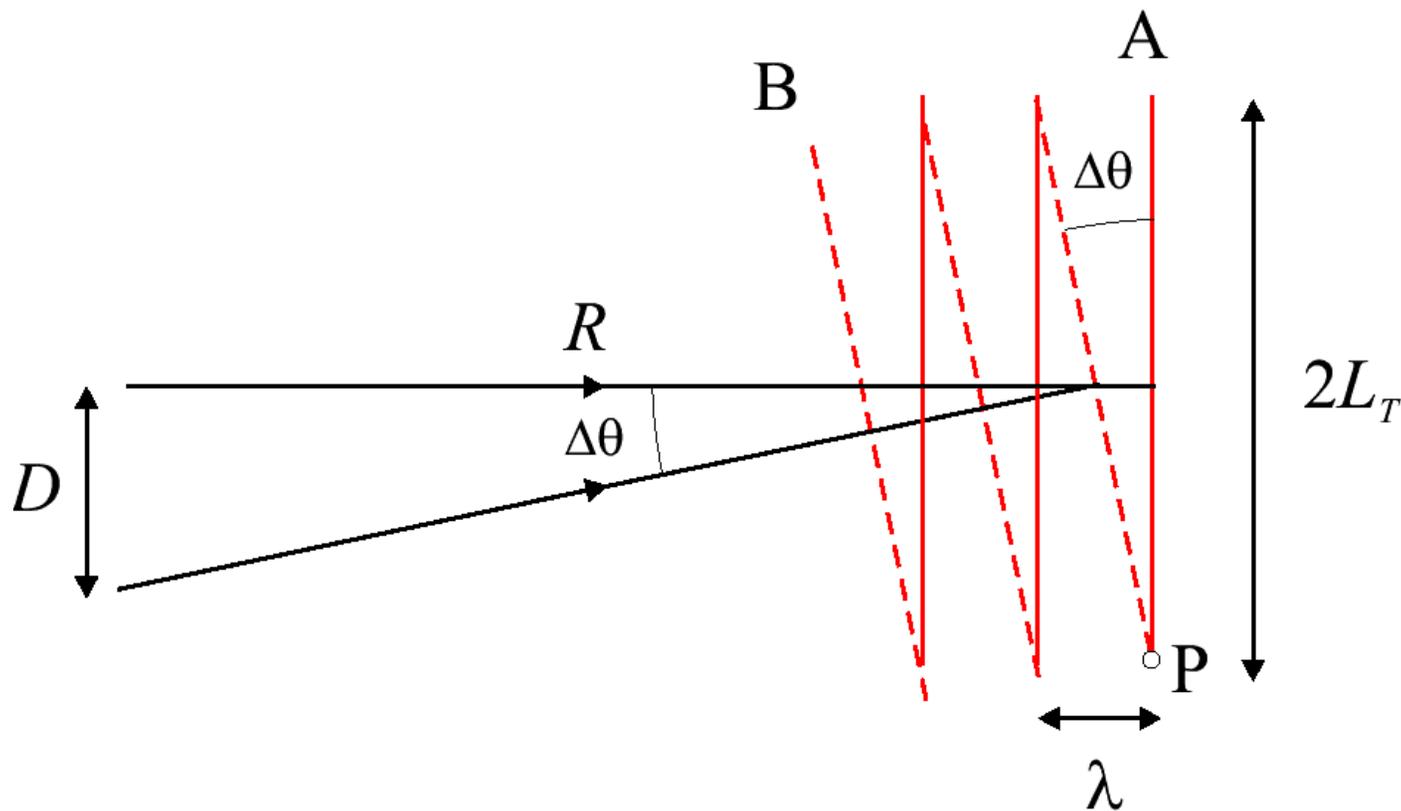
# Longitudinal Coherence



$$L_L = \frac{1}{2} \frac{\lambda^2}{\Delta\lambda}$$

Als-Nielsen and McMorro (2001)

# Lateral (Transverse) Coherence



$$L_T = \frac{\lambda R}{2 D}$$

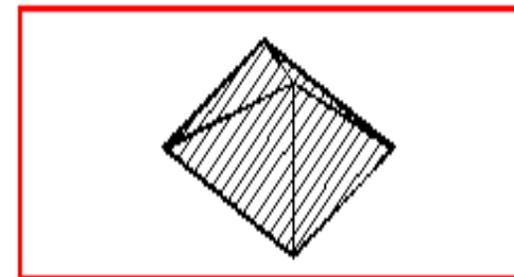
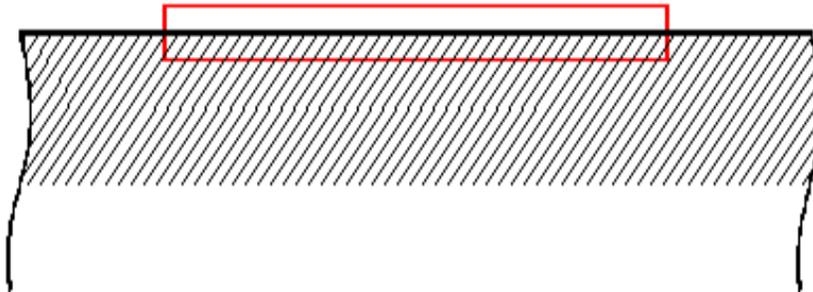
Als-Nielsen and McMorro (2001)

# Coherence at the APS or ESRF

Typical of 3rd Generation (undulator) Synchrotron Source

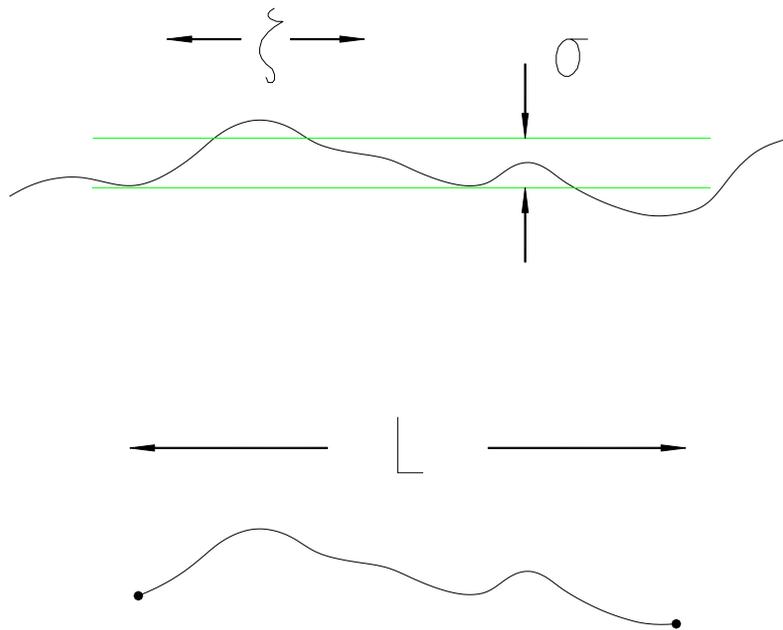
Coherence of	$\xi_{\text{VER}}$	$\xi_{\text{HORIZ}}$	$\xi_{\text{LONG}}$	Flux
Raw Undulator	35 $\mu\text{m}$	9 $\mu\text{m}$	0.004 $\mu\text{m}$	$2 \times 10^{12}$
Si(111) Monochromator	35 $\mu\text{m}$	9 $\mu\text{m}$	1 $\mu\text{m}$	$1 \times 10^{10}$
C(111) Monochromator	35 $\mu\text{m}$	9 $\mu\text{m}$	3 $\mu\text{m}$	$3 \times 10^9$

Coherent region defined by slits

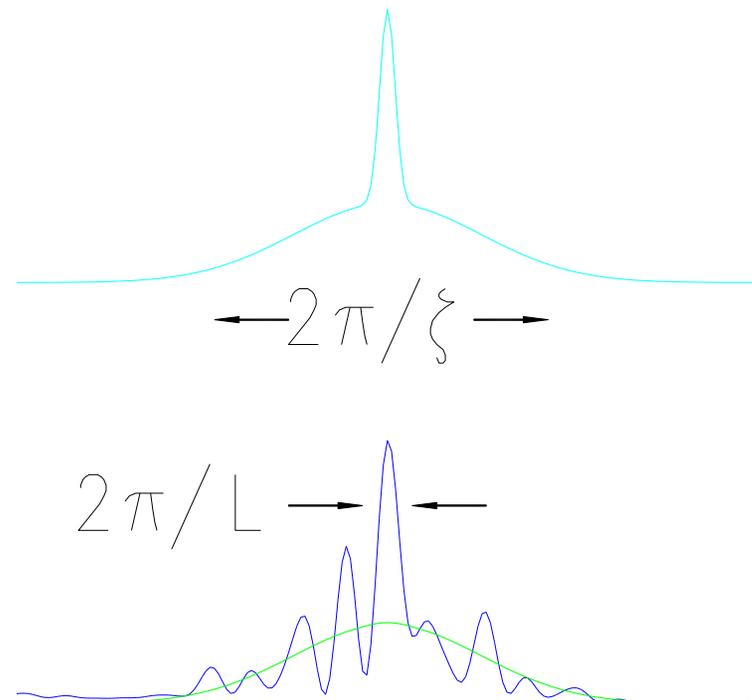


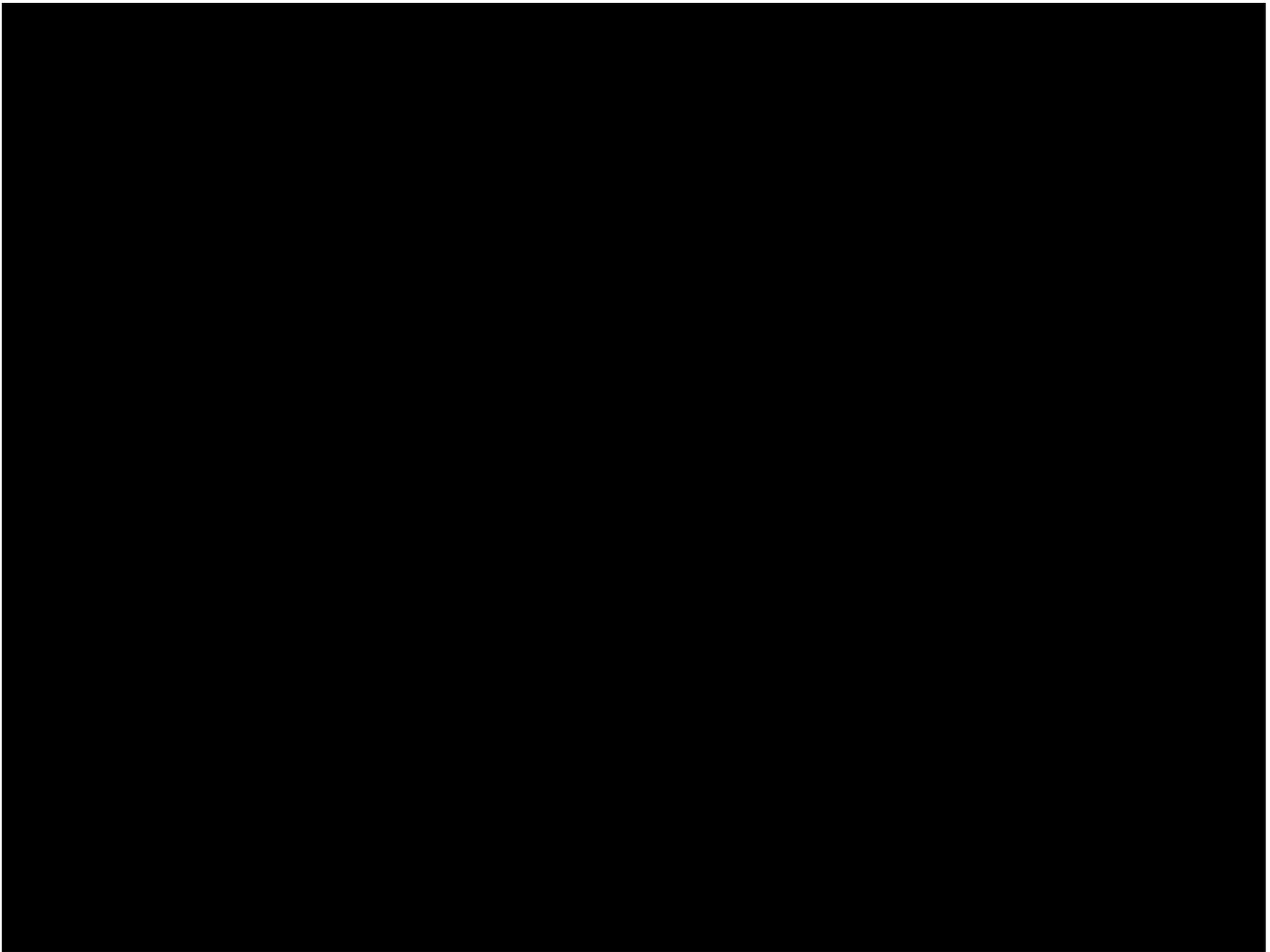
# Diffuse Scattering acquires Structure using CXD

Real Space



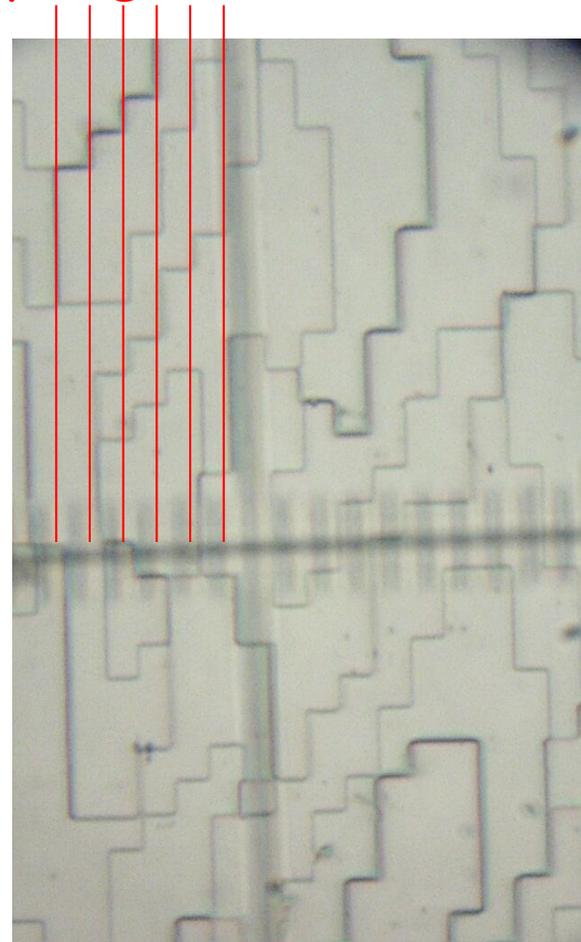
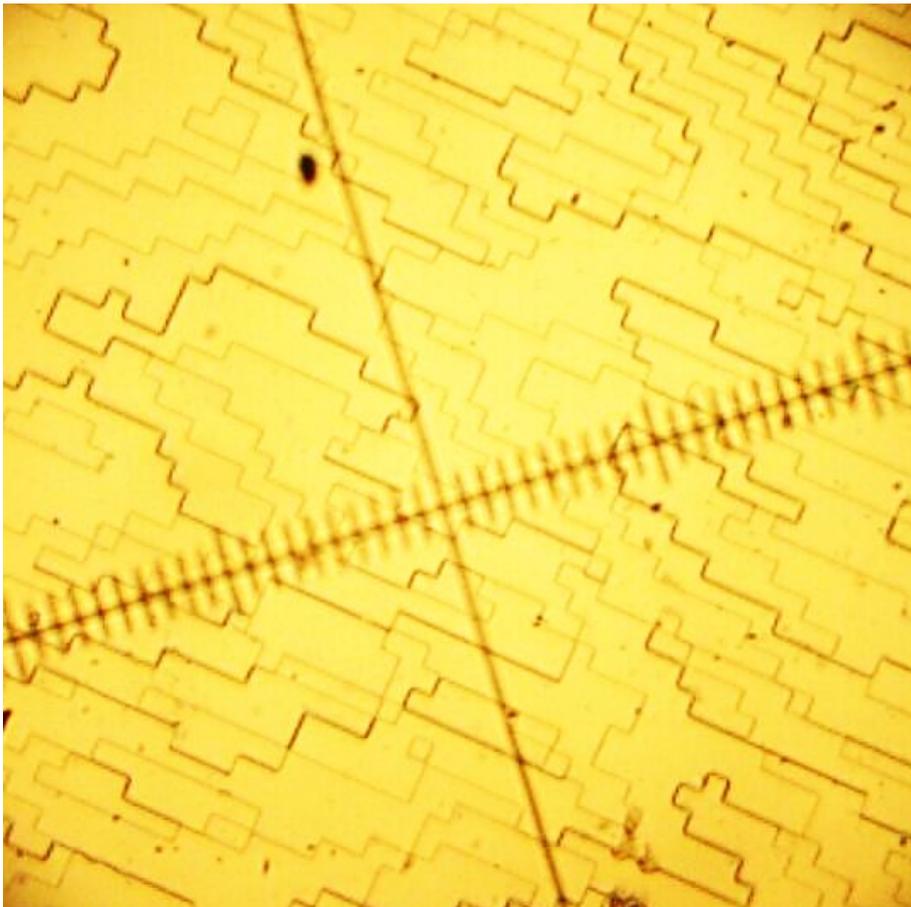
Reciprocal Space

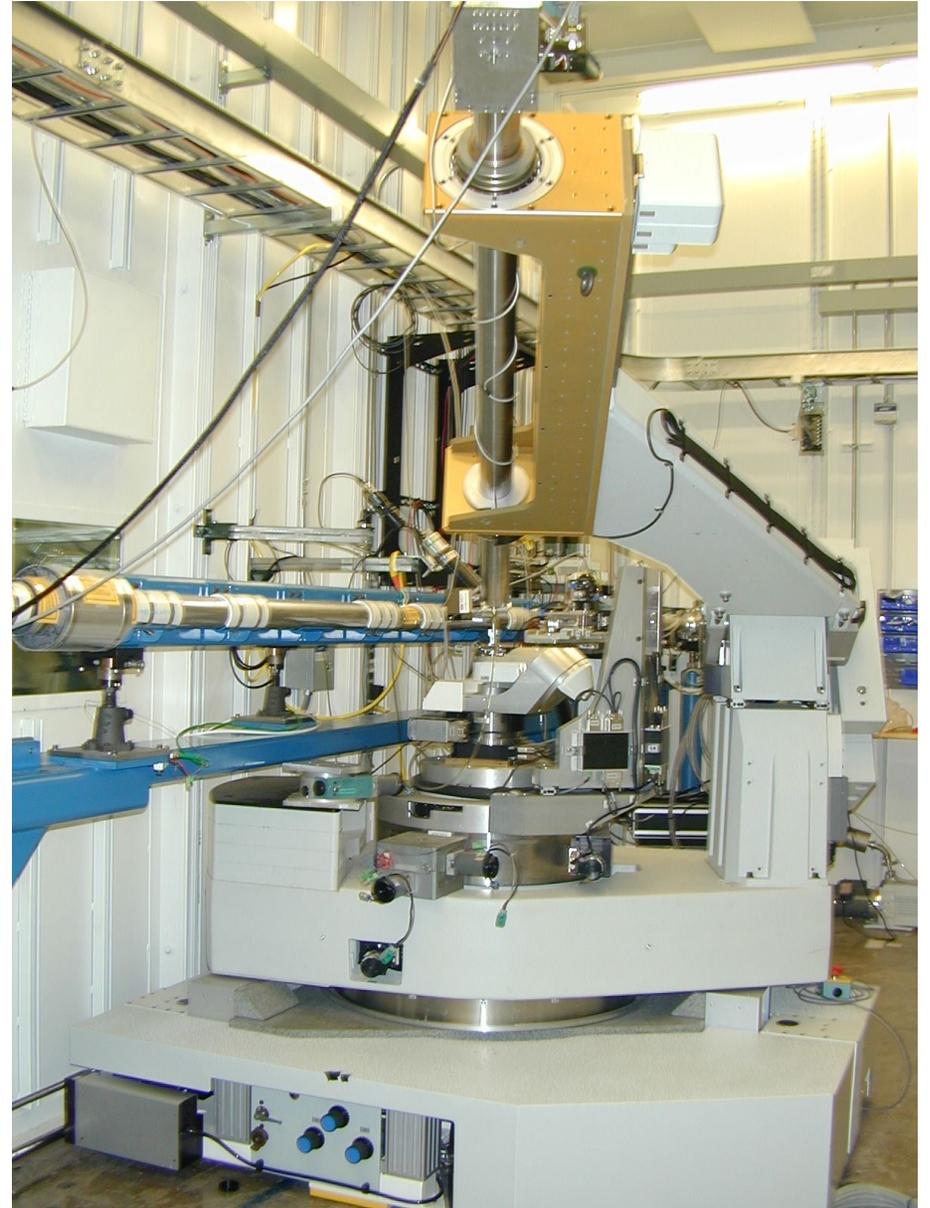
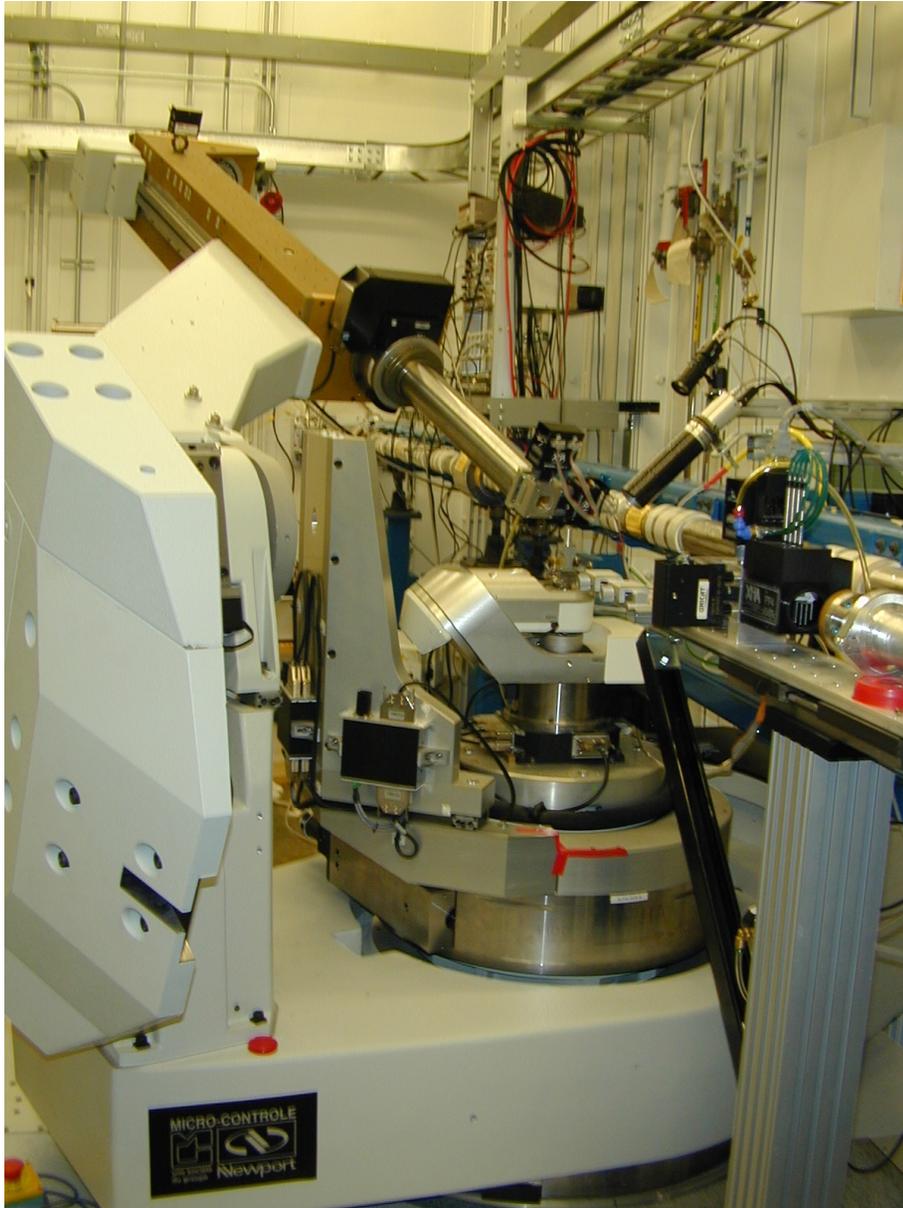


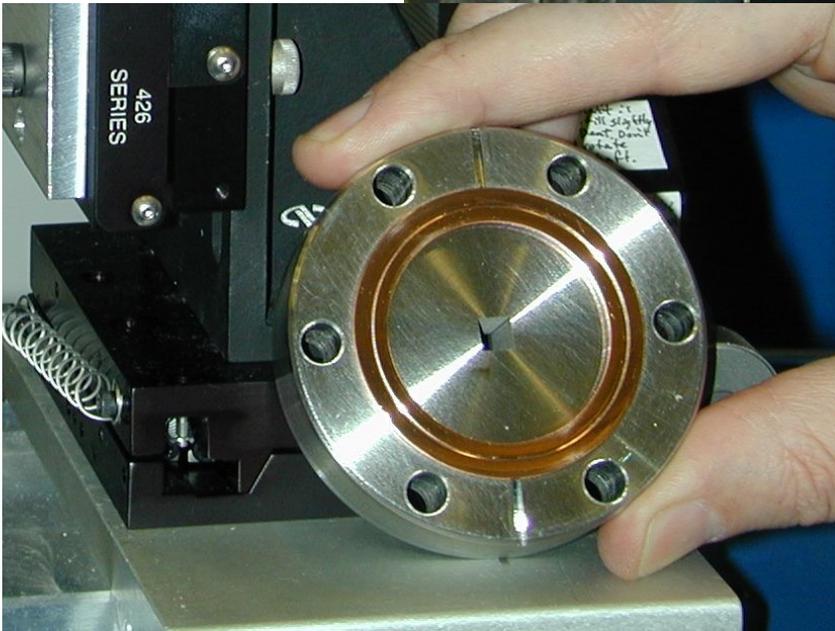
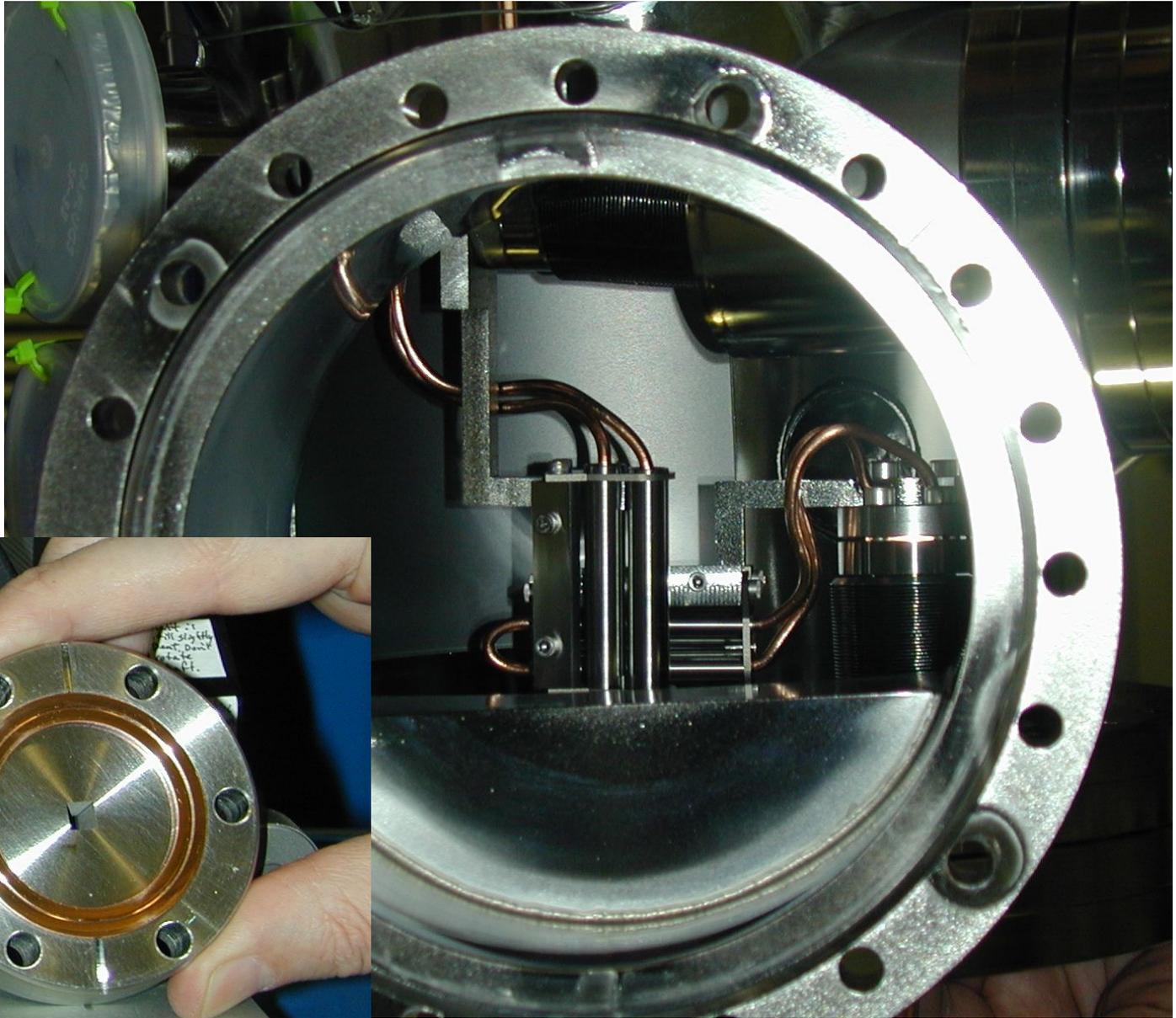


# Microscope Images of Gratings

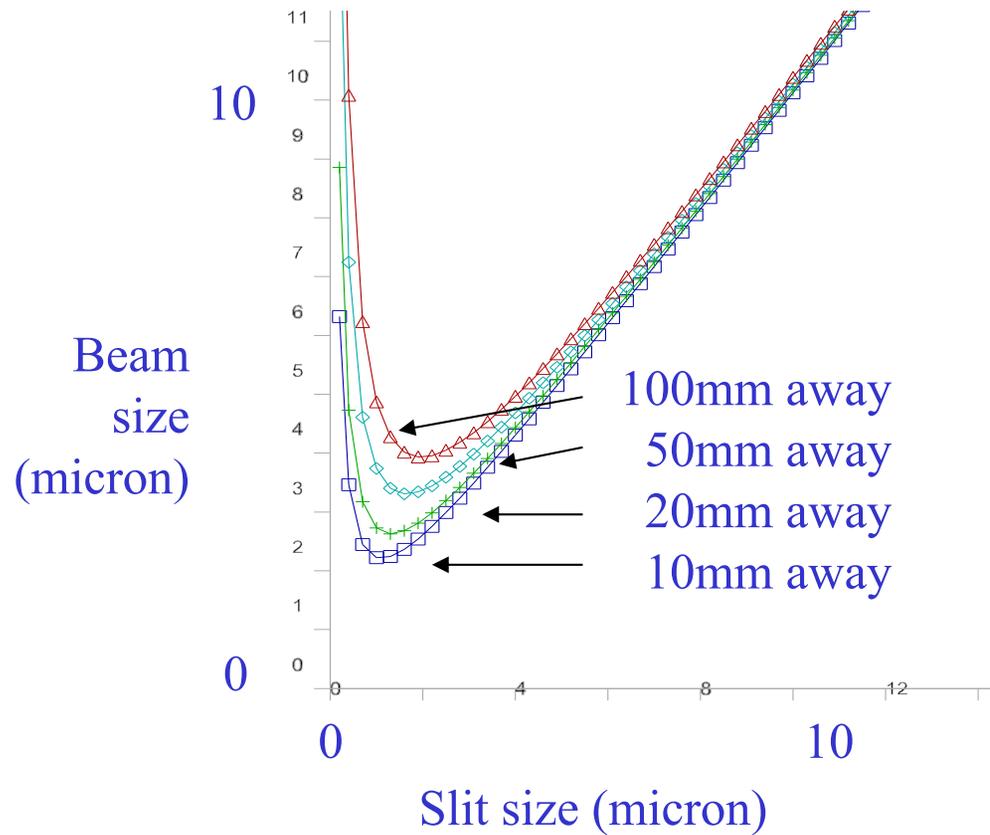
5 $\mu$ m grid lines







# Smallest Beam using Slits (9keV)

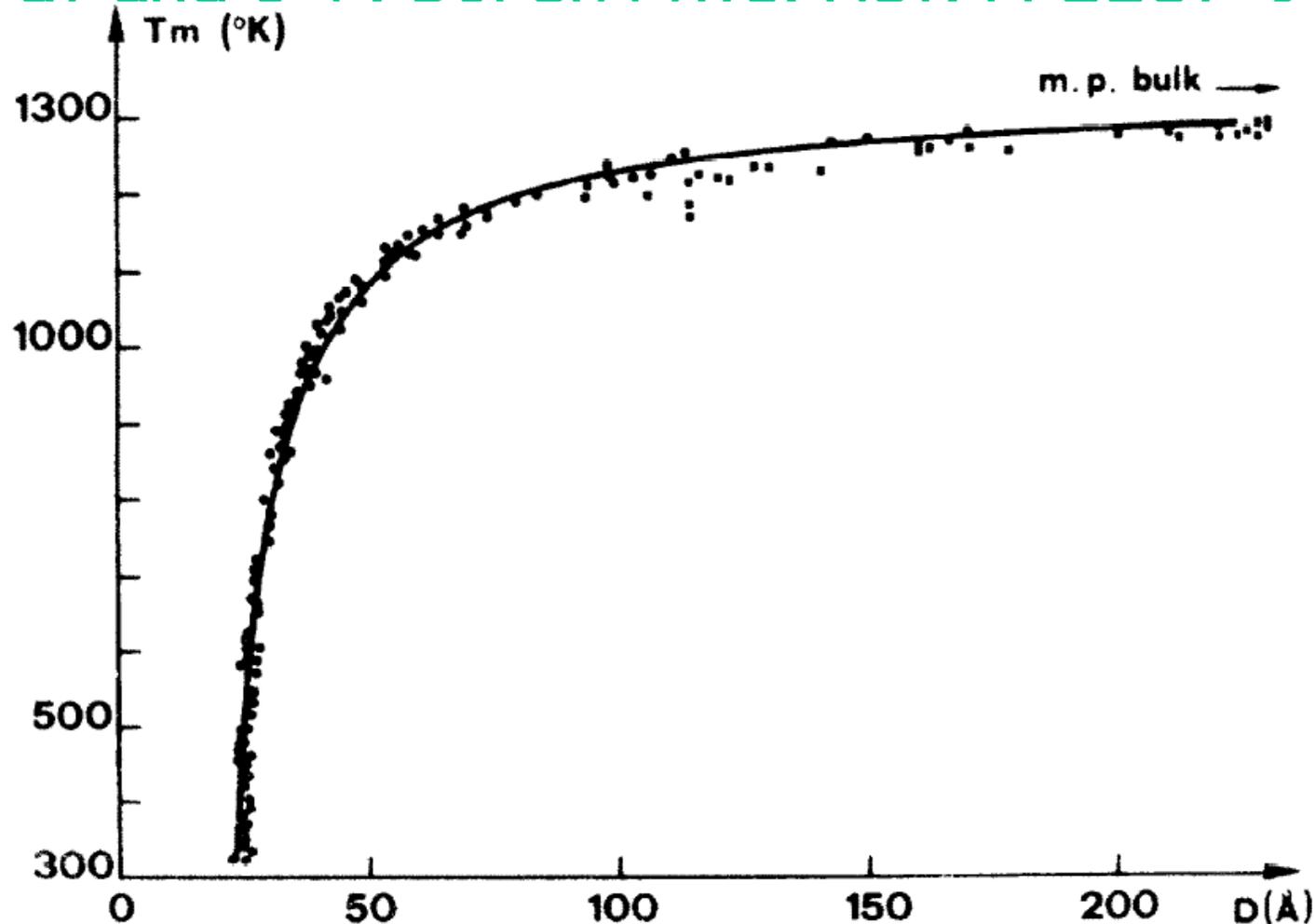


$$y(x) = x + \frac{\lambda d}{x}$$

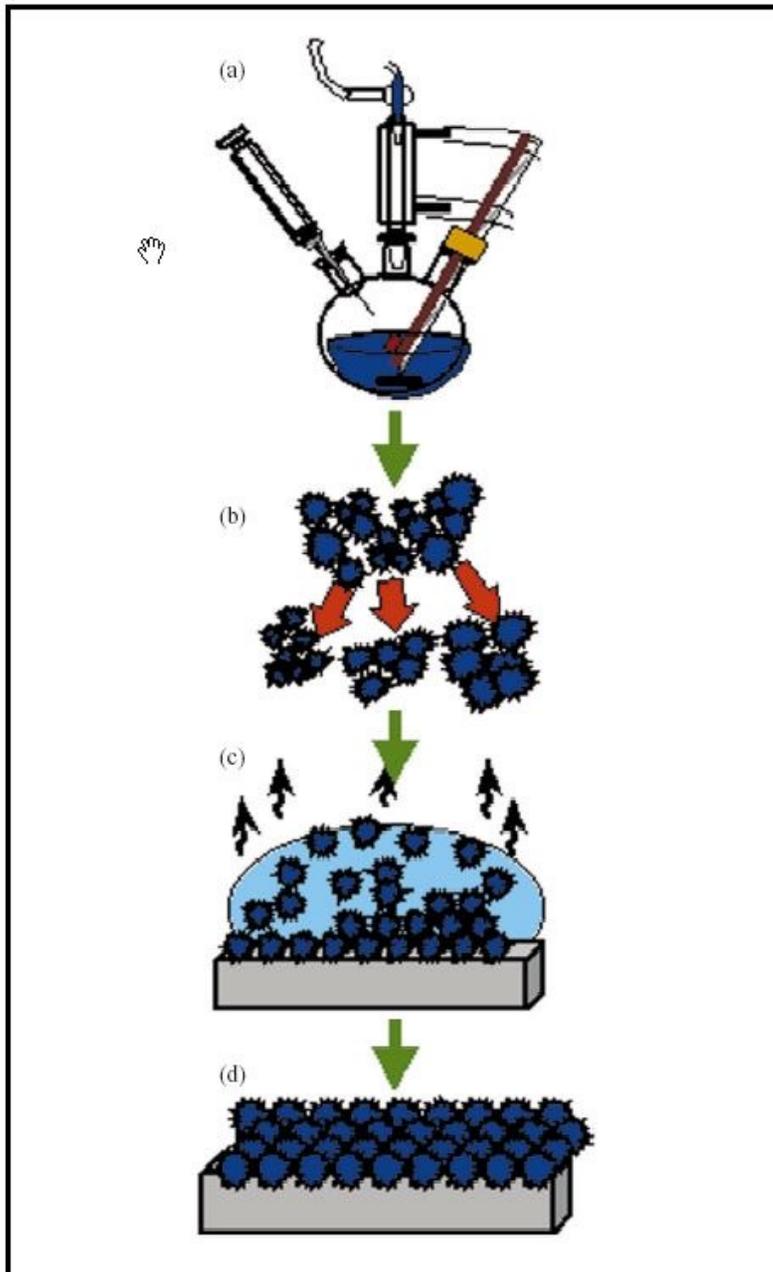


# Size-dependent Melting of Au Particles

P. Buffat and J-P. Borel. Phys. Rev. A 2287-97 (1975)

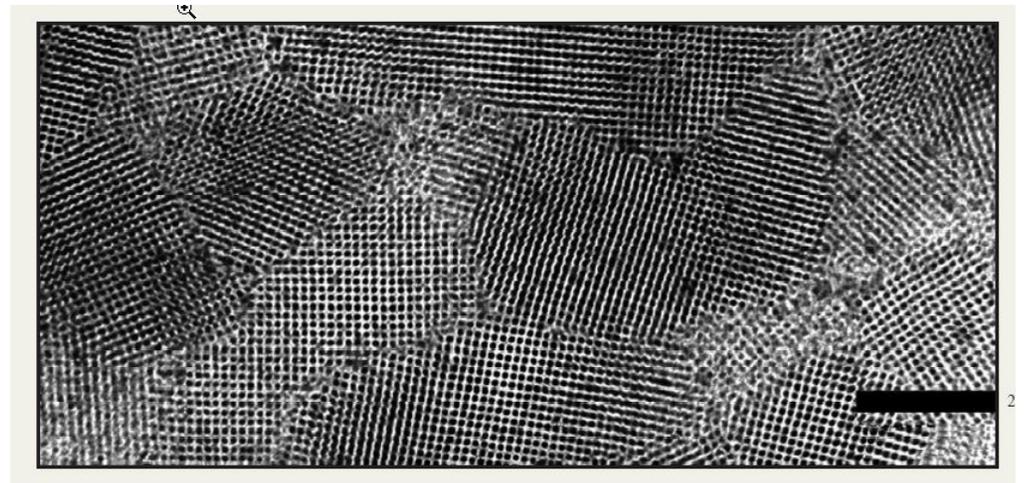


# Chemical Synthesis of Nanocrystals



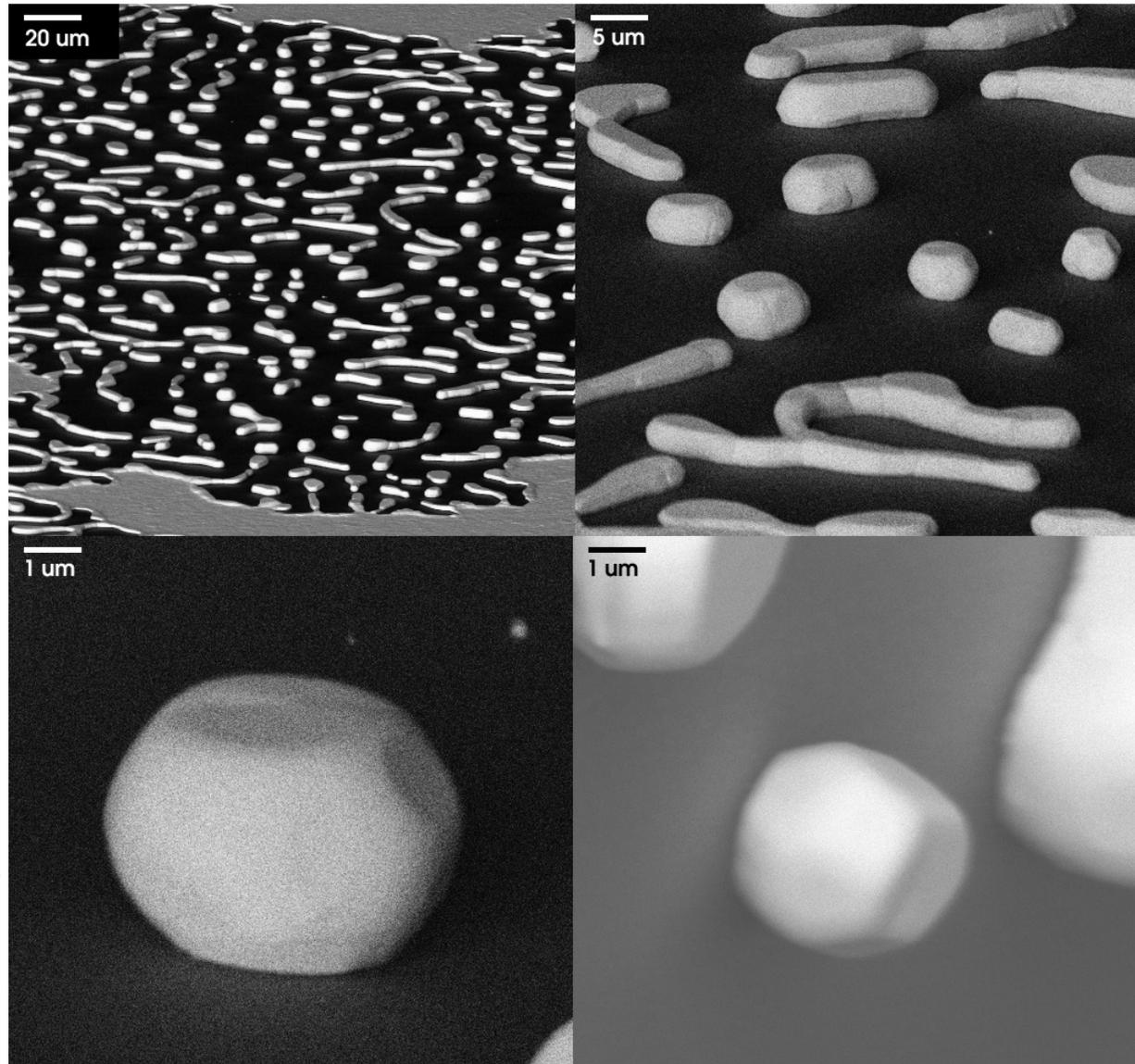
- Reactants introduced rapidly
- High temperature solvent
- Surfactant/organic capping agent
- Square superlattice (200nm scale)

C. B. Murray, IBM J. Res. & Dev. 45  
47 (2001)

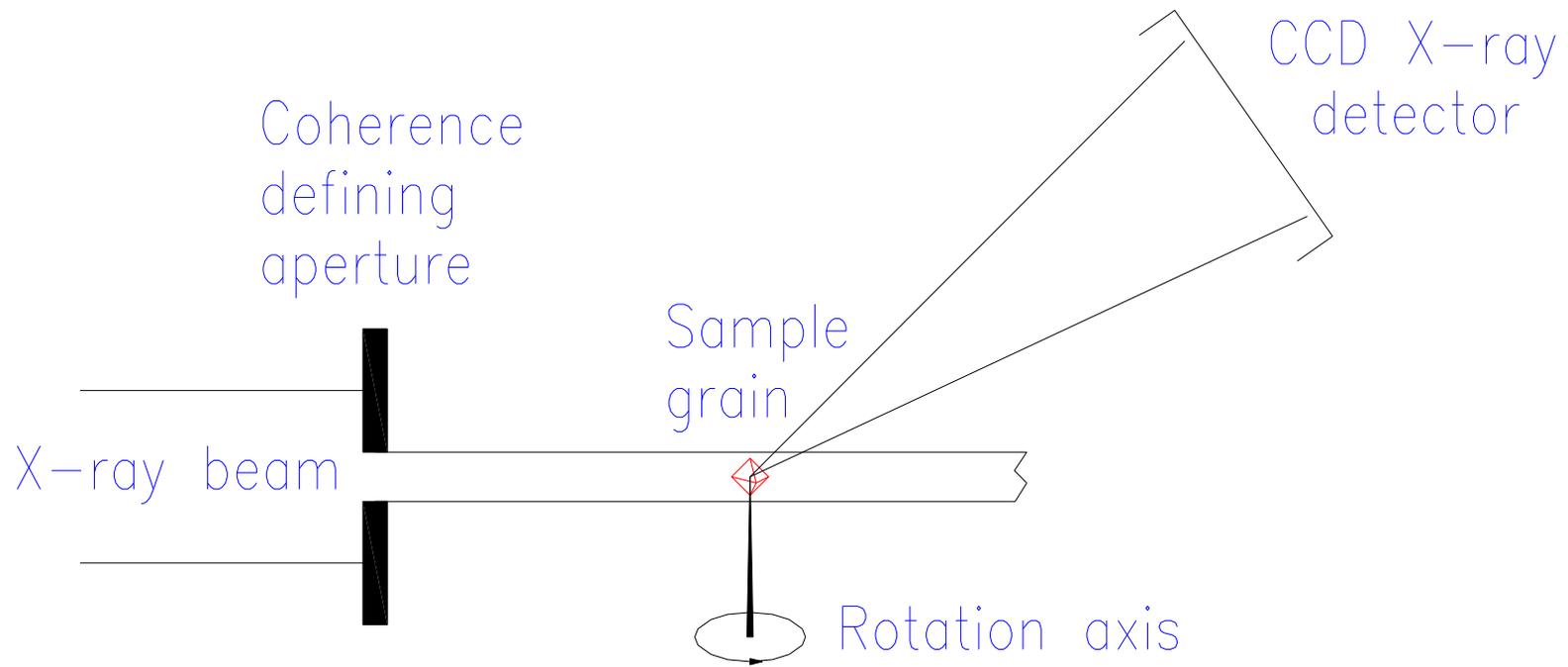


# SEMS

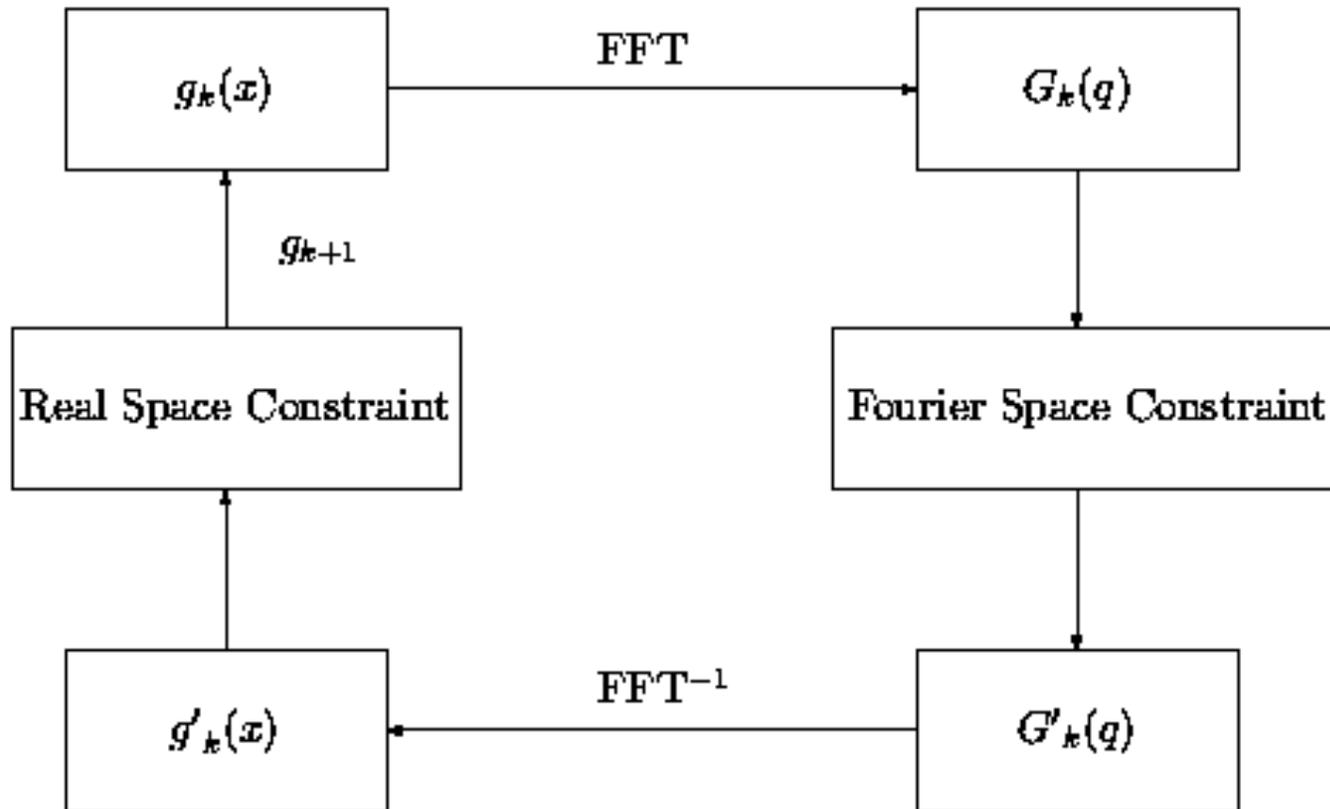
- Au blanket film
- Quartz substrate
- Annealed at 950°C for 70 hrs.



# Lensless X-ray Microscope



# Generic "Error Reduction" method



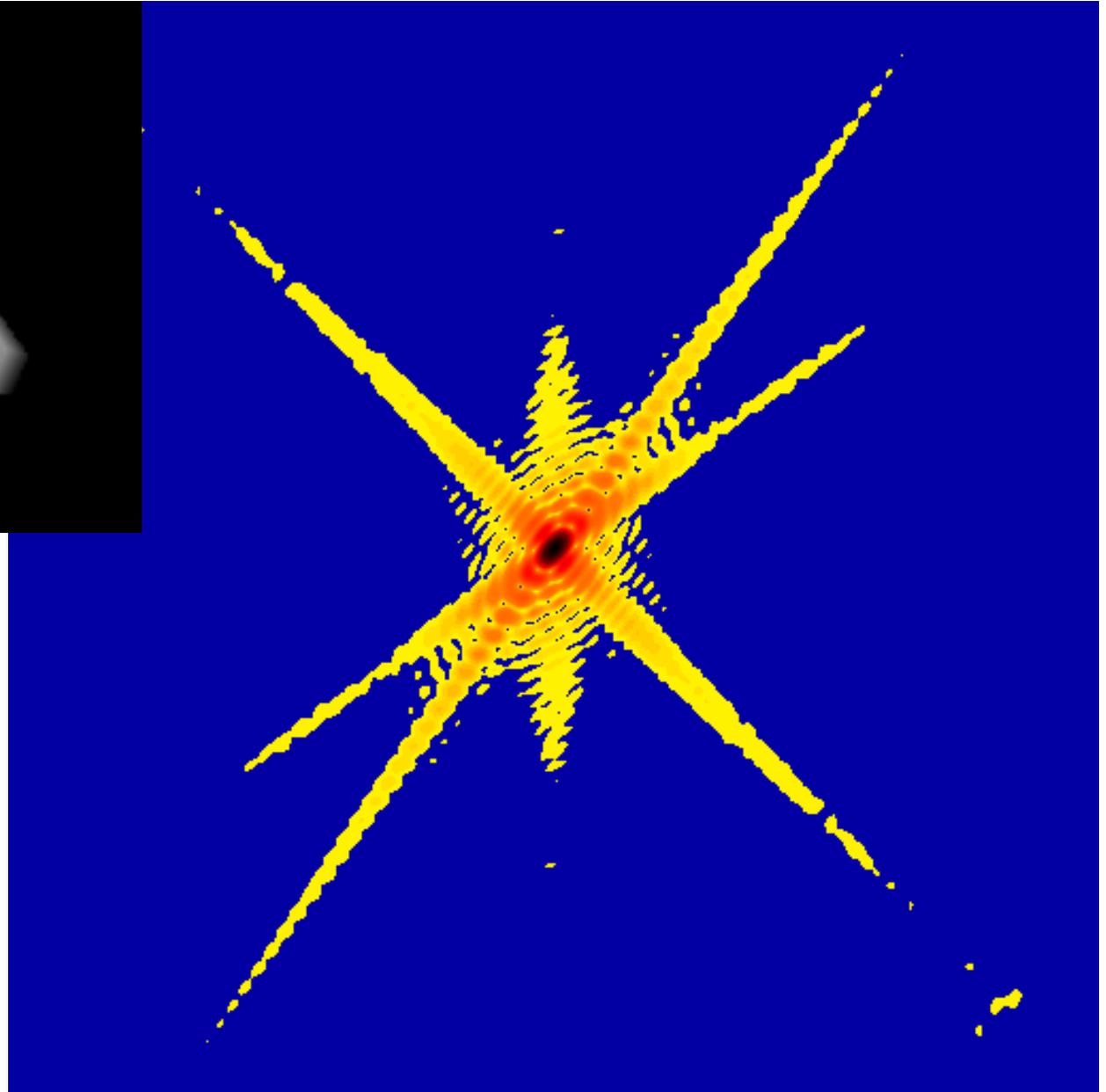
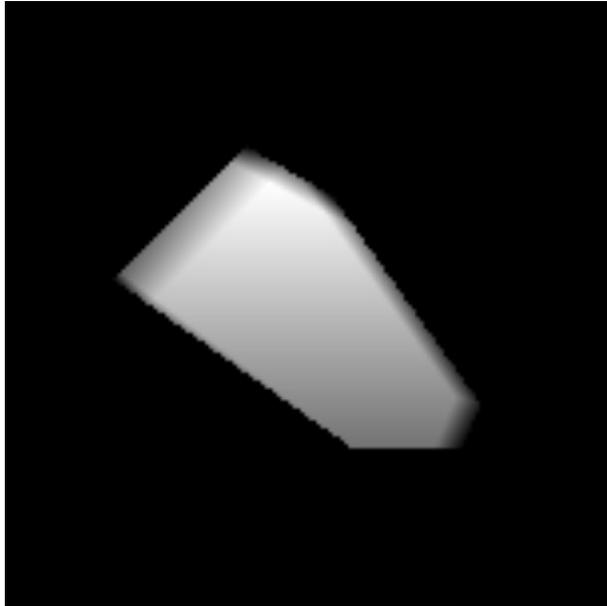
J. R. Fienup Appl. Opt. 21 2758 (1982)

R. W. Gerchberg and W. O. Saxton Optik 35 237 (1972)

# Real-space Constraints in Crystallography

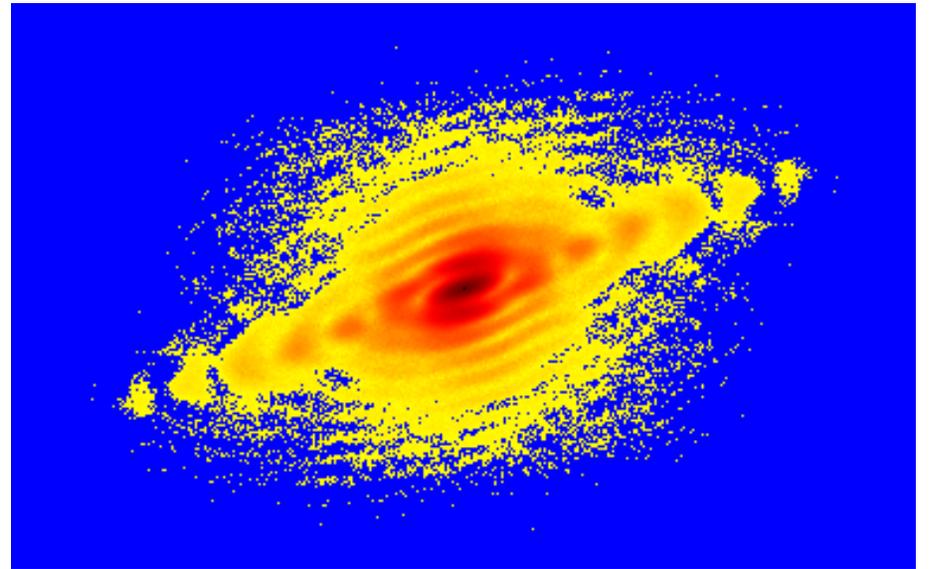
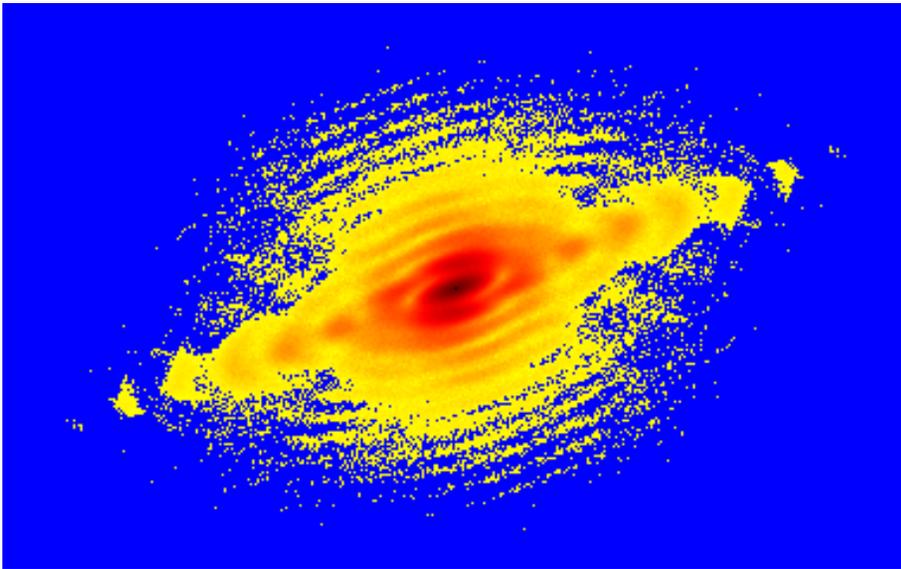
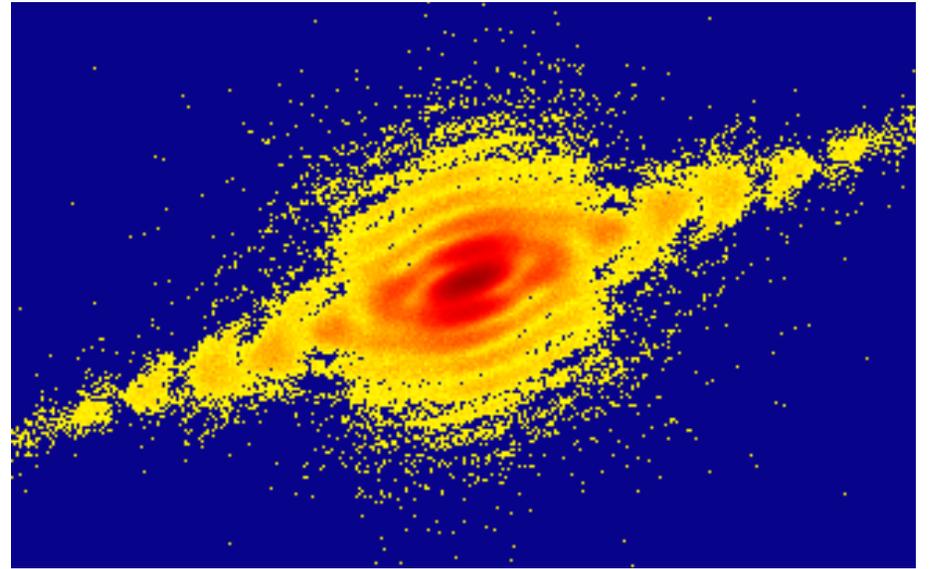
R. P. Millane, J. Opt. Soc Am. A 13 725 (1996)

- 'Positivity' and 'Atomicity' constraints (Sayre)
- Finite **support**, molecular envelope
- Solvent flattening/Molecular replacement
- Non-crystallographic symmetry
- Non-uniqueness is 'pathologically rare' ( $d > 1$ )
- Uses memory to avoid stagnation (Fienup HIO)



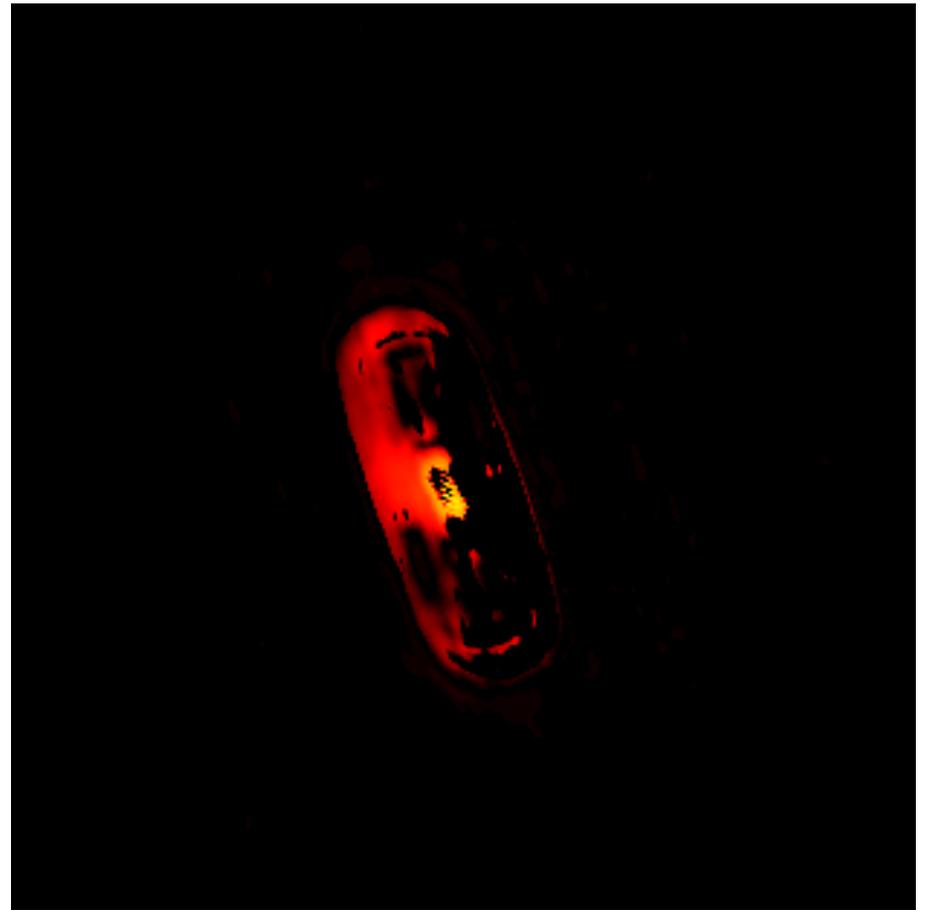
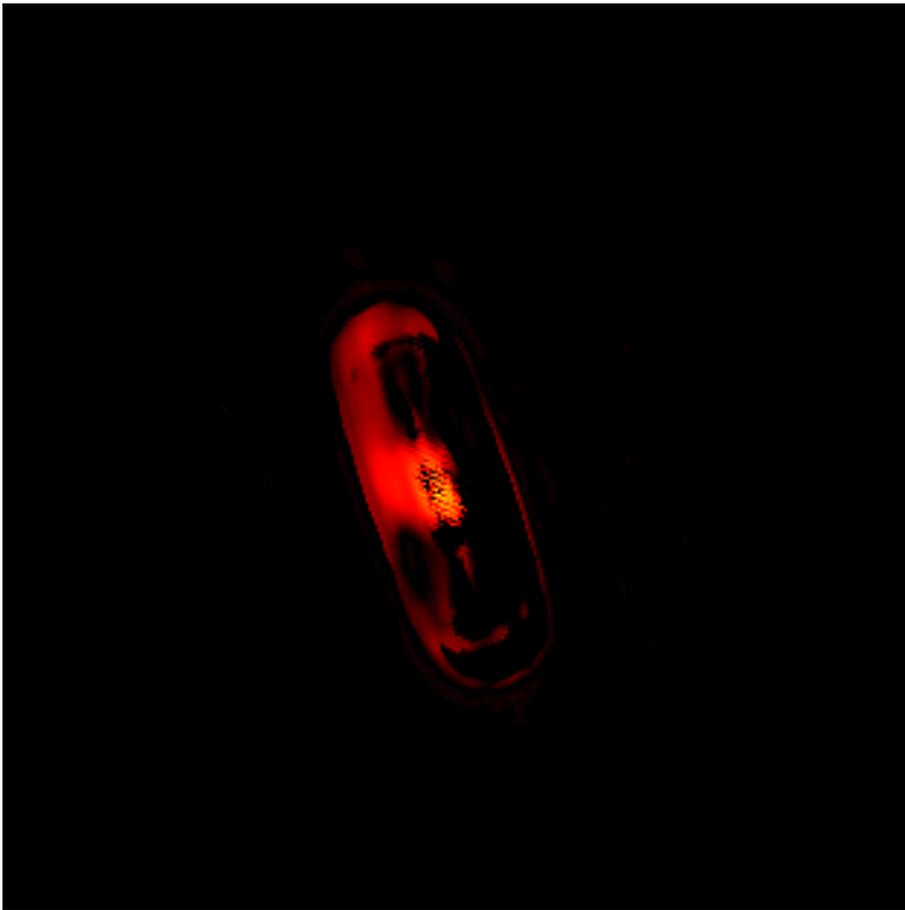
# Symmetrized Data and two best fits

$\text{Chisq}=0.0005$



# 2D Reconstructions

chisquare = 0.0005



# 3D Diffraction Method

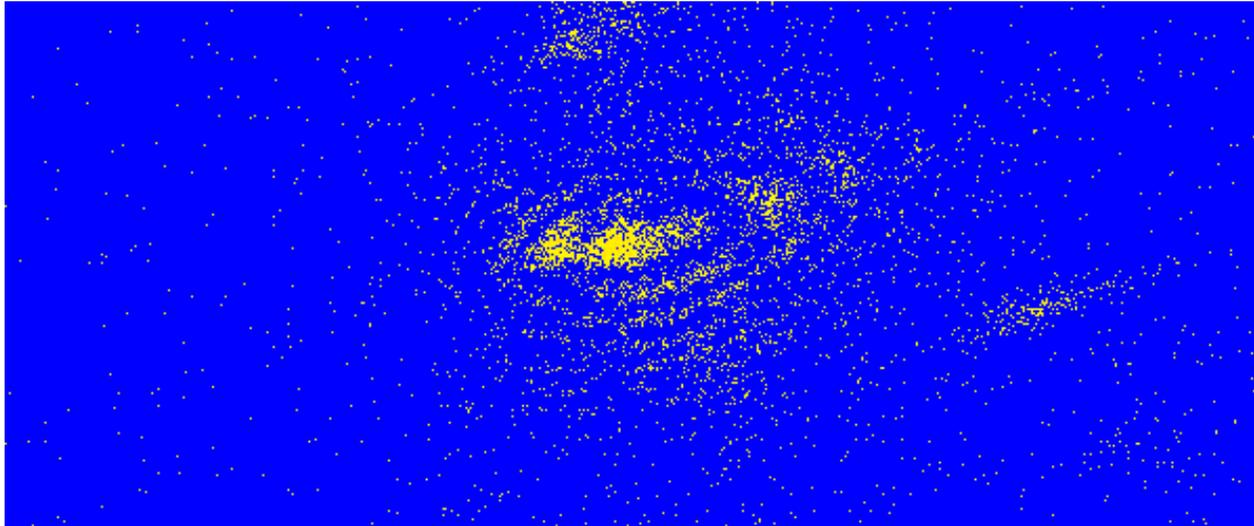
$k_f$

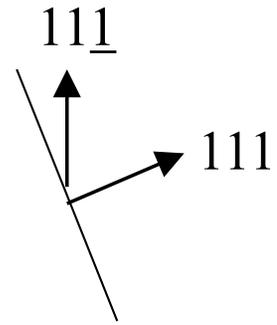
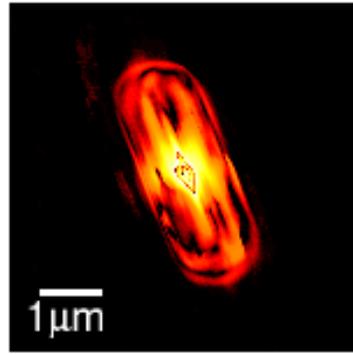
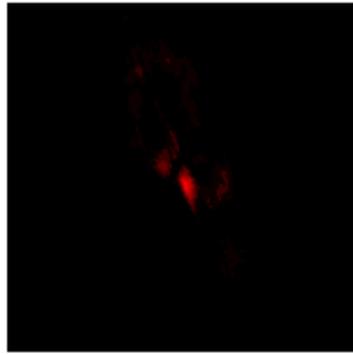


$Q = k_f - k_i$

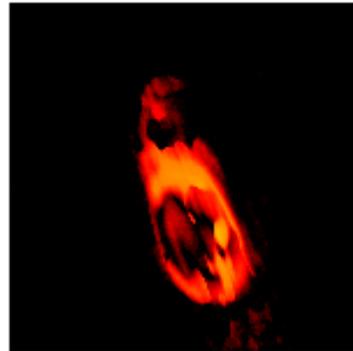
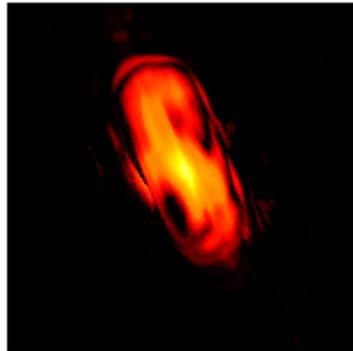
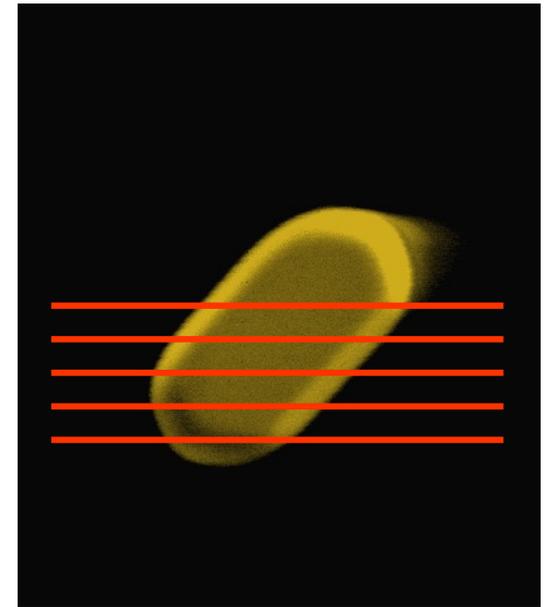
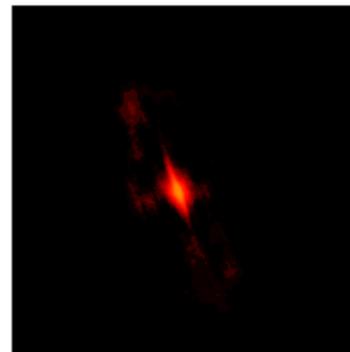
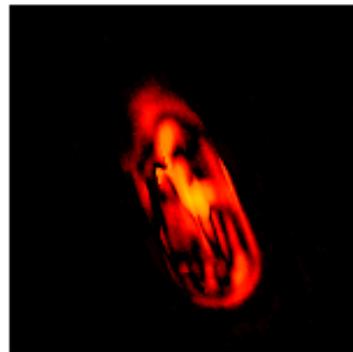
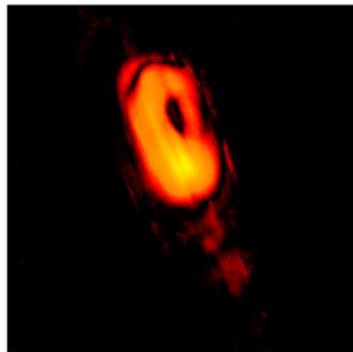
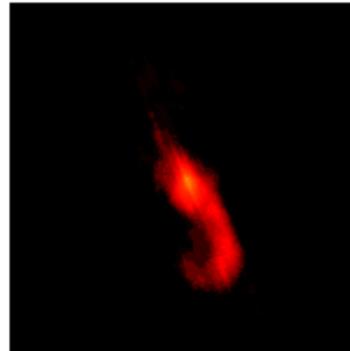
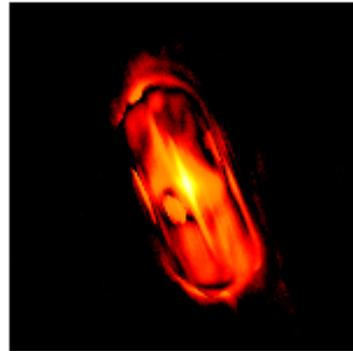
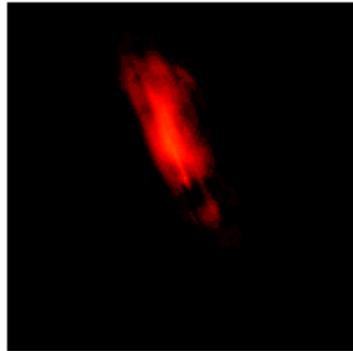
$k_i$

# 3D Diffraction Data 1 micron Au crystal



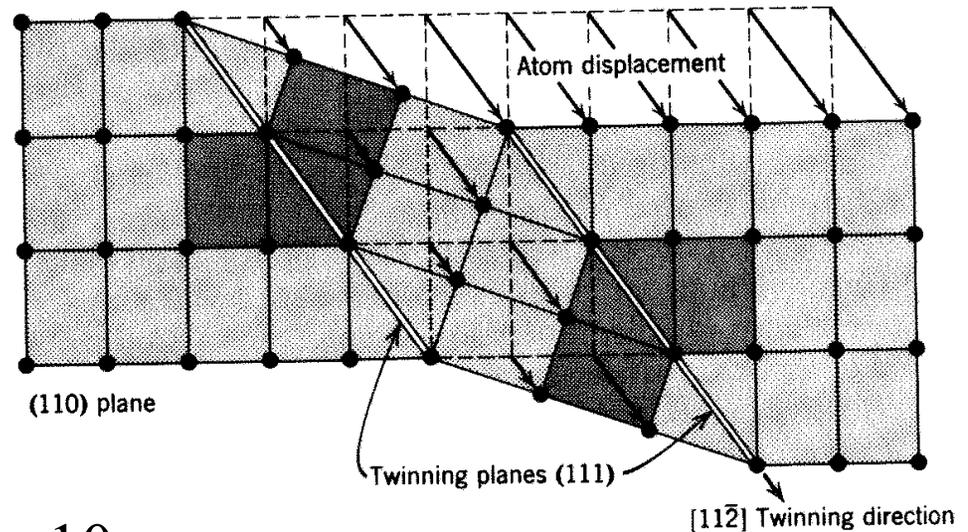


Slices through  
plan view SEM:



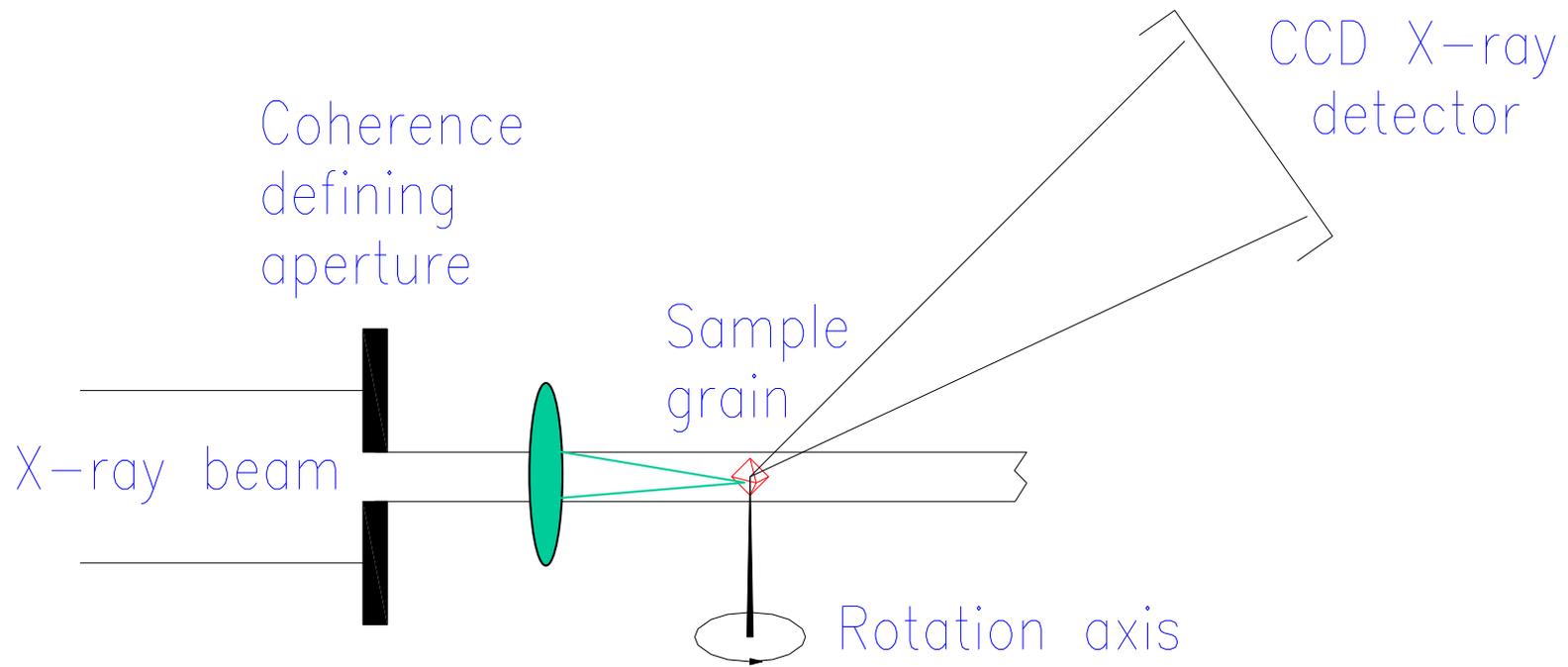
# Twinning in deformed FCC metals

J. Wulff, "Structure and Property of Materials III" (1965)



■  $\sim 10\mu\text{m}$   
Cu

# Lensless X-ray Microscope

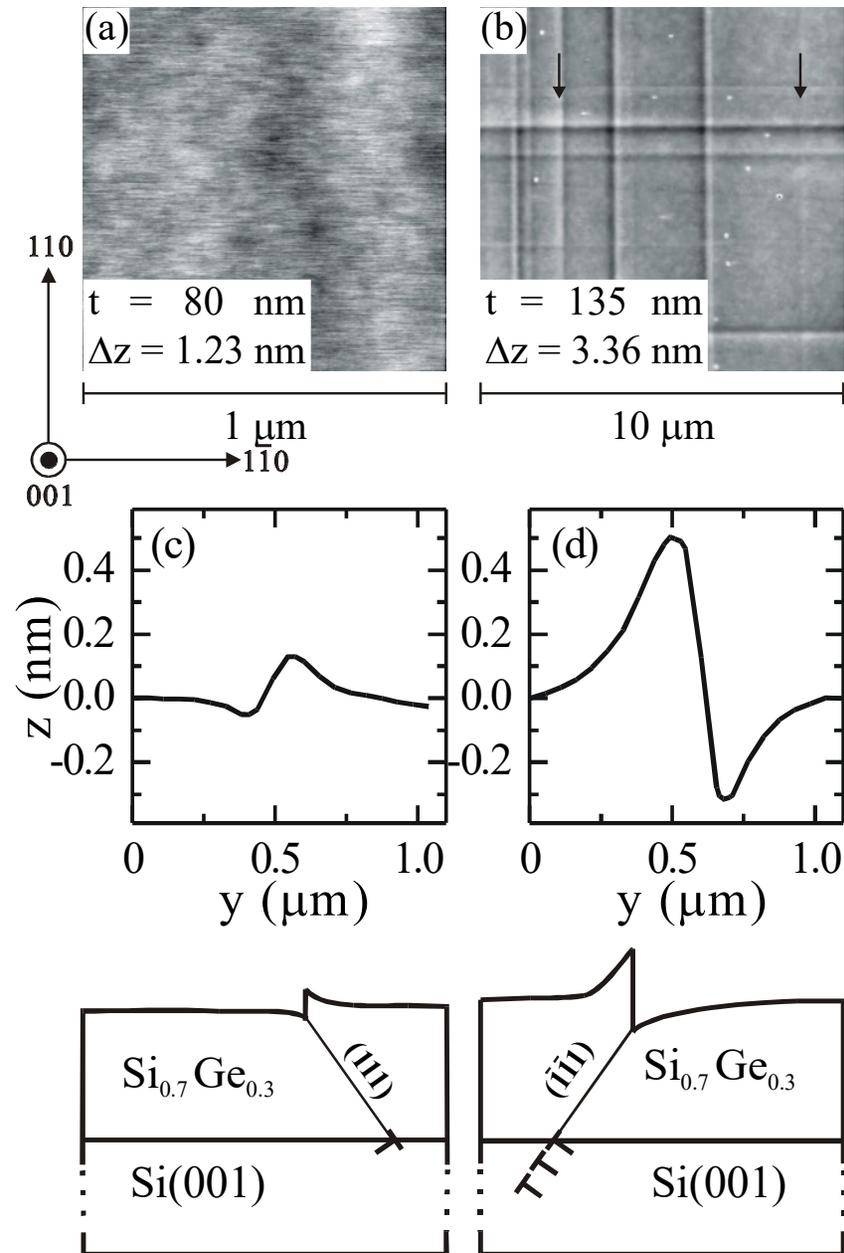


Low dislocation density GeSi films

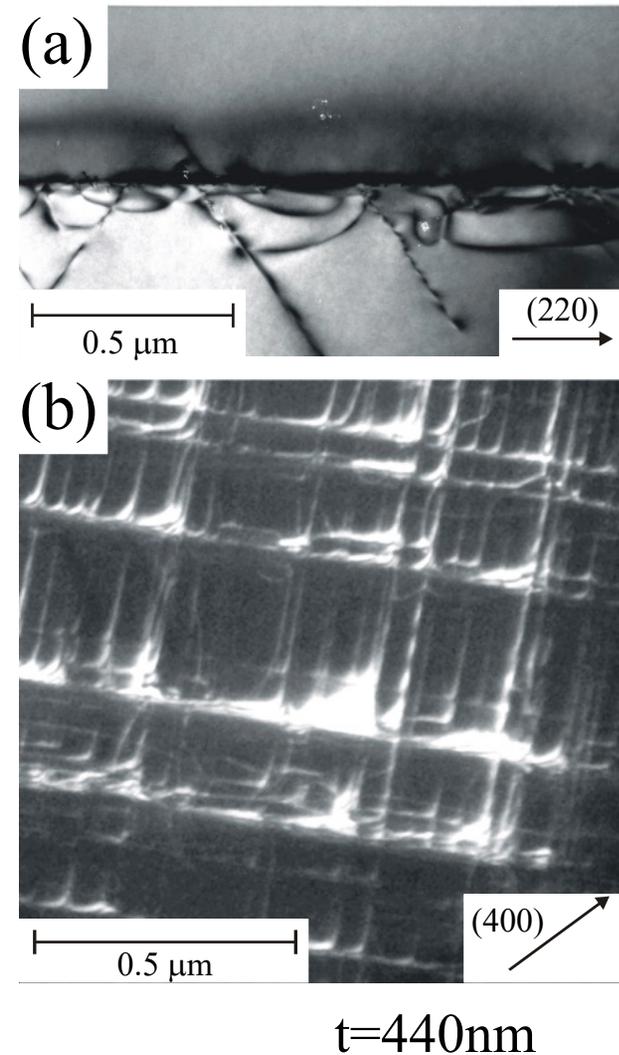
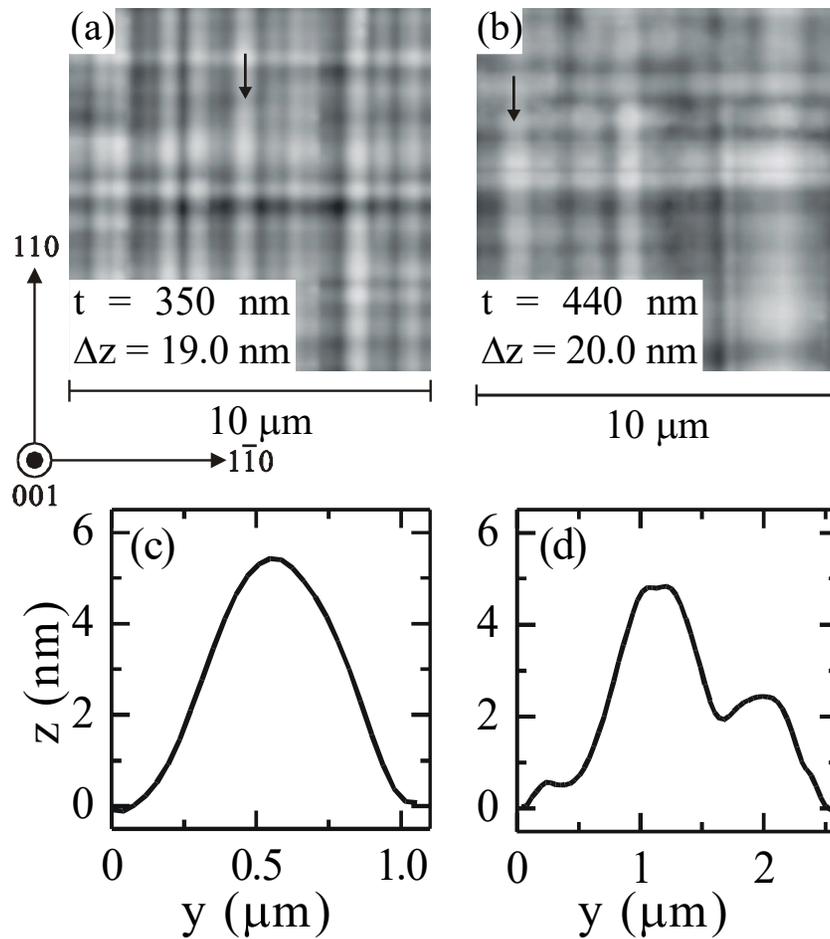
Thickness close to critical thickness

Dislocations aggregate at interface and glide to surface along {111}

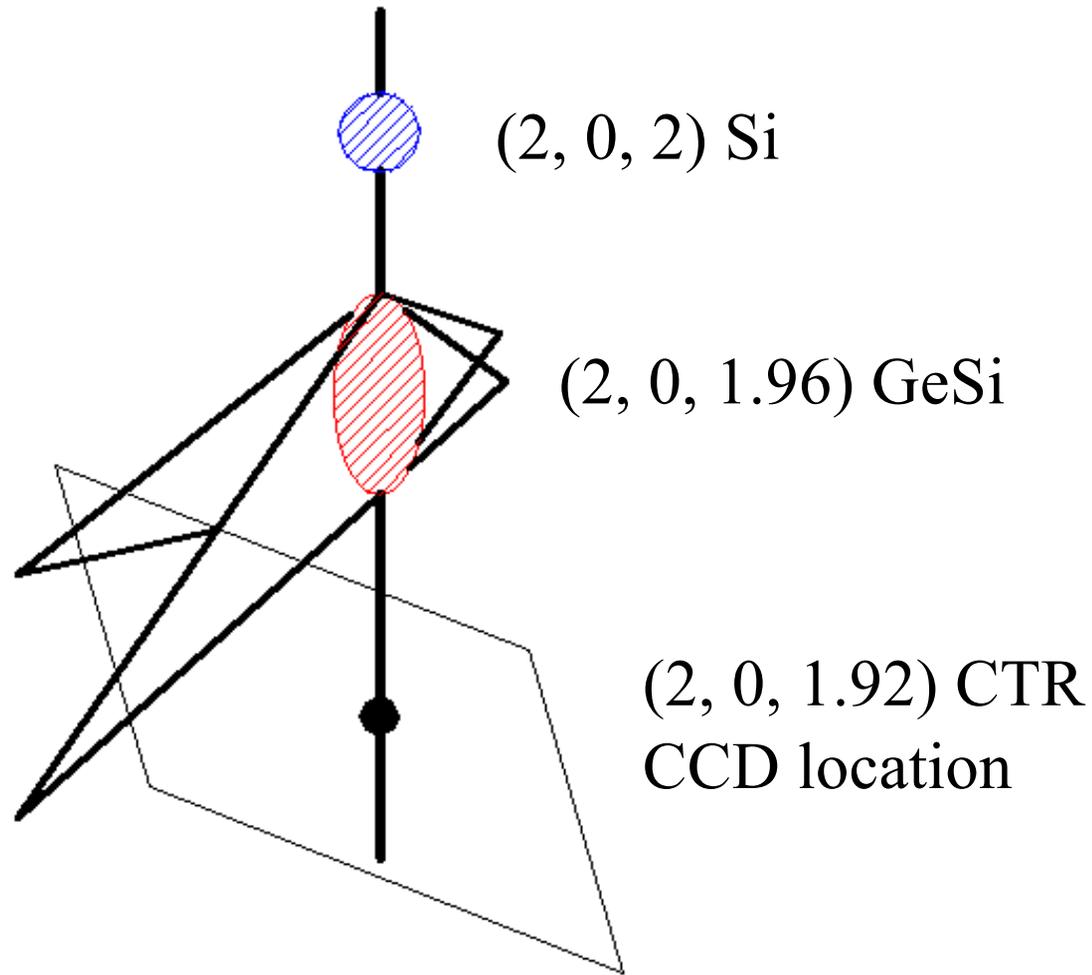
T. Spila, UIUC Thesis



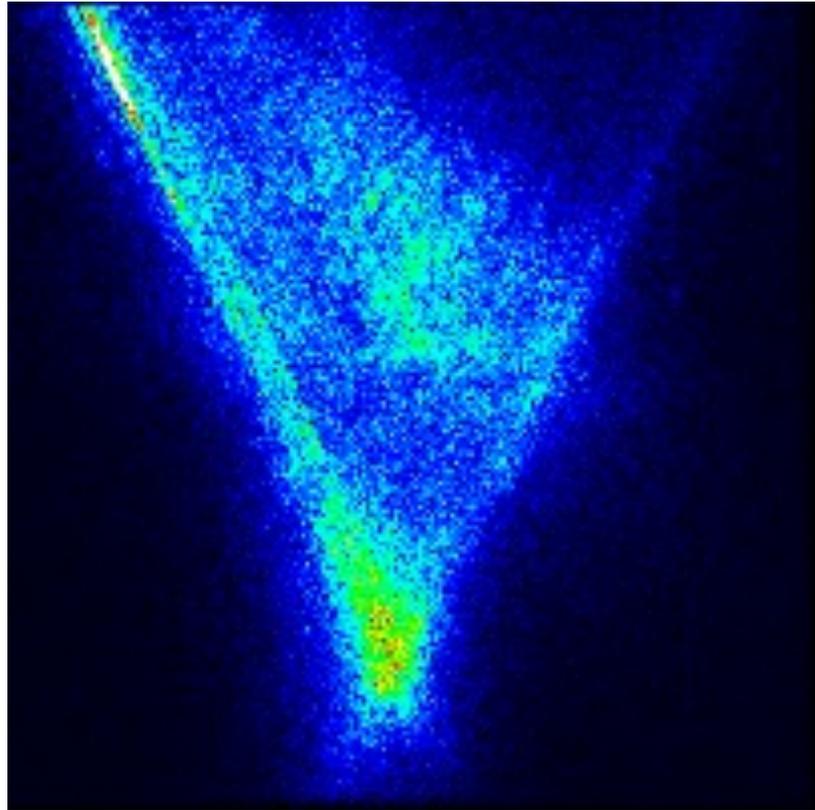
# Higher Dislocation Density $\text{Ge}_{0.3}\text{Si}_{0.7}$



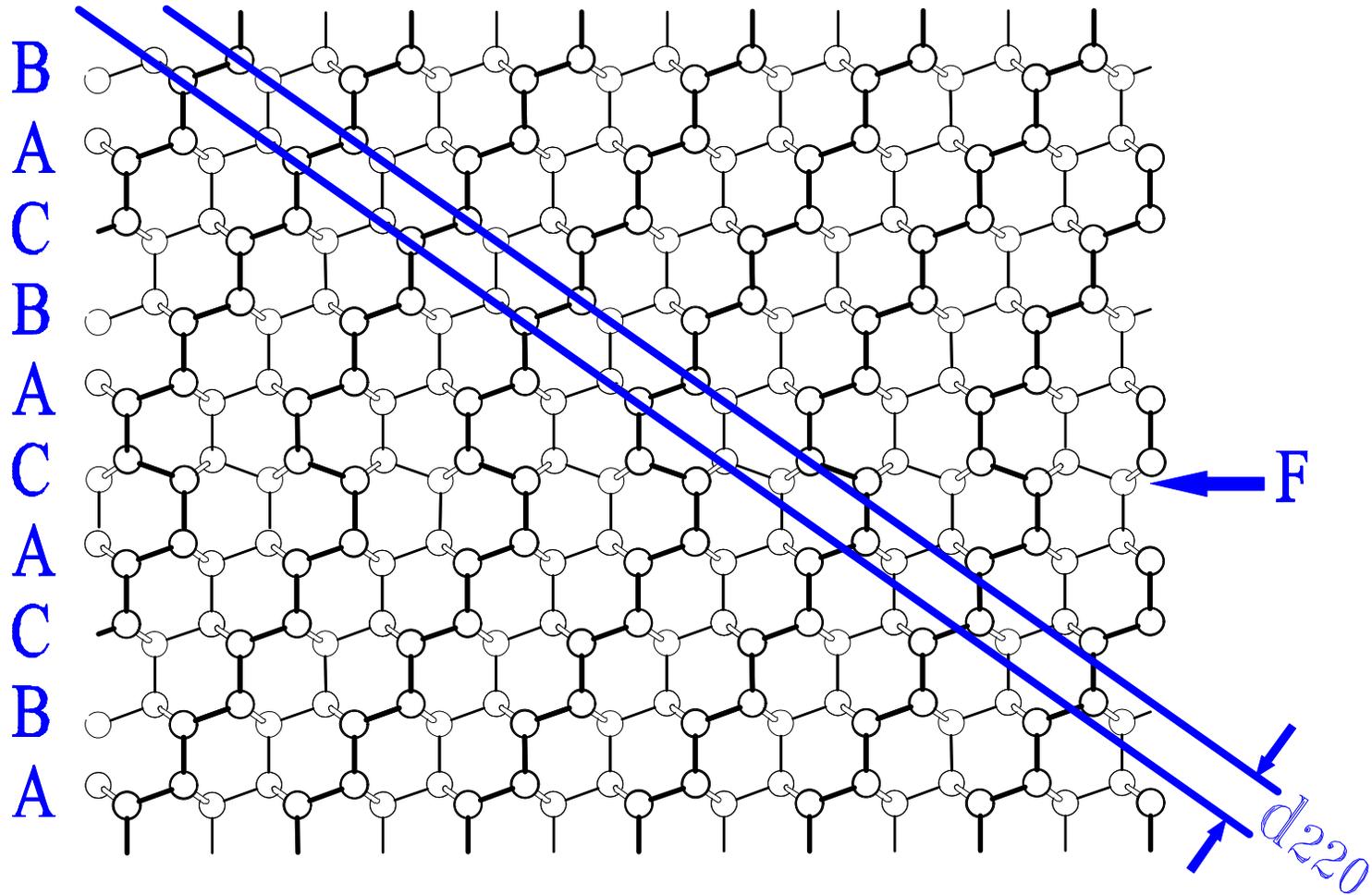
# Dislocations in Reciprocal Space



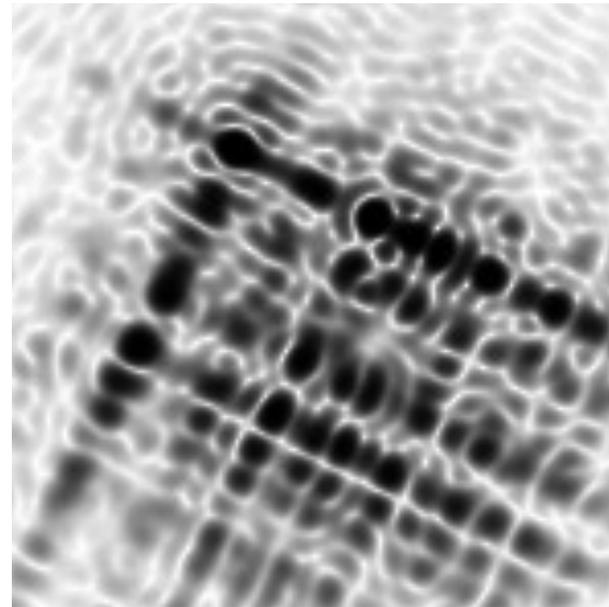
Use CCD and scan sample position  
(2,0,1.92) 280nm thick  $\text{Ge}_{0.3}\text{Si}_{0.7}$



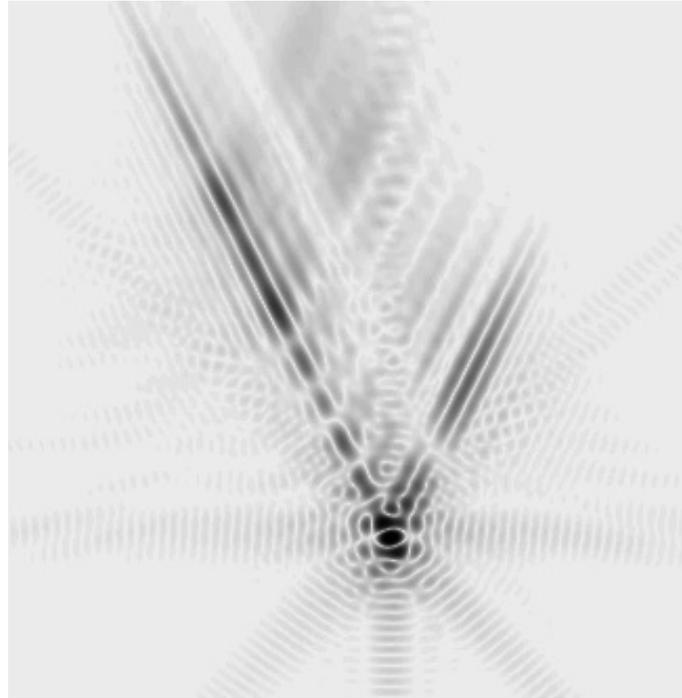
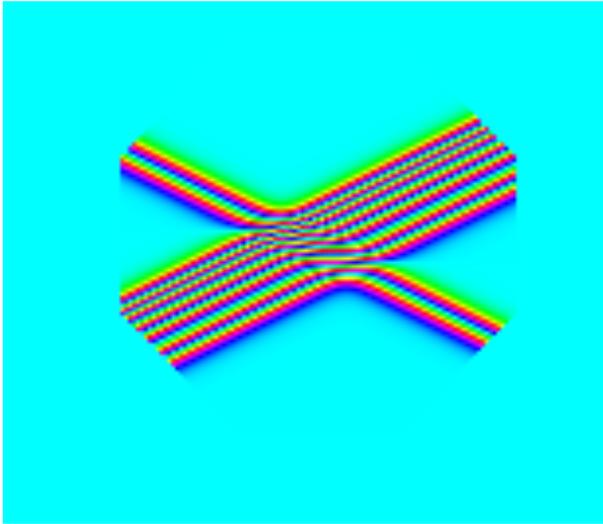
# Deformation fault in Silicon



# Dislocation Diffraction



# Simulation of GeSi data



# Conclusions and Outlook

- Inversion of CXD demonstrated
- Internal structure of Au Nanocrystals
- Phasing by computation instead of lens
- Preservation of coherence upon focussing
- Dislocations give asymmetric patterns
- Strain fields should be accessible by inversion