

Limits of Coherent X-ray Diffraction for Imaging Small Crystals

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- Franz Pfeiffer
- Mark Pfeifer
- Garth Williams

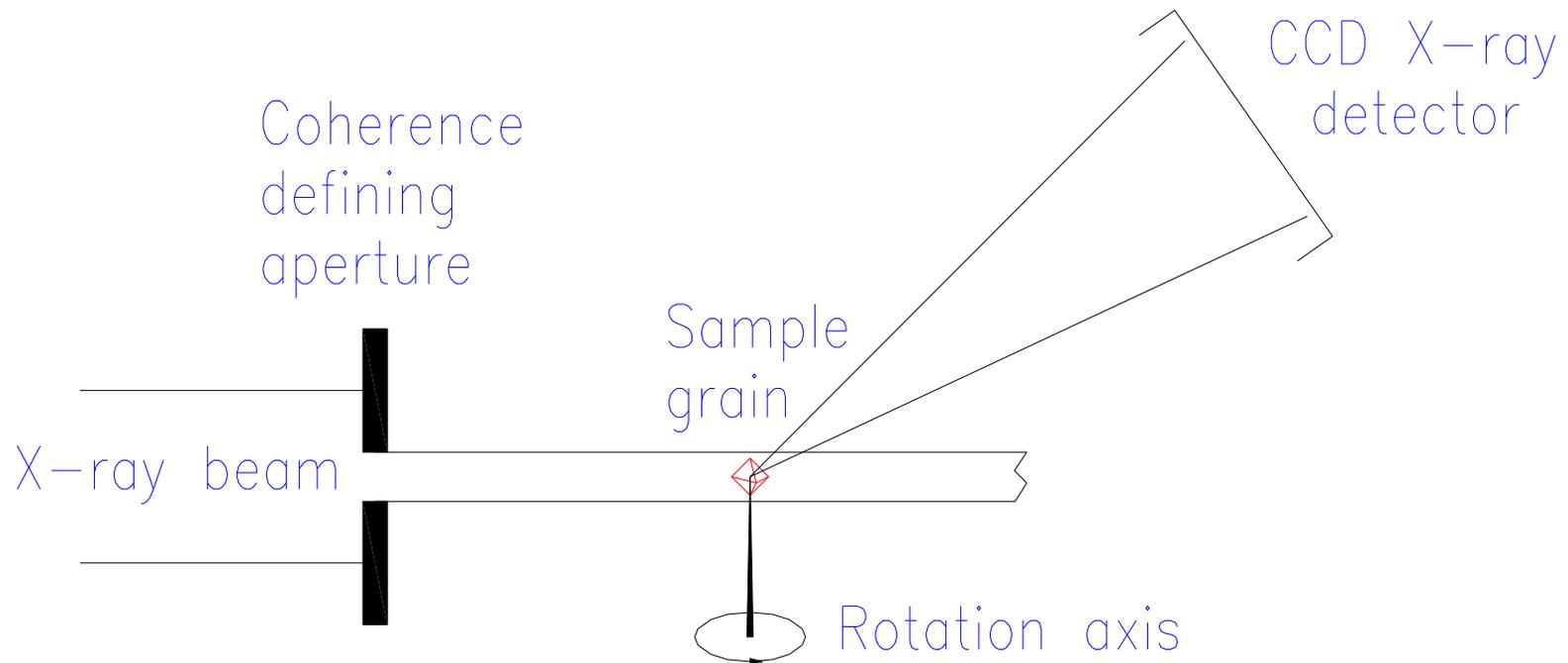
Department of Physics
University of Illinois

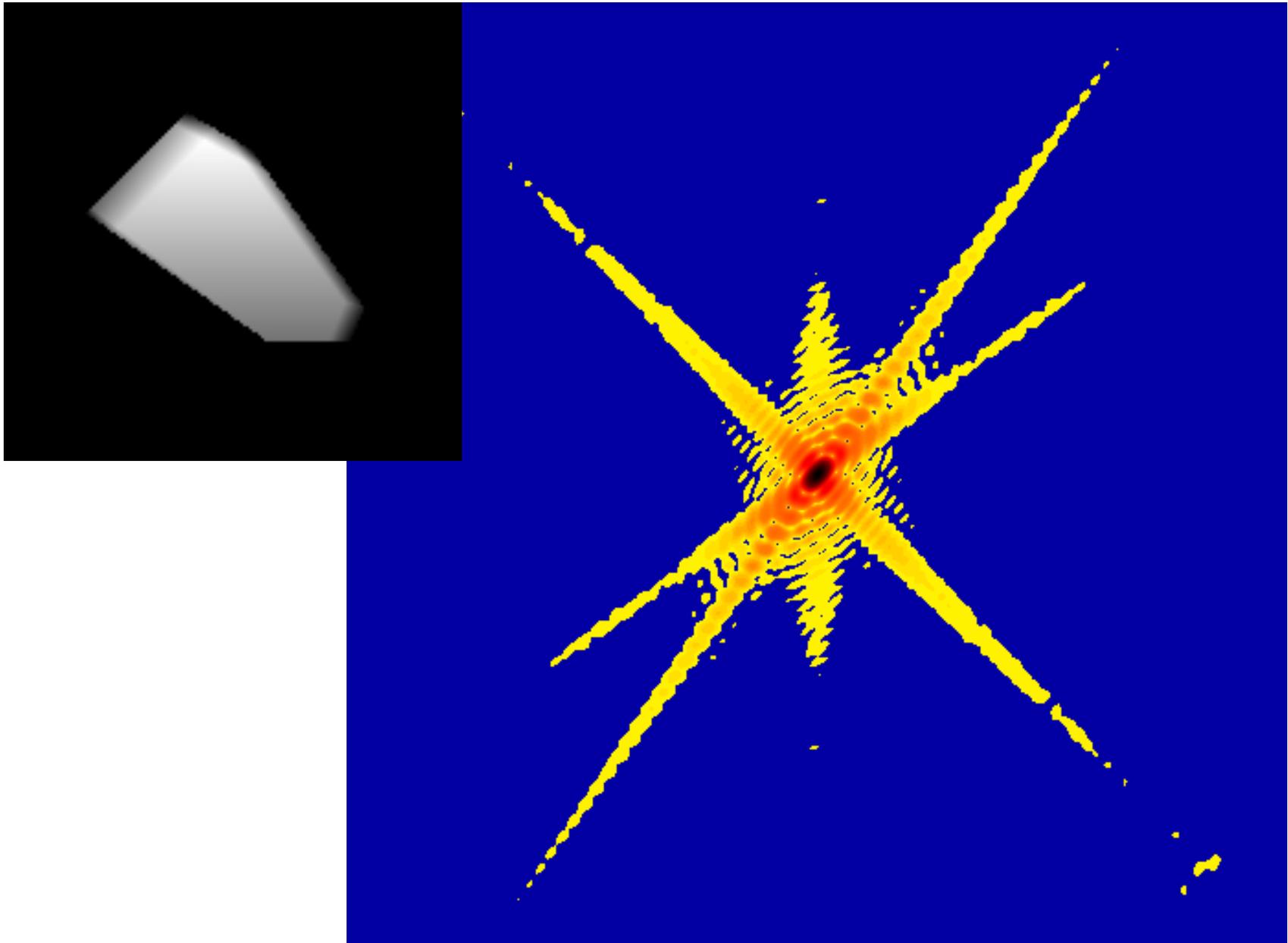
Second International Workshop on
Noncrystallographic Phase Retrieval

Outline

- Nanocrystal Shapes
- Vortices During Phasing
- How small can we go?
- Future Directions of CXD

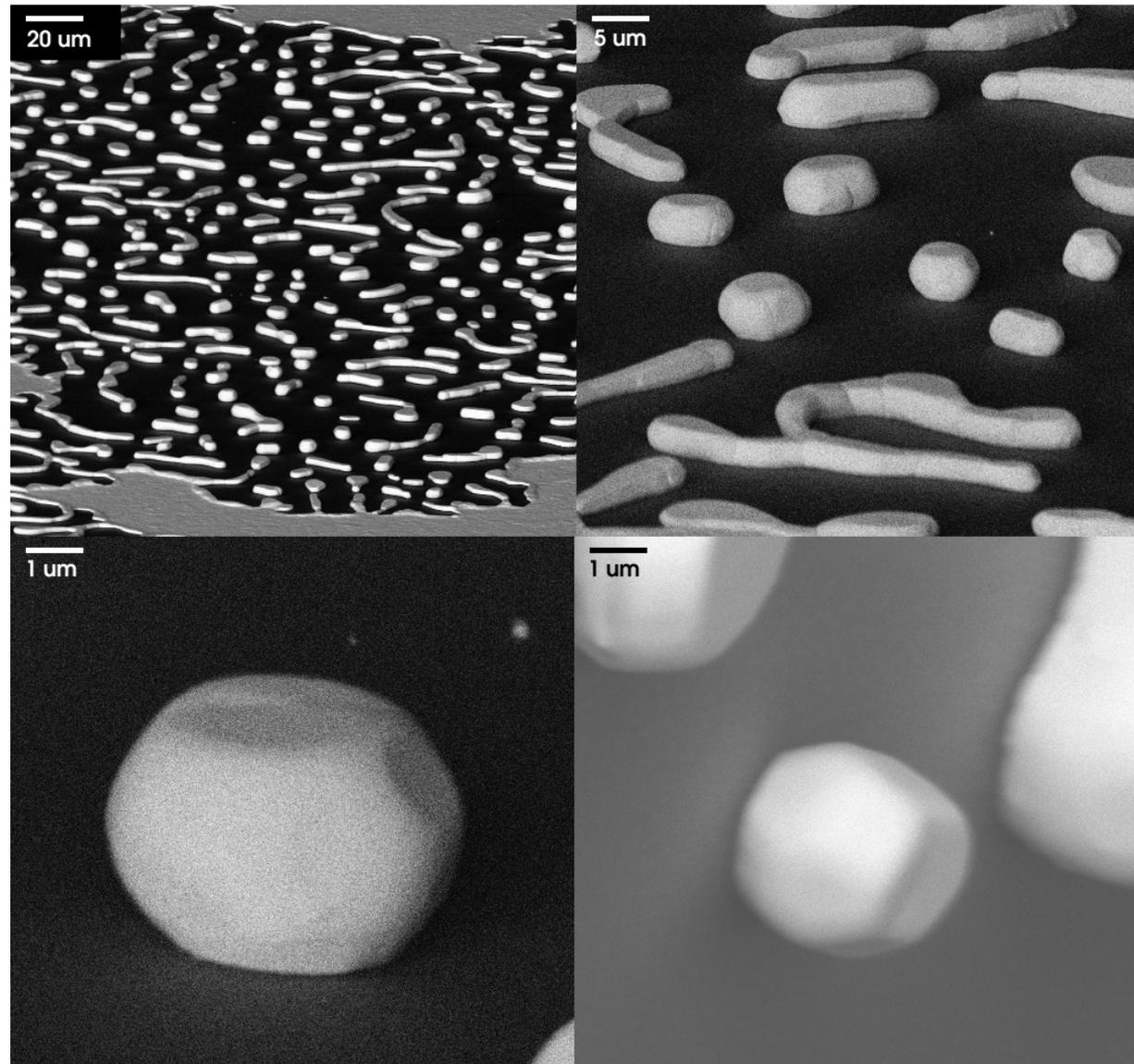
Lensless X-ray Microscope



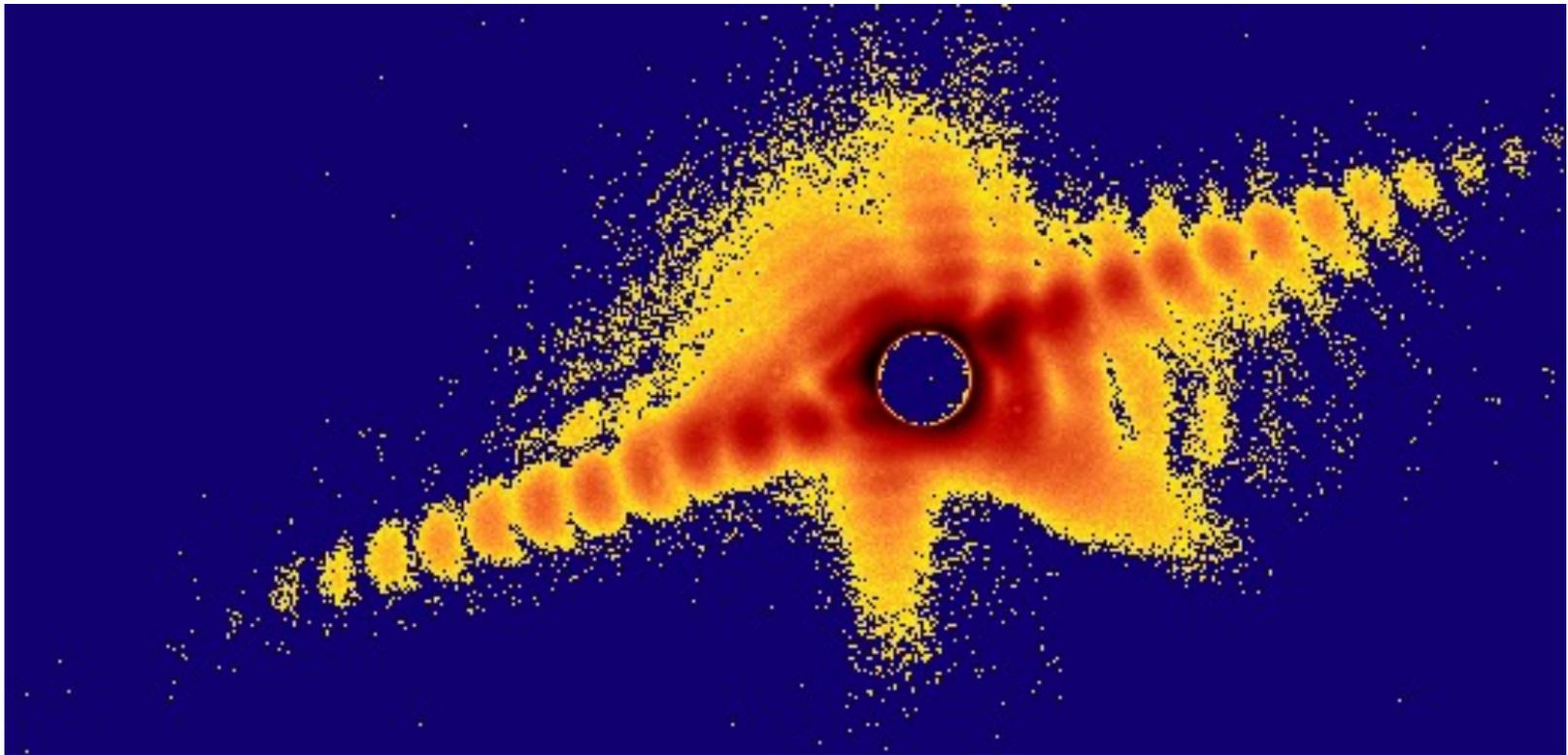


SEMS

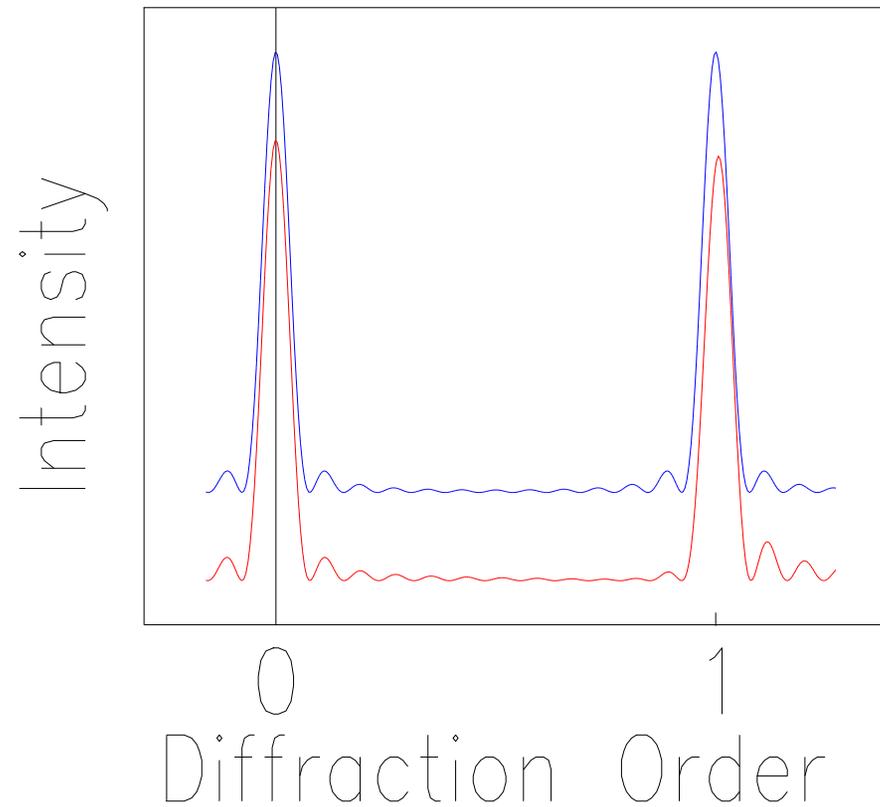
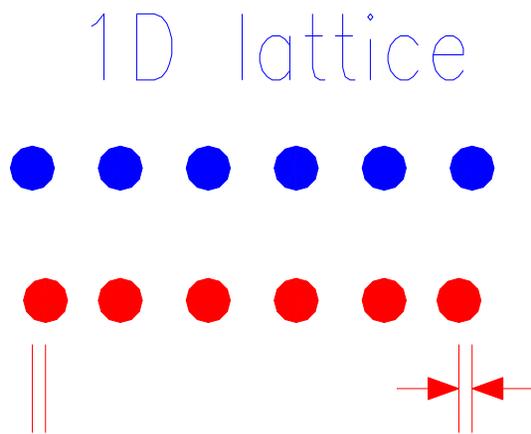
- Au blanket film
- Quartz substrate
- Annealed at 950°C for 70 hrs.



Micron-sized gold crystal: (111) Bragg reflection

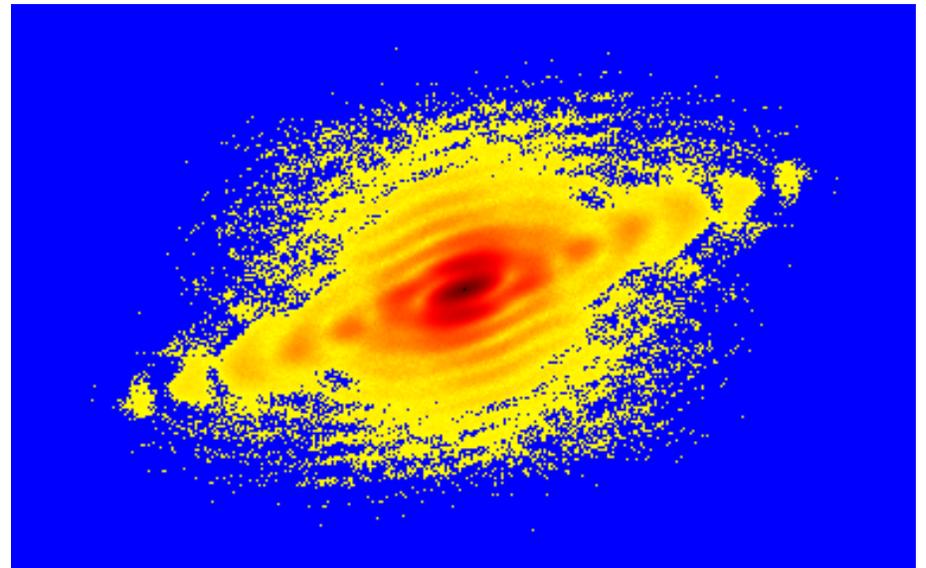
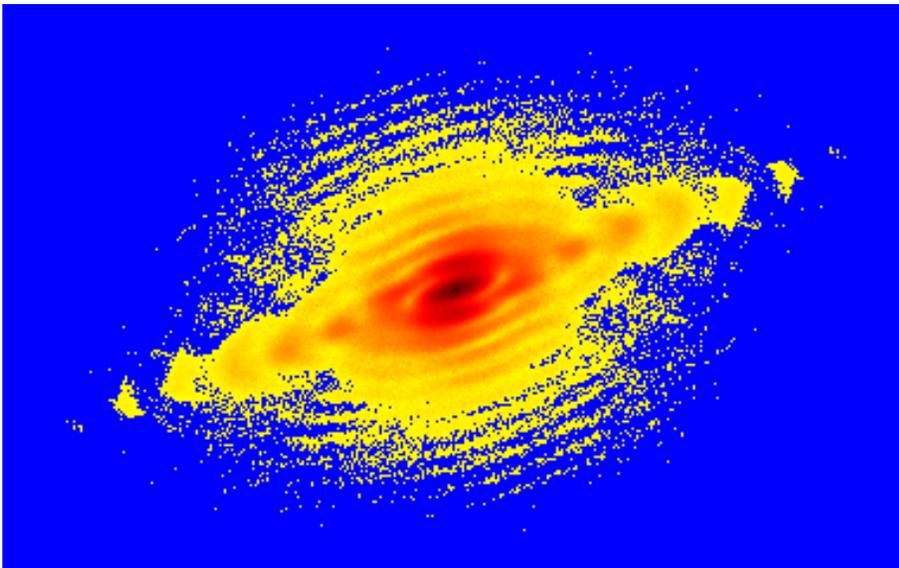
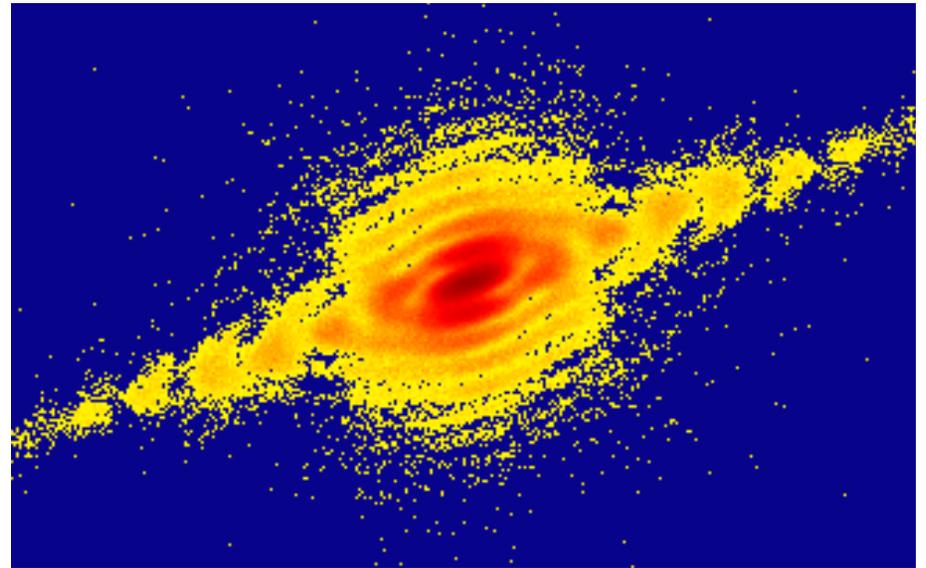


Imaging of Lattice Strains



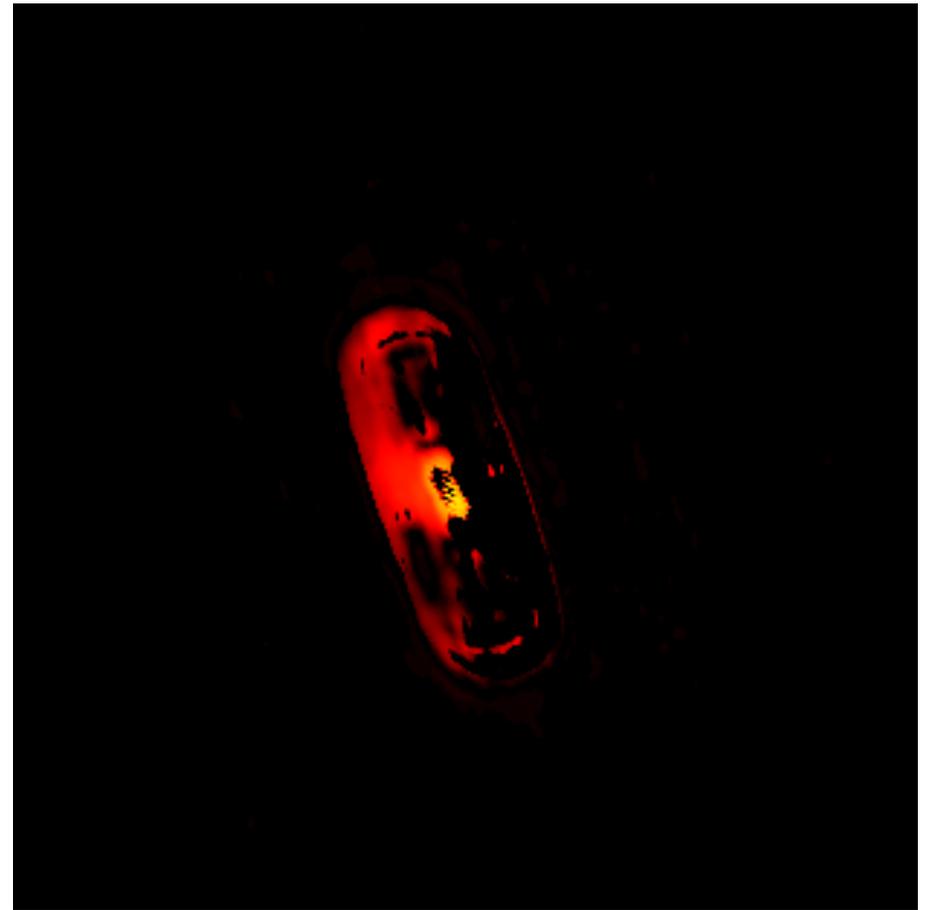
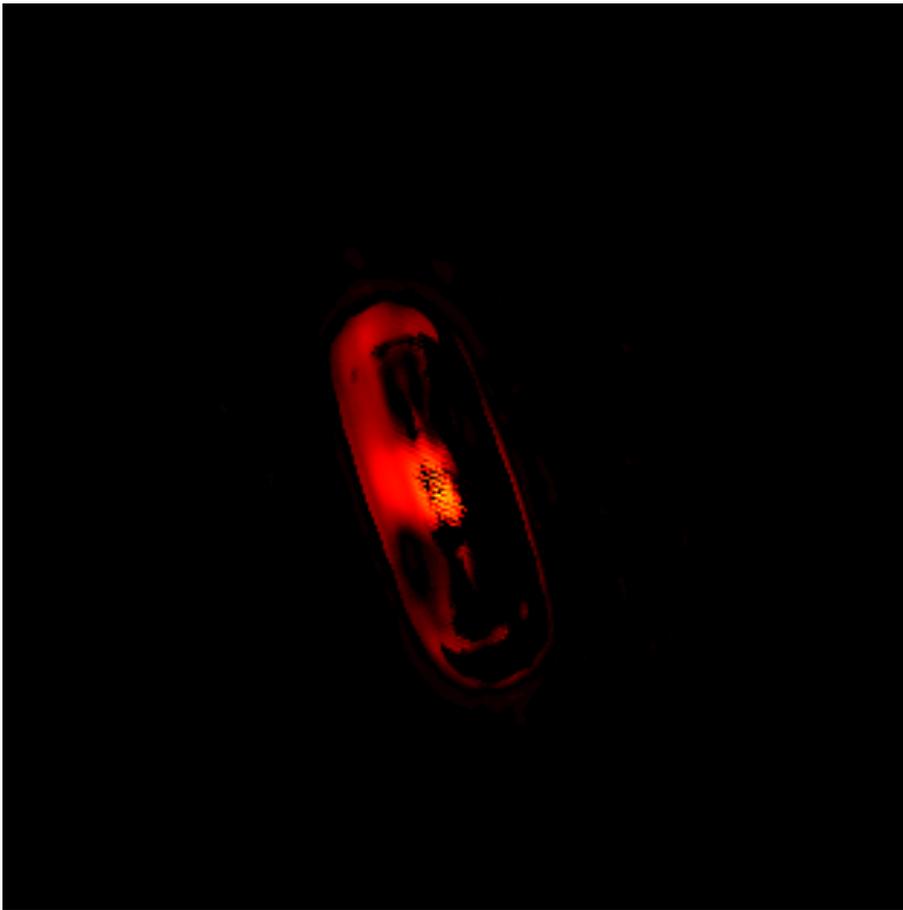
Symmetrized Data and two best fits

Chisq=0.0005

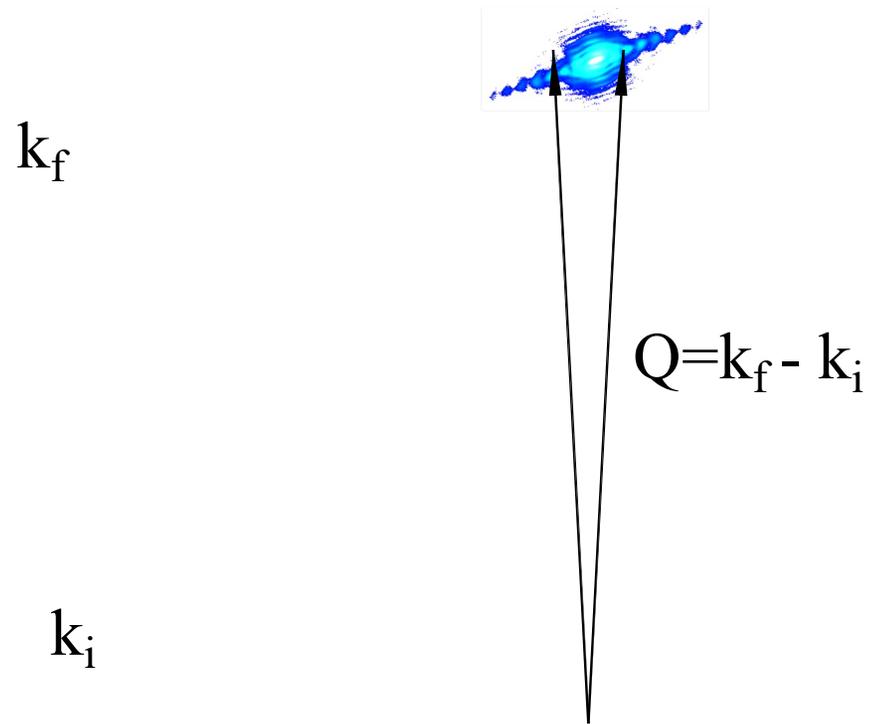


2D Reconstructions

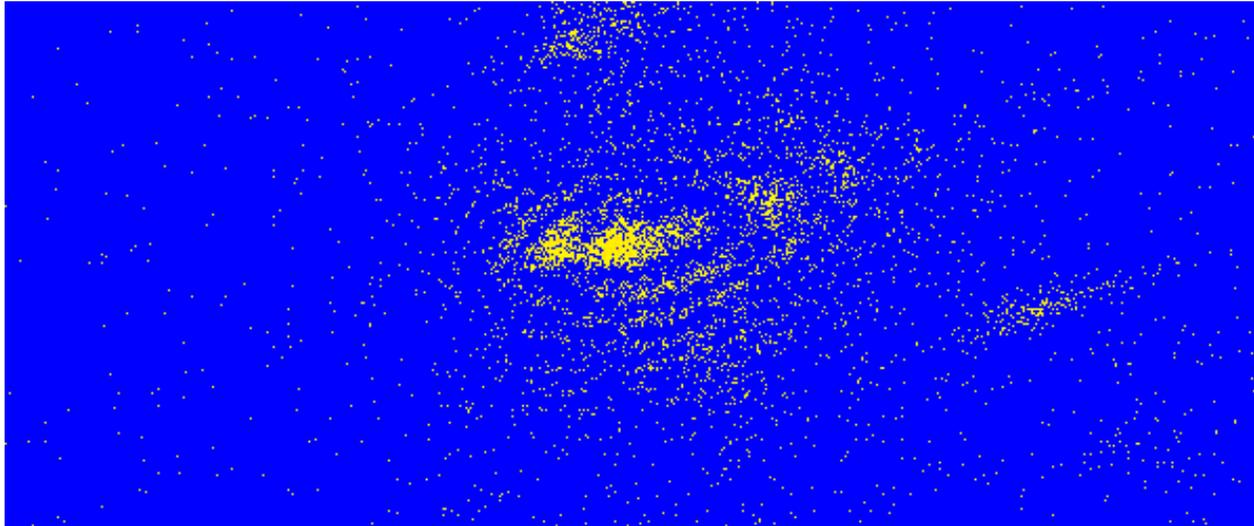
chisquare = 0.0005



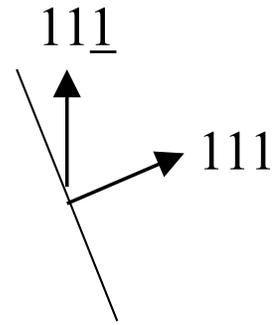
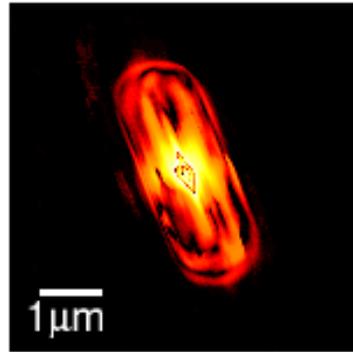
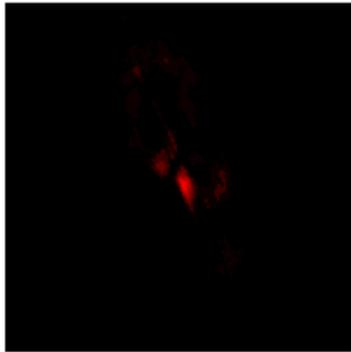
3D Diffraction Method



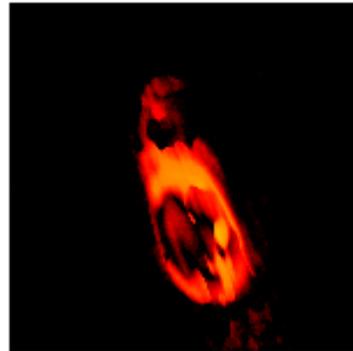
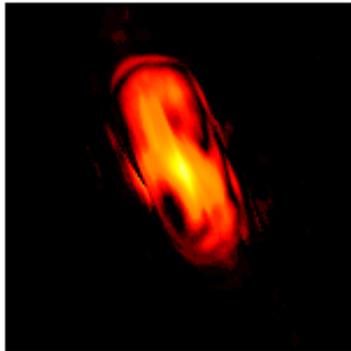
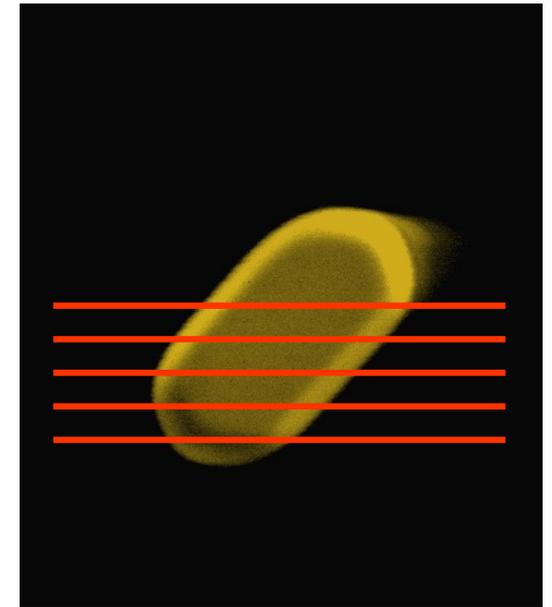
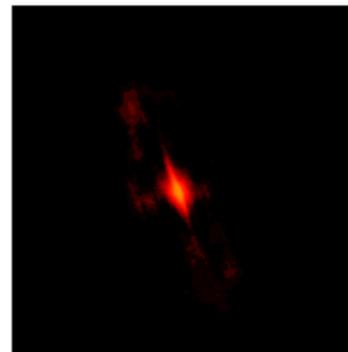
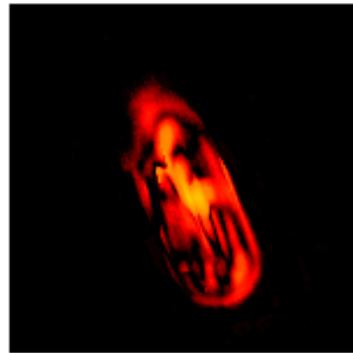
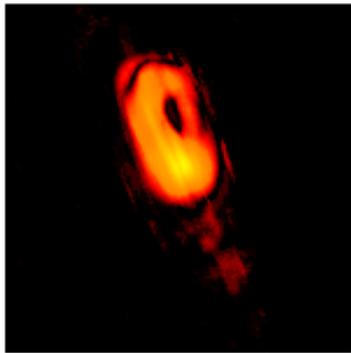
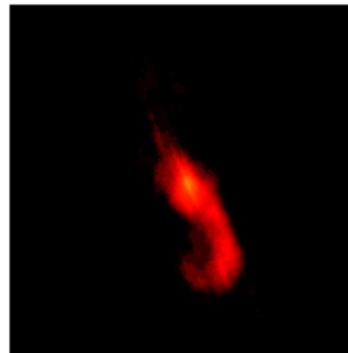
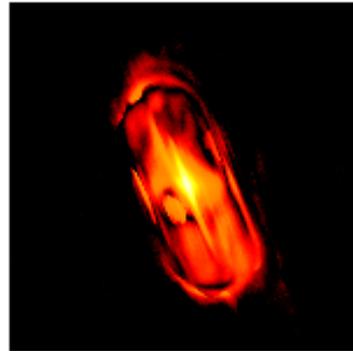
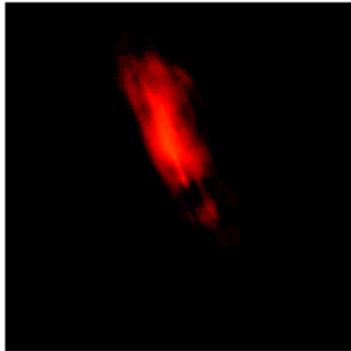
3D Diffraction Data 1 micron Au crystal



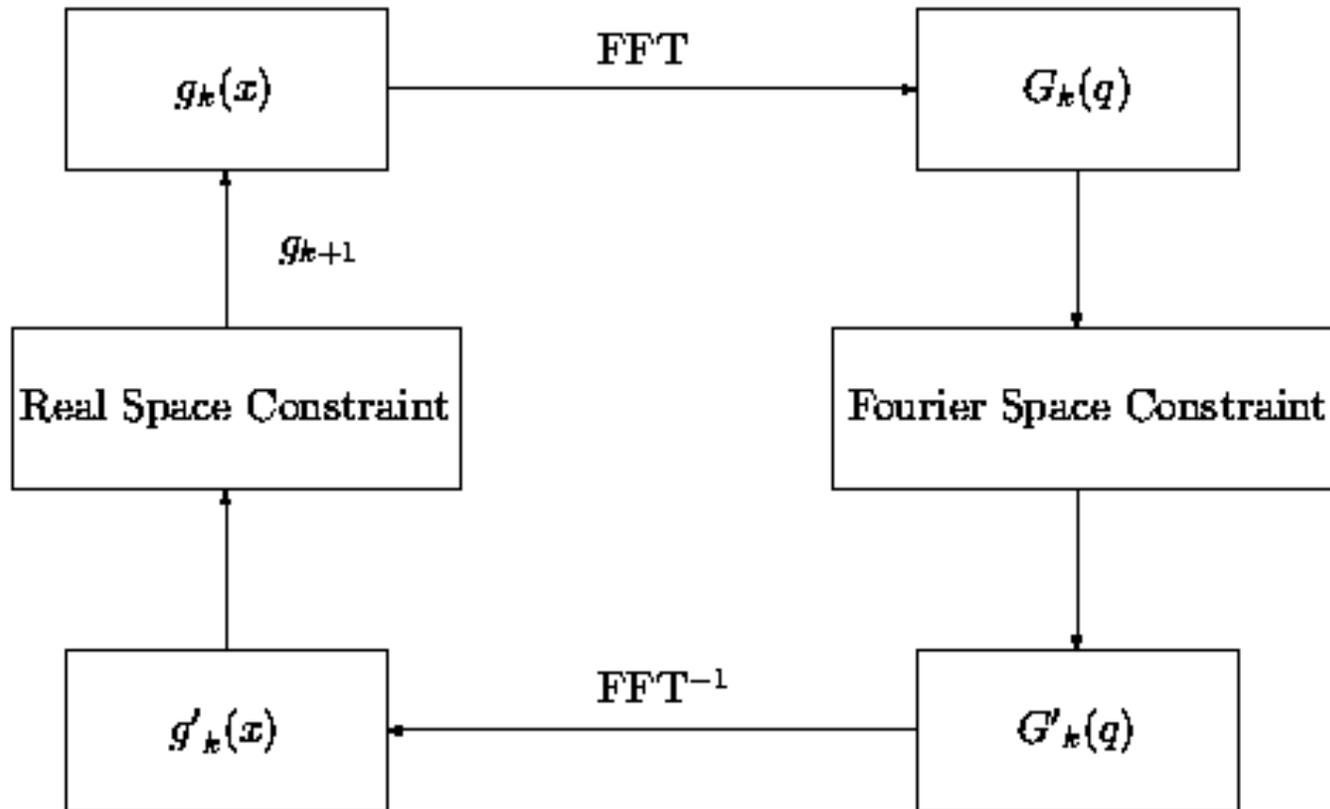
* Center is Symmetric *



Slices through
plan view SEM:



Generic “Error Reduction” method



J. R. Fienup *Appl. Opt.* 21 2758 (1982)

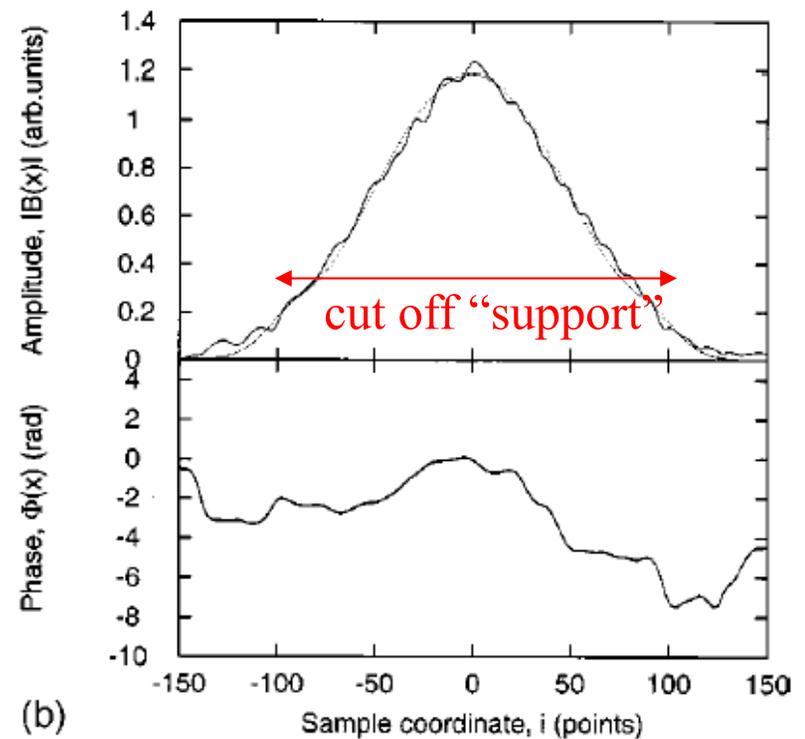
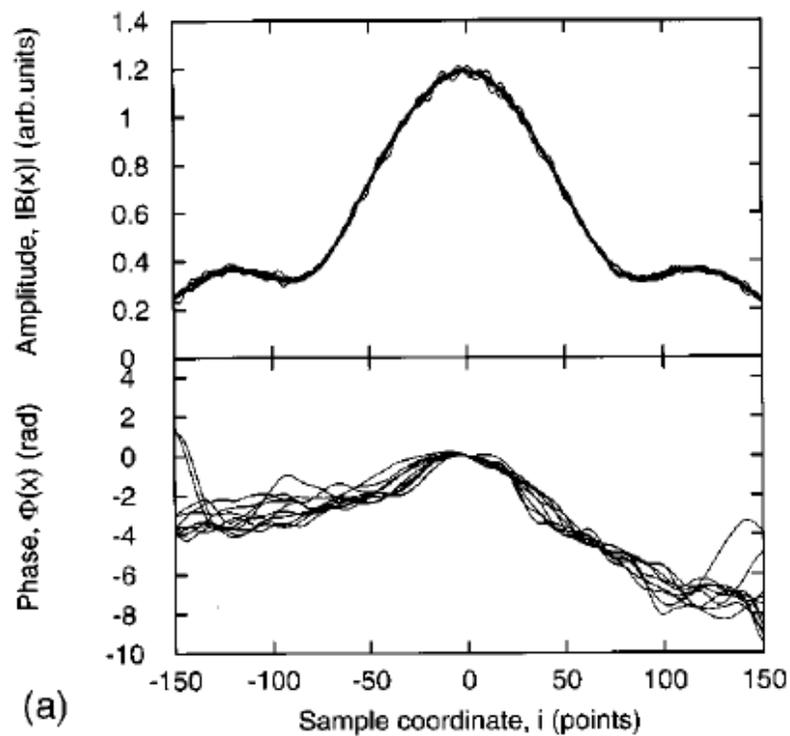
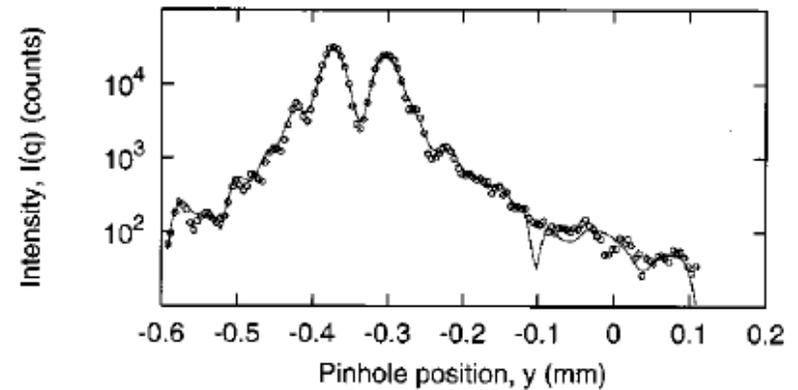
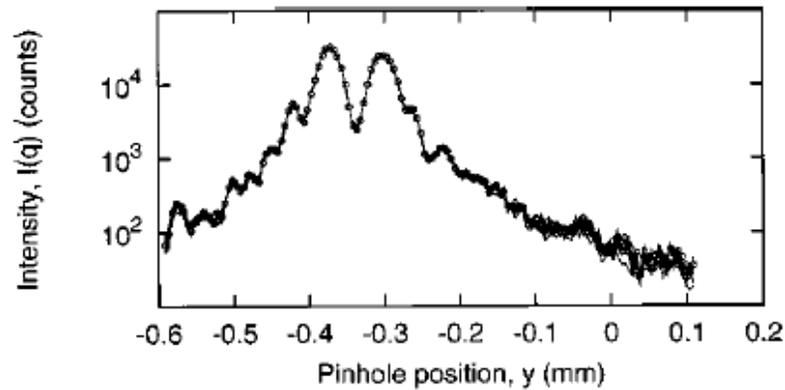
R. W. Gerchberg and W. O. Saxton *Optik* 35 237 (1972)

Real-space Constraints in Crystallography

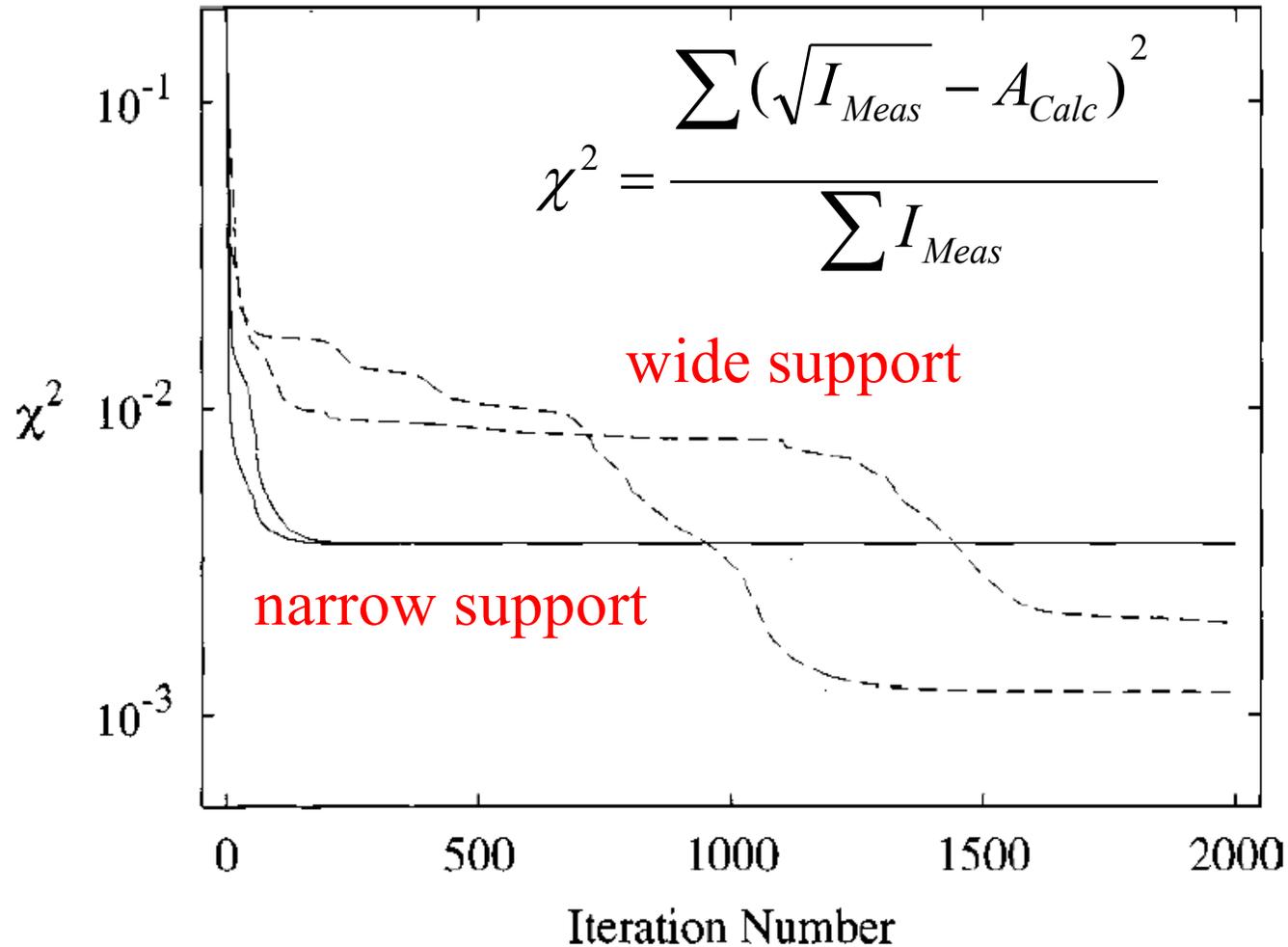
R. P. Millane, J. Opt. Soc Am. A **13** 725 (1996)

- ‘Positivity’ constraint (Sayre)
- Finite **support**, molecular envelope
- Solvent flattening
- Molecular replacement
- Non-crystallographic symmetry
- Non-uniqueness is ‘pathologically rare’ ($d > 1$)

Phasing using G-S Algorithm

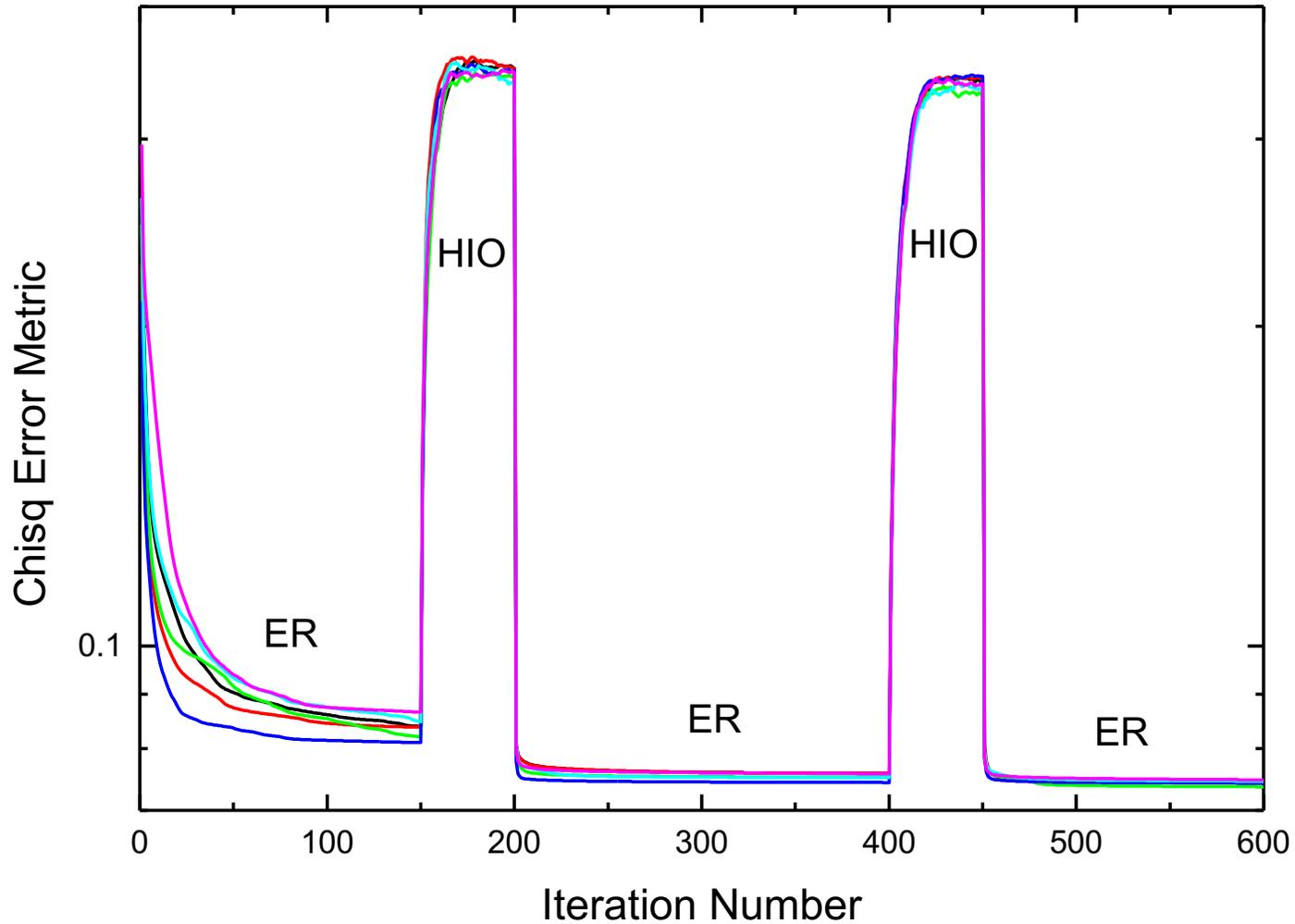


Convergence Trajectory



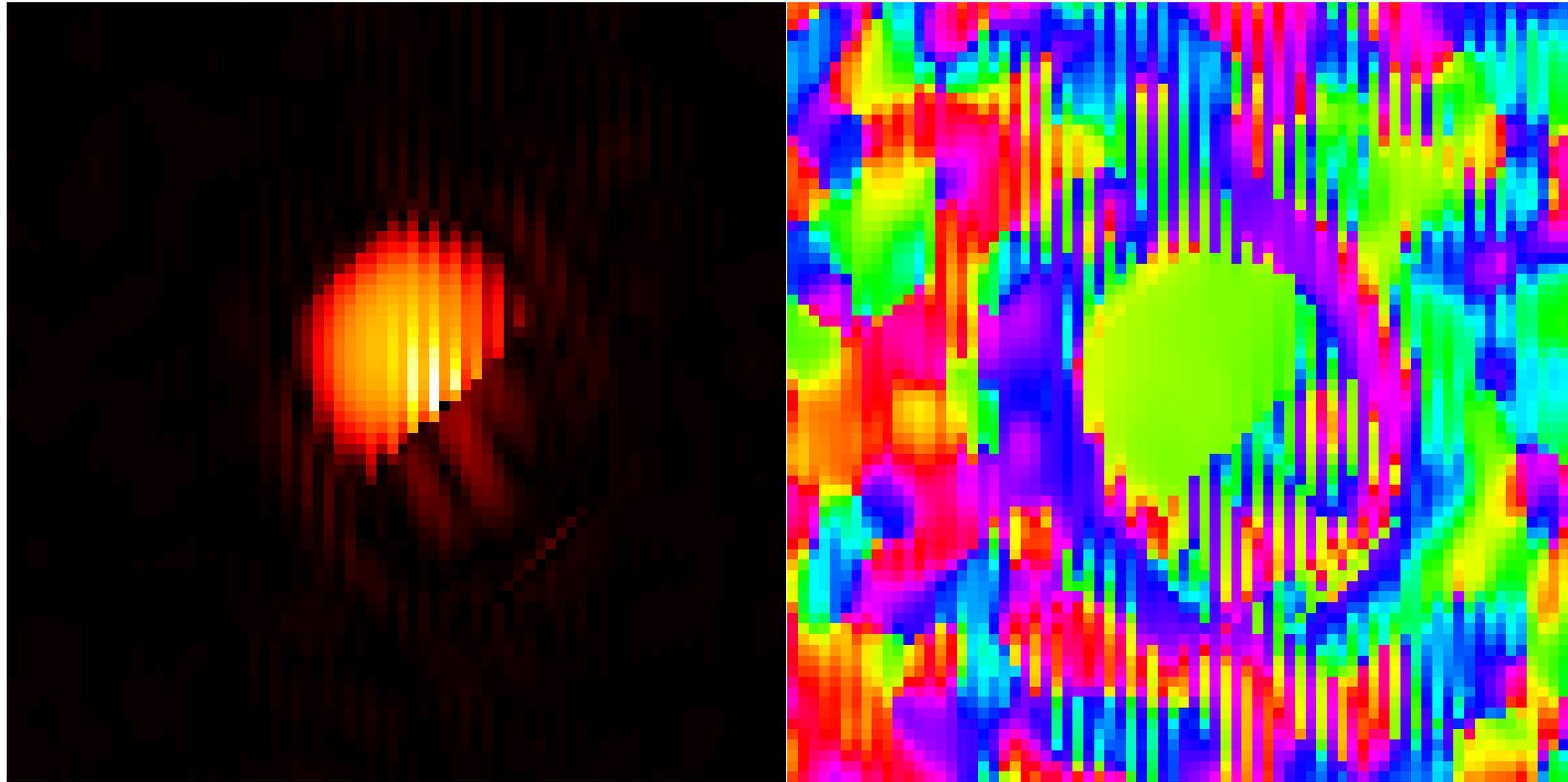
Alternation of ER and HIO

Helps to avoid stagnation



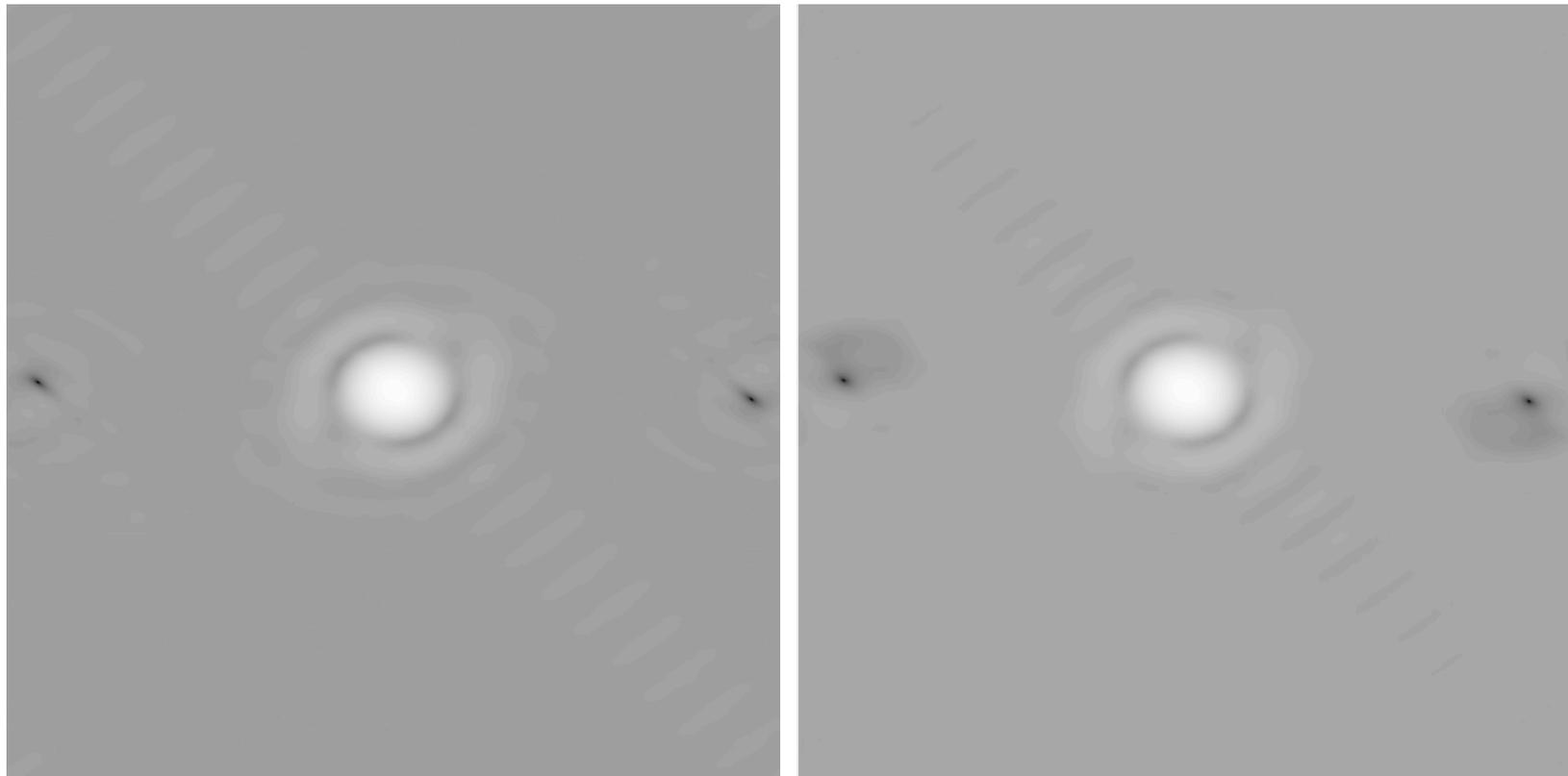
Incomplete Reconstruction can be Striped

0.5 micron Pb crystal on SiO₂ substrate

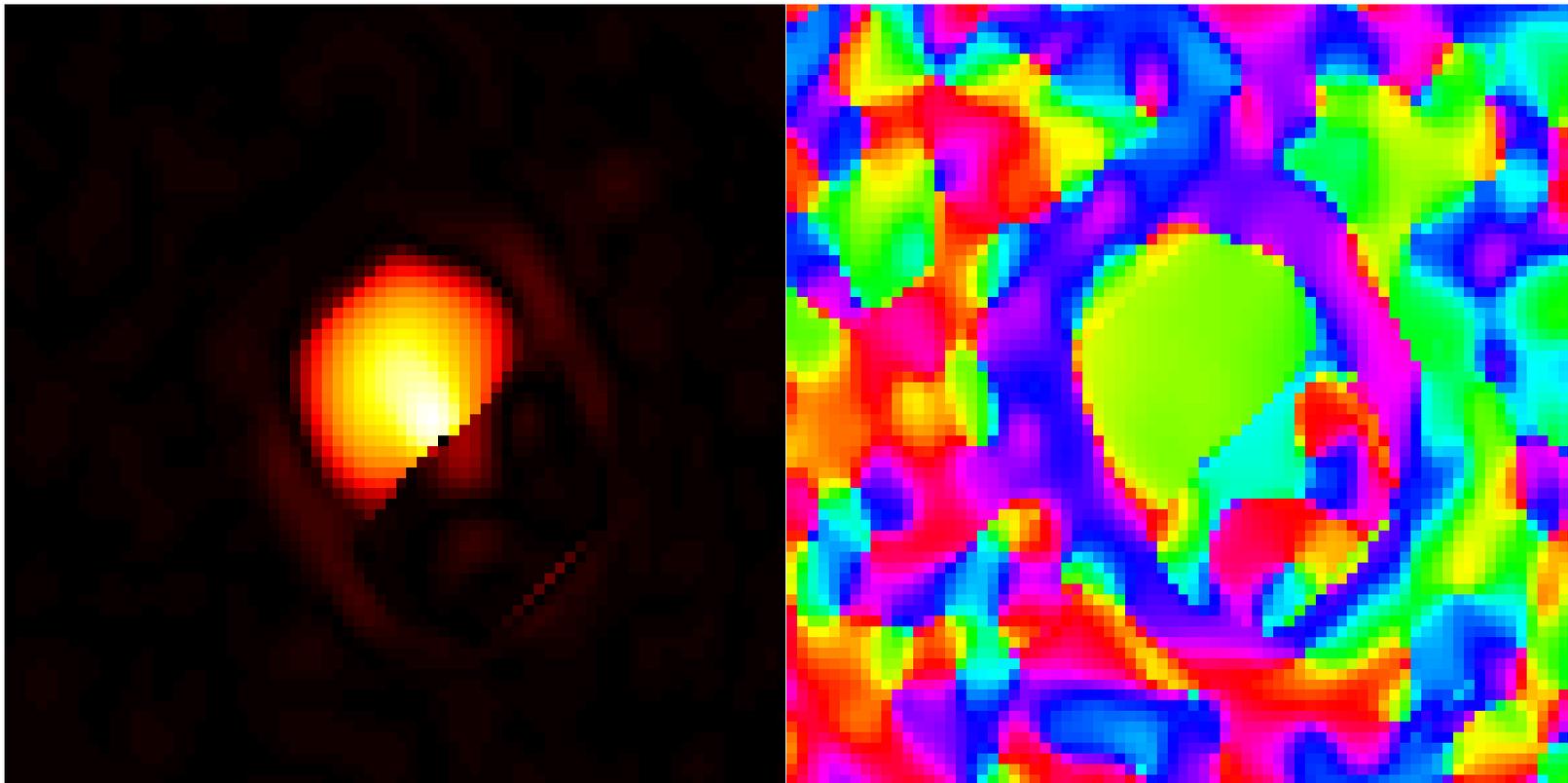


Stripes caused by “Vortices”

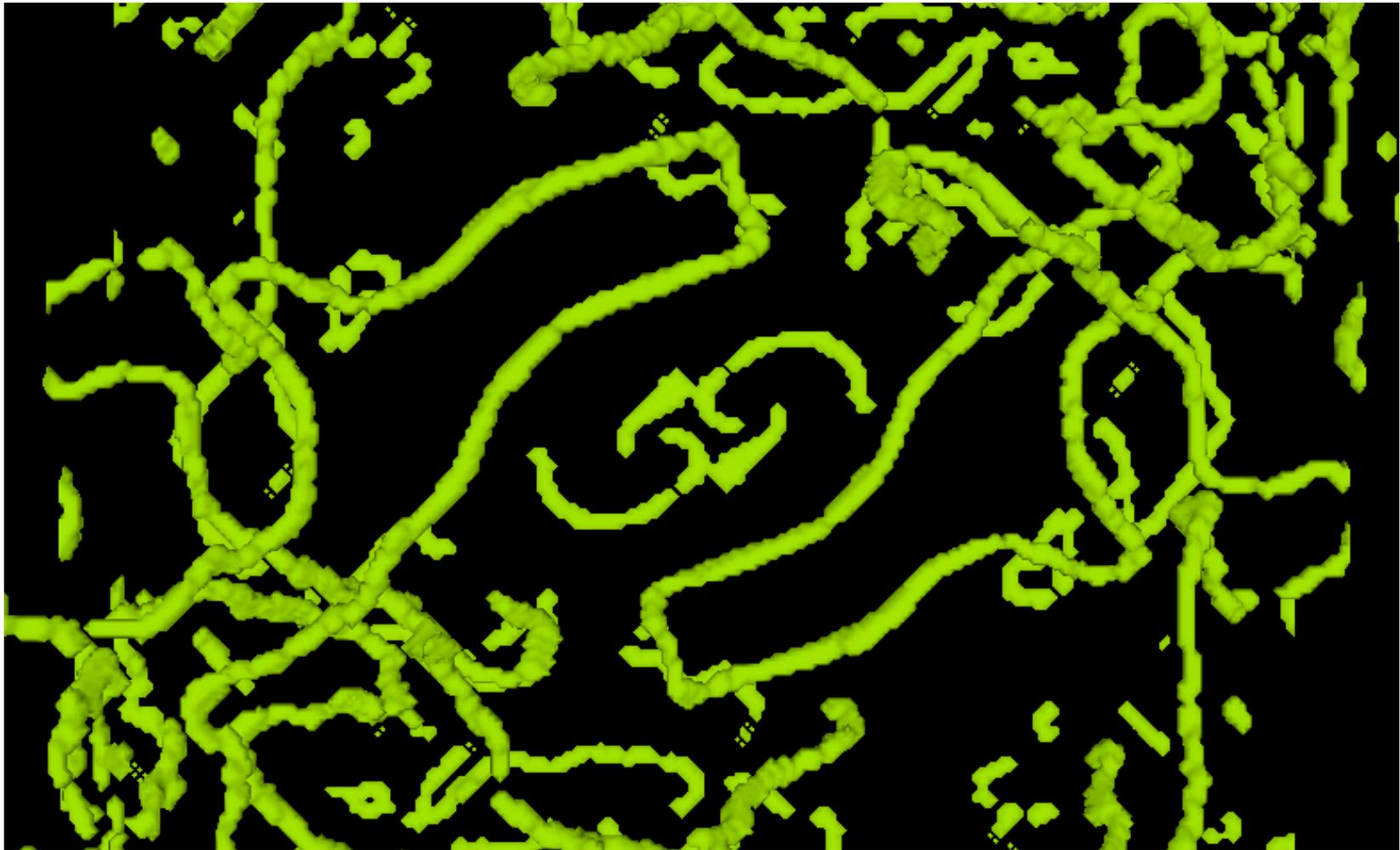
Vortex pairs separated by inverse of stripe spacing



Result of “Patching” in 2D

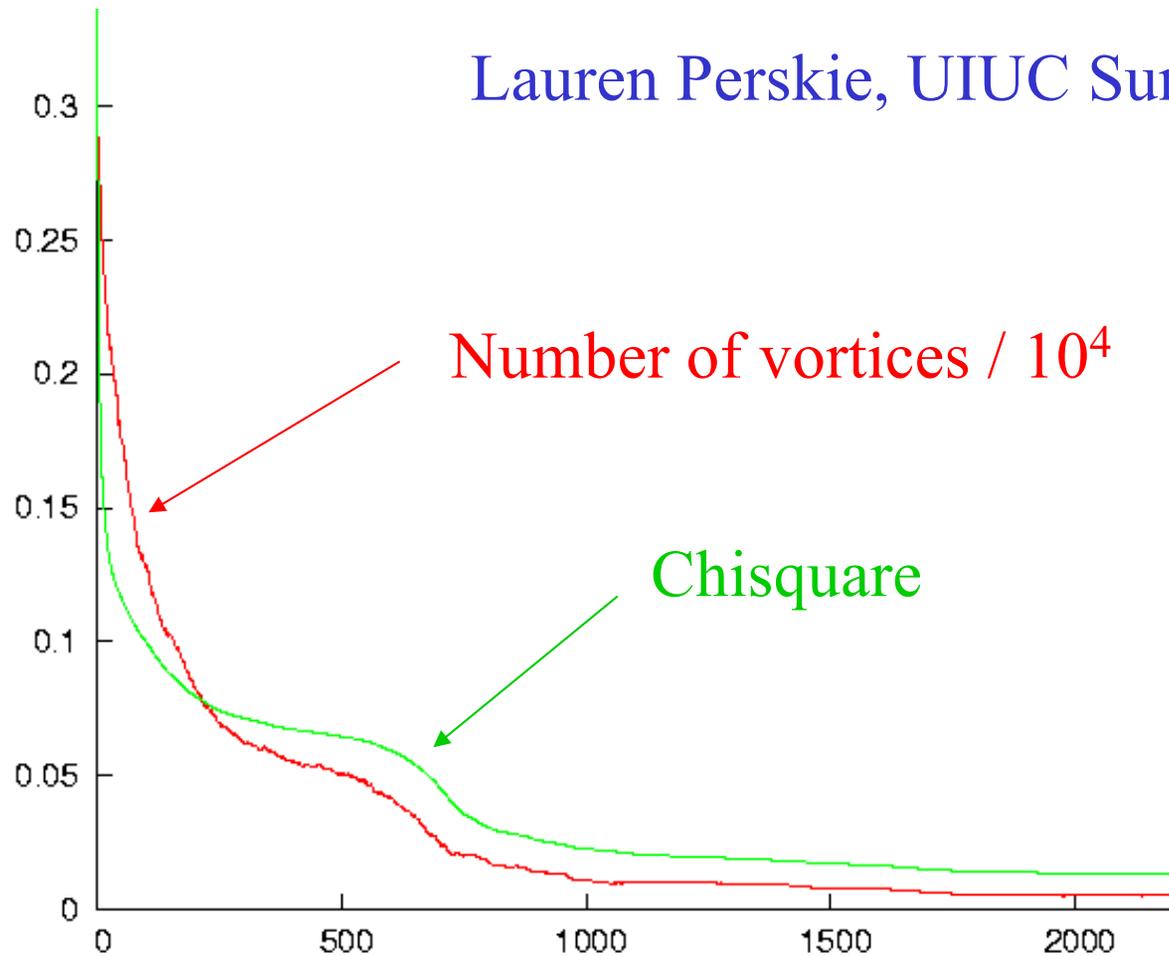


3D Vortices Form Pairs of Loops

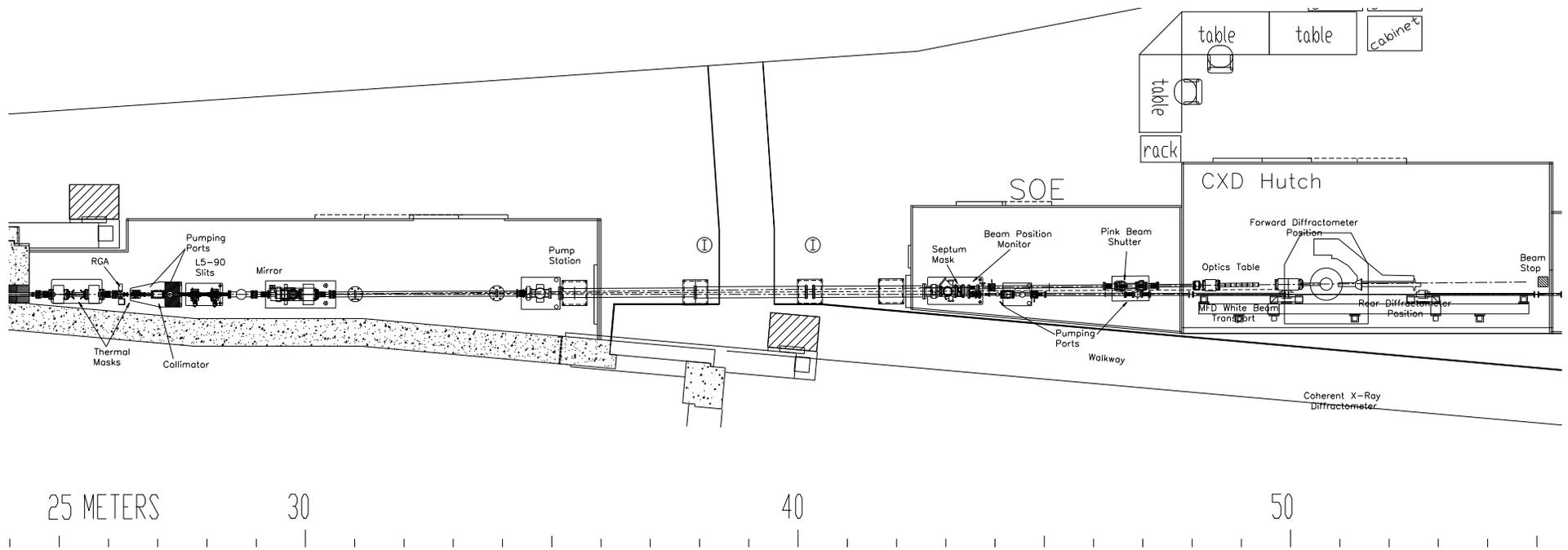


Vortices are a Cause of Stagnation during Error Reduction

Lauren Perskie, UIUC Summer student

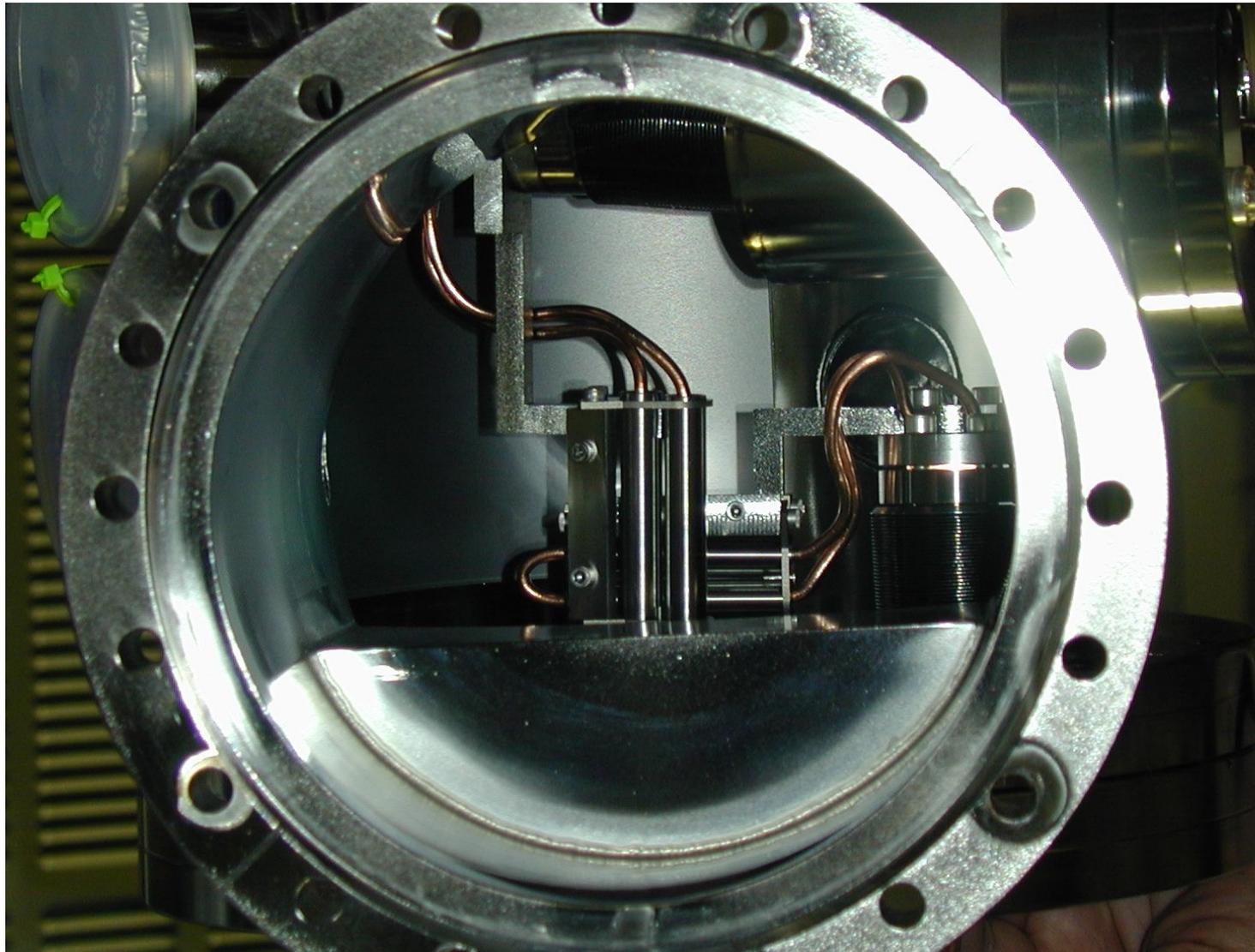


CXD Beamline at APS Sector 34

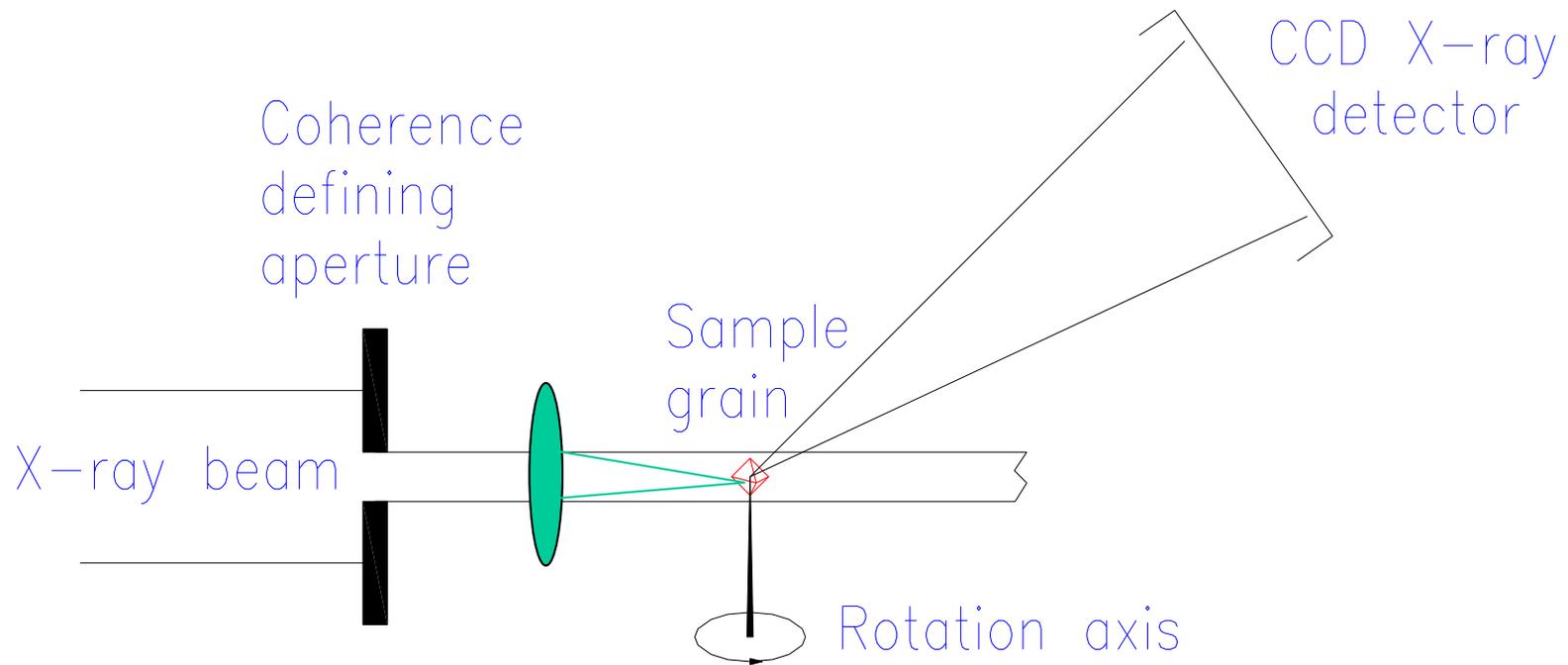




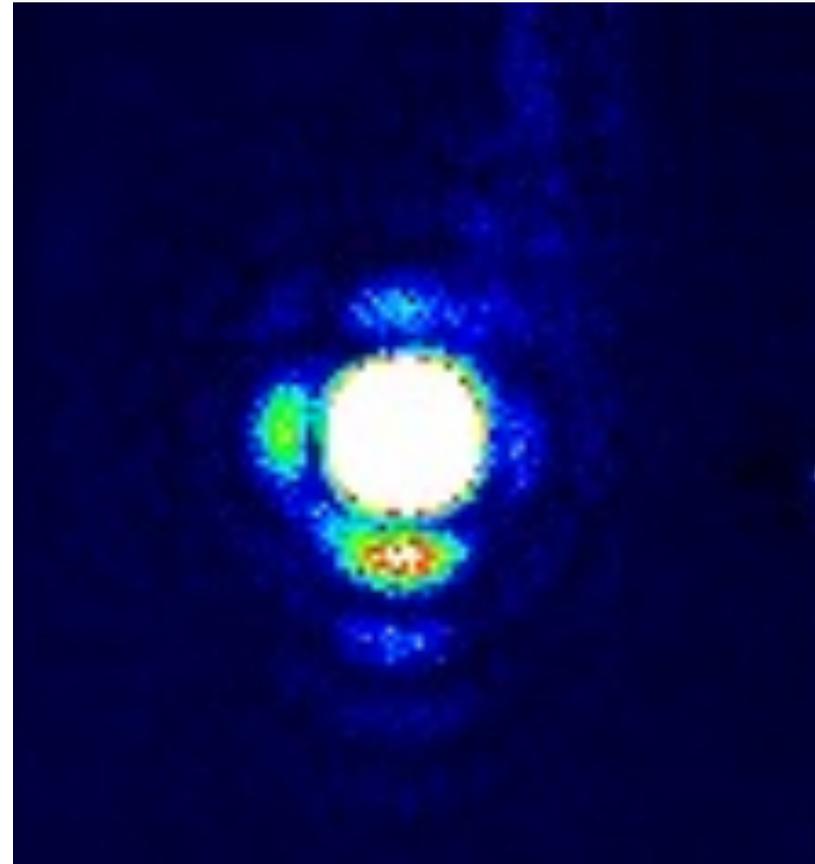
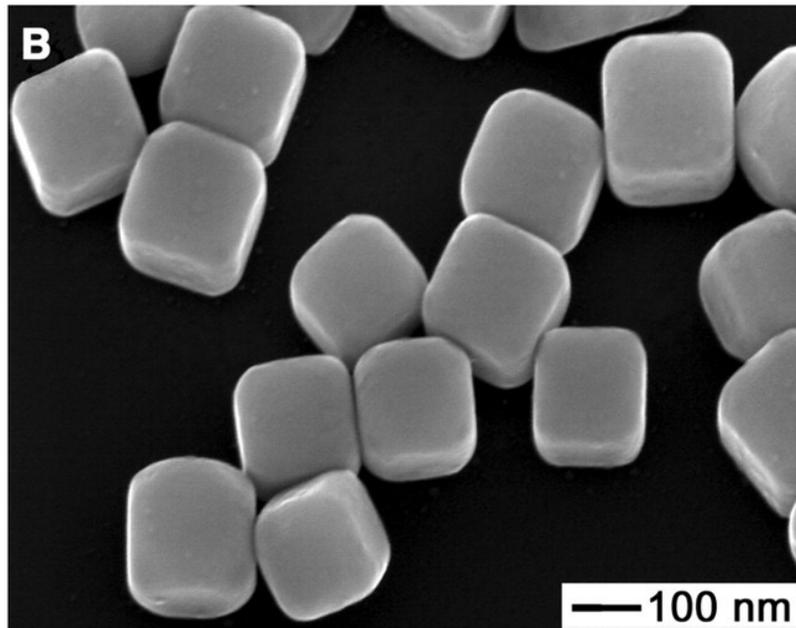
Roller-Blade Slits in UHV



Lensless X-ray Microscope

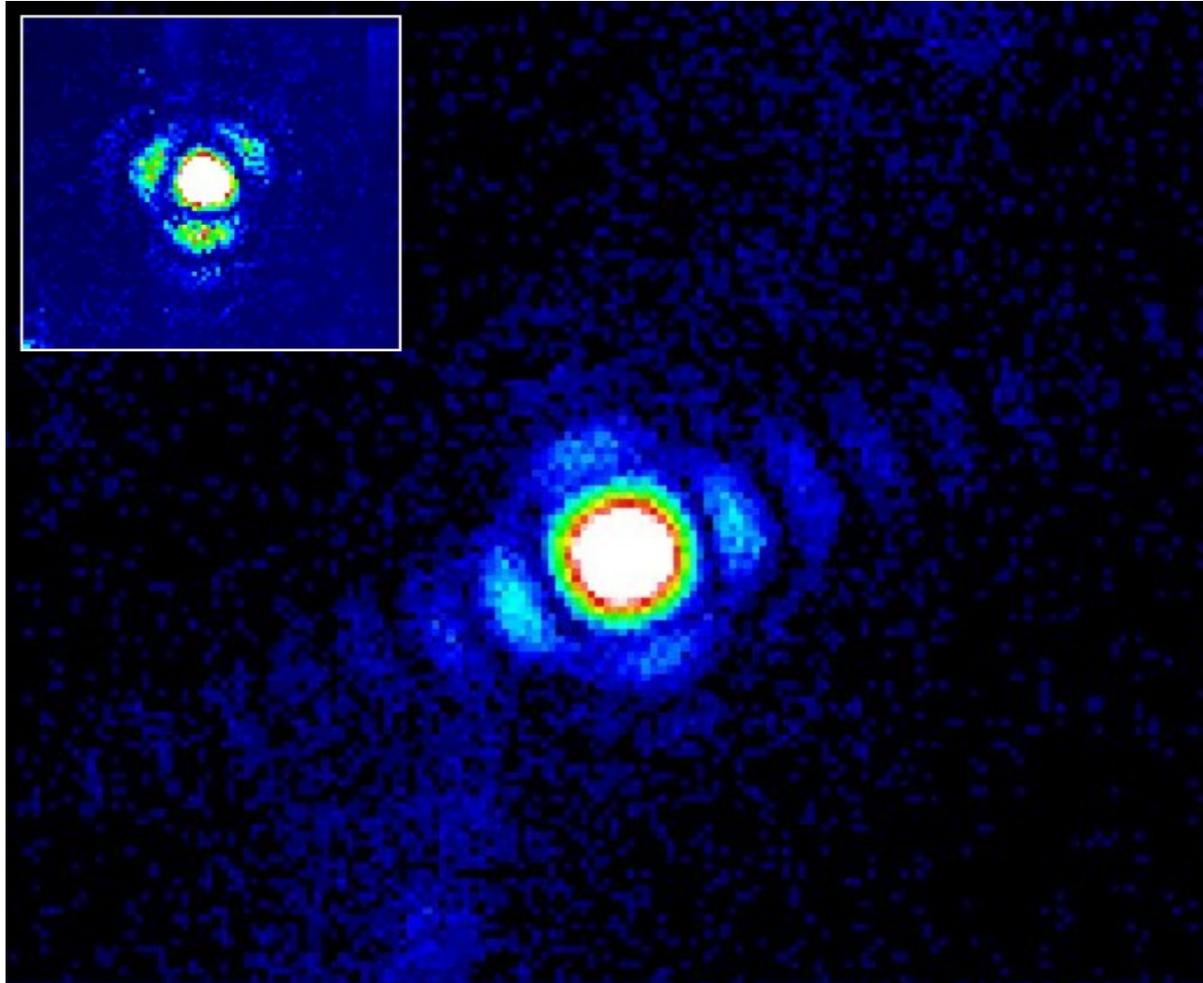


CXD from Silver Nanocubes



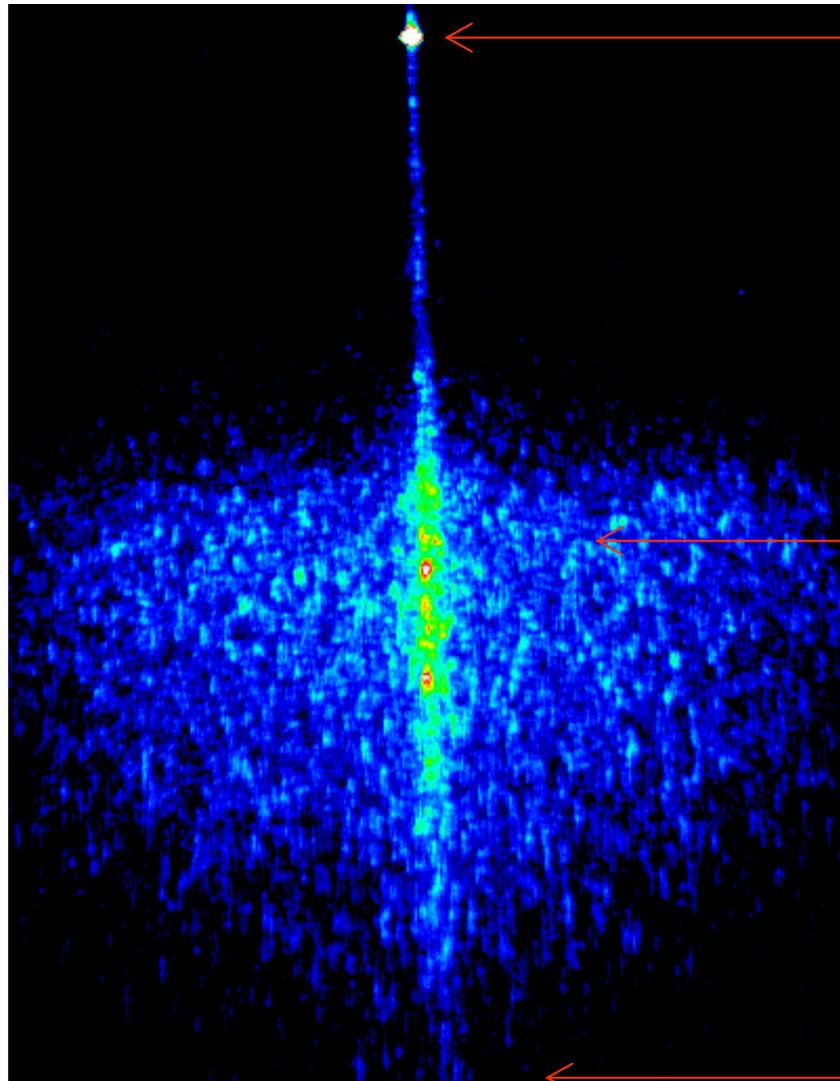
Yugang Sun and Younan Xia,
Science 298 2177 (2003)

170nm Silver Nanocubes



Structure in “Yoneda” Peak

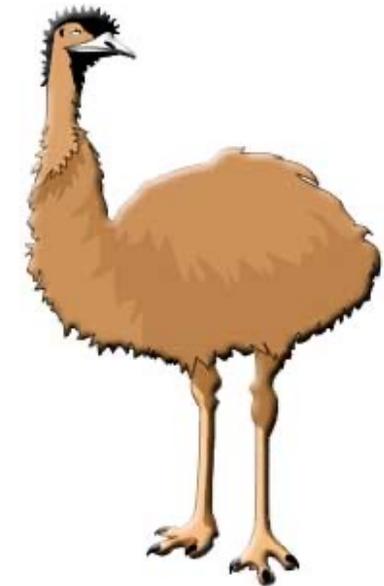
Grazing-exit diffraction from a 1000Å Au polycrystalline film



Specular ($\alpha_f \sim \alpha_i$)

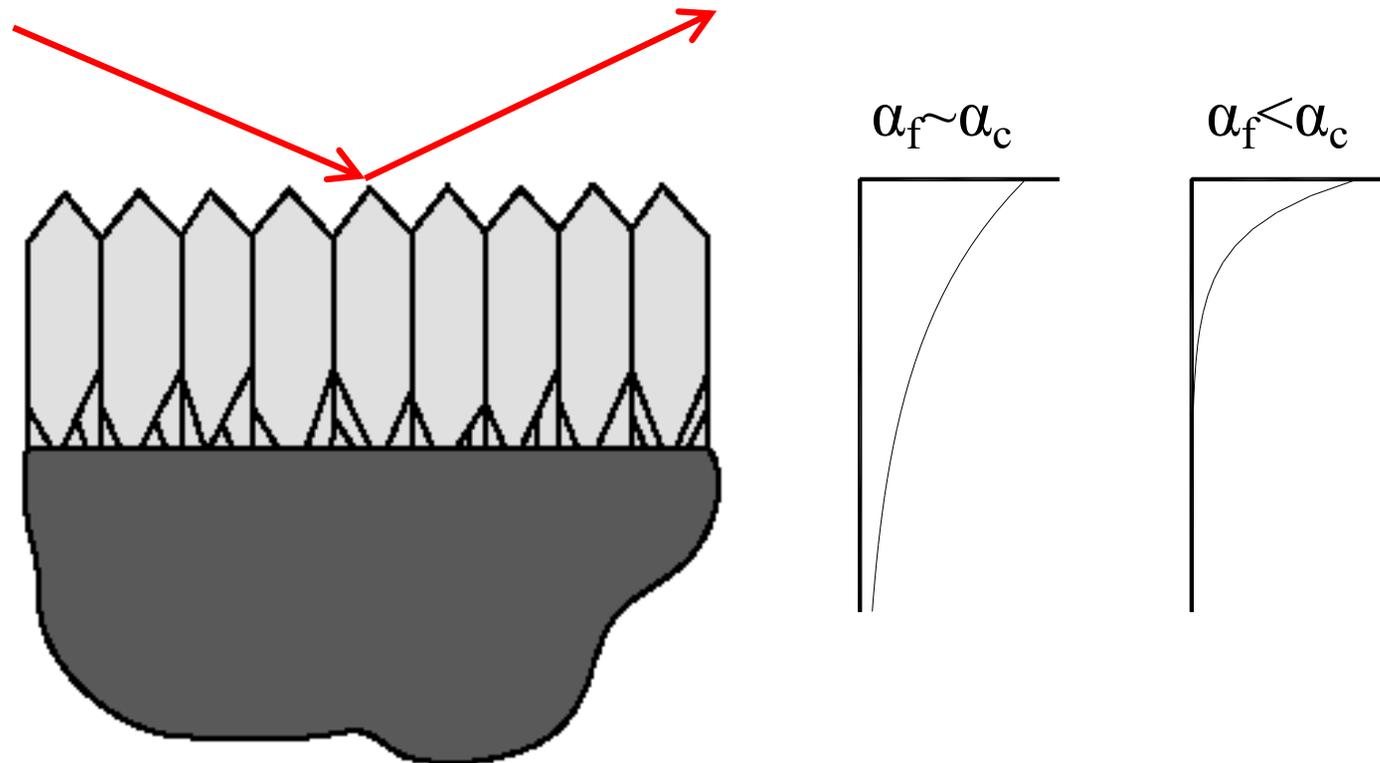
$\alpha_f \sim \alpha_c$

$\alpha_f \sim 0$

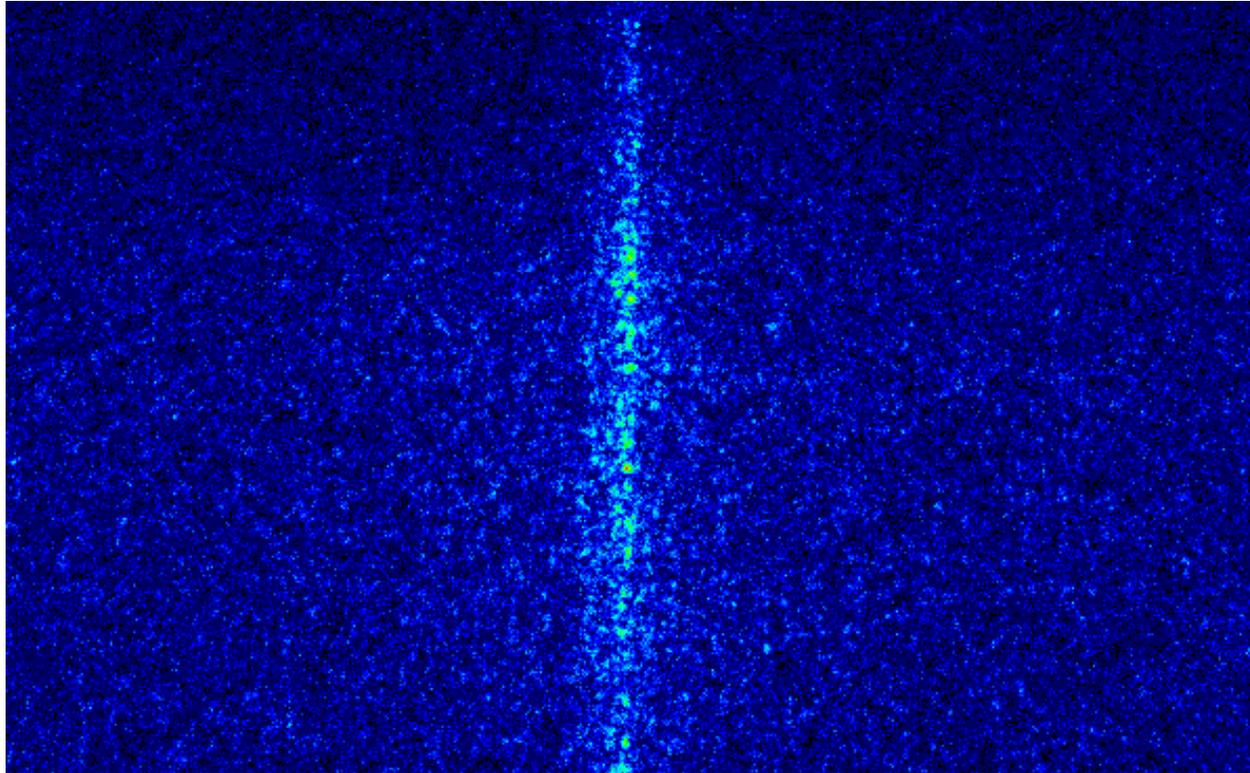


Competitive Grain Growth

C. V. Thompson, *Ann. Rev. Mat. Sci.* **30** 159 (2000)



Angle series, 0.01° steps

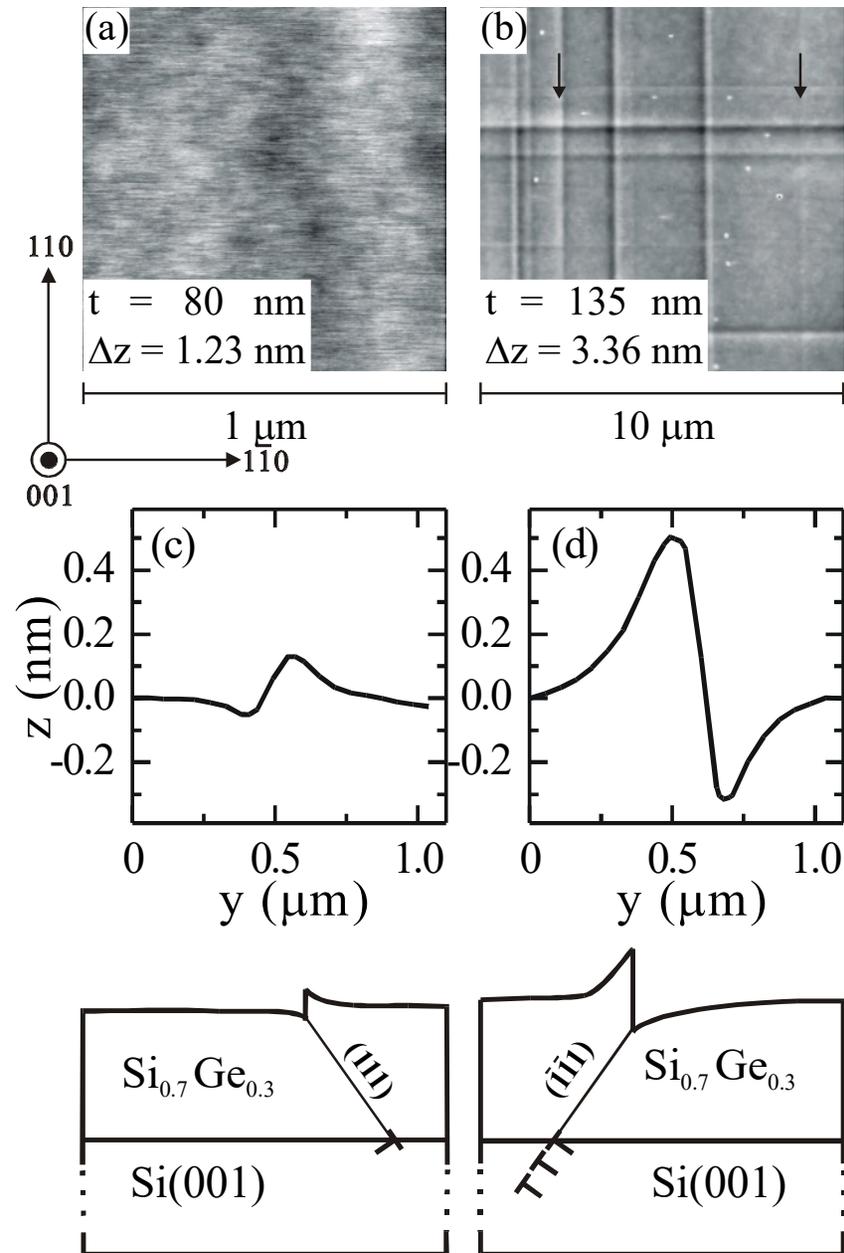


Low dislocation density GeSi films

Thickness close to critical thickness

Dislocations aggregate at interface and glide to surface along $\{111\}$

T. Spila, UIUC Thesis

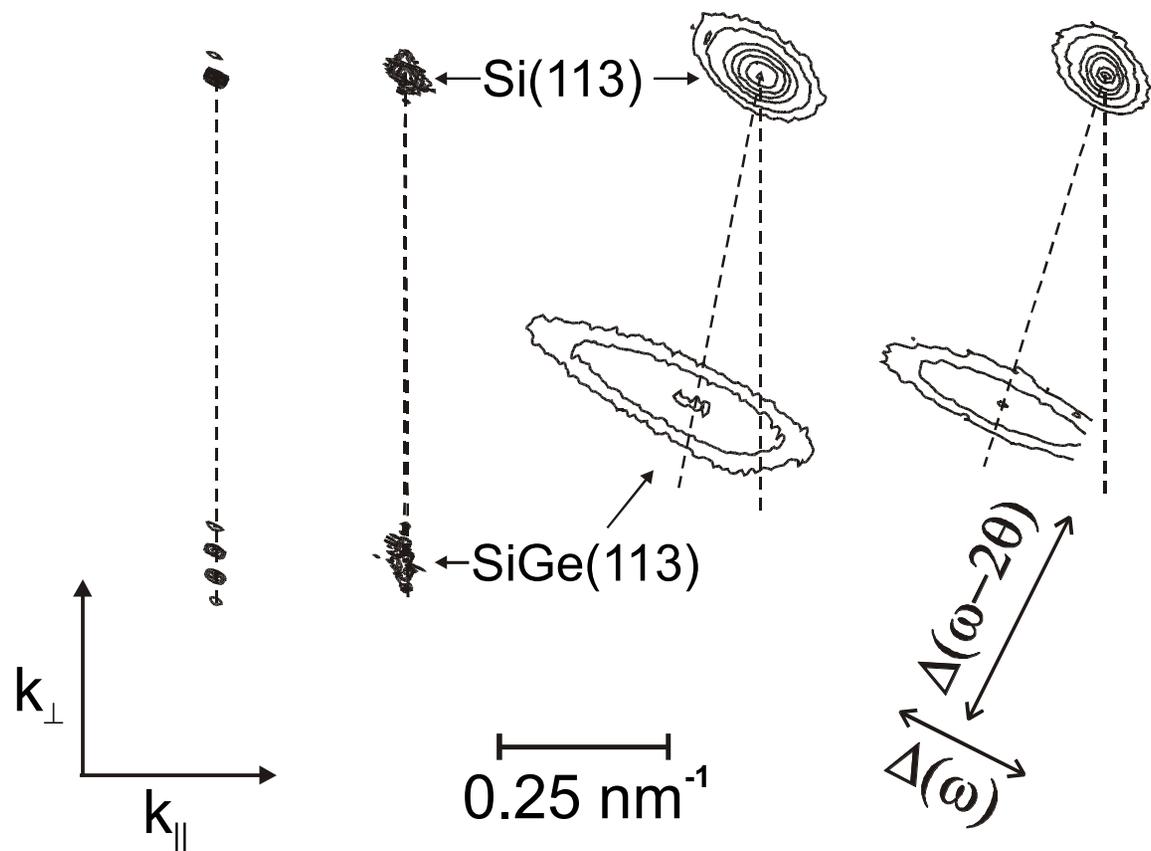


$\text{Si}_{0.7}\text{Ge}_{0.3}/\text{Si}(001)$

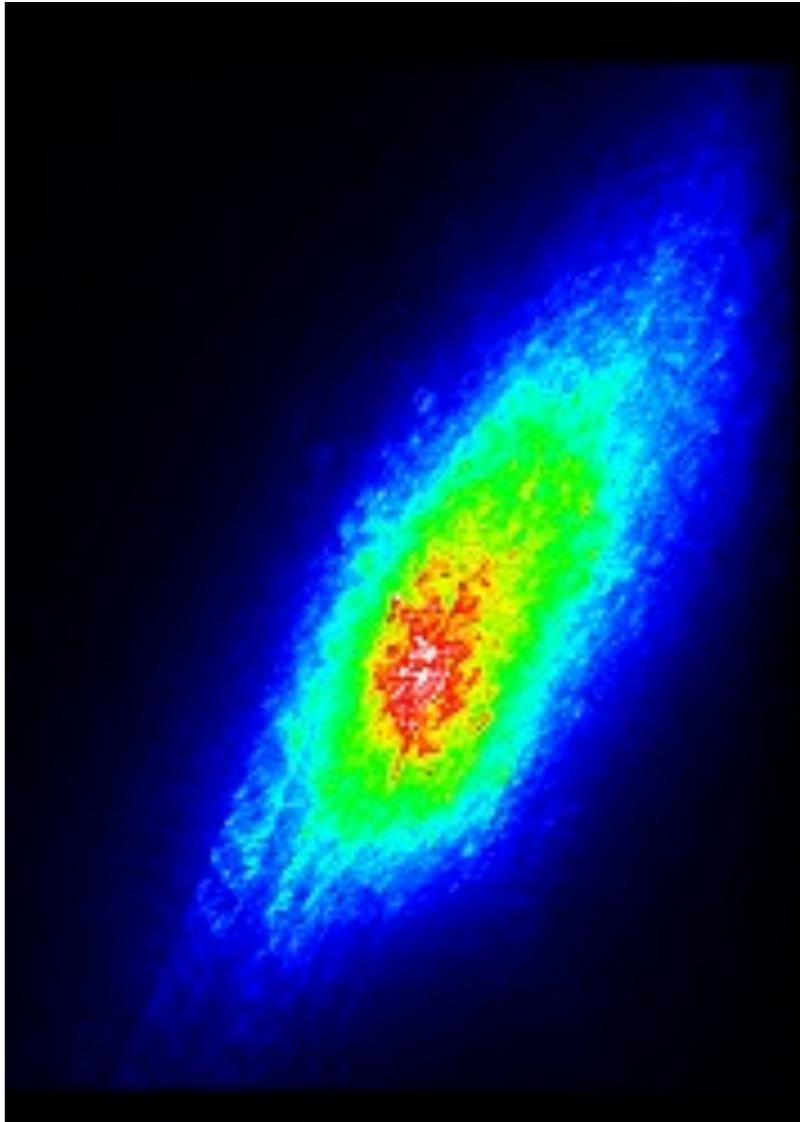
$T_s = 450\text{ }^\circ\text{C}$

$t =$ 80 135 350 440 nm

$R =$ 0 1 51 78 %



$\text{Ge}_x\text{Si}_{1-x}$ Film Diffraction



- 202 Bragg Peak
- 2800Å film
- 2° incidence angle
- 8.5 keV
- 20μm × 40μm beam
onto KB mirror
- 1μm × 1μm focus
- 0.5μm sample steps
- APS 34-ID-C

Conclusions and Outlook

- Inversion of CXD by ER-HIO methods
- Internal structure of Au Nanocrystals
- Preservation of coherence upon focussing
- Smallest size now down to 170nm
- New CXD-Yoneda geometry
- Dislocation strain structure may be possible