

# Thermodynamic Analysis of the System MgO-FeO-SiO<sub>2</sub> at High Pressure and the Structure of the Lowermost Mantle

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Semi-empirical thermodynamic potentials, based on Debye theory and Birch-Murnaghan finite strain theory are constructed for important mantle phases in the MgO-FeO-SiO<sub>2</sub> system. The parameters are constrained by inverting experimental compression, thermal expansion, thermochemical and phase equilibria data on the relevant phases. Calculated pseudo-unary MgSiO<sub>3</sub> and Mg<sub>2</sub>SiO<sub>4</sub>, and pseudo-binary Mg<sub>2</sub>SiO<sub>4</sub>-Fe<sub>2</sub>SiO<sub>4</sub> phase diagrams agree well with available data. Analysis of the complete ternary system shows that (Mg,Fe)SiO<sub>3</sub> perovskite is stable throughout the likely pressure-temperature and compositional regime of the Earth's mantle. The breakdown of perovskite to its constituent oxides is unlikely in the Earth, even at the extreme pressure-temperature conditions of the core-mantle boundary. This reaction had been proposed to reconcile proposed geotherms with predicted silicate melting curves and seismic observations of the deep mantle.

## INTRODUCTION

The constitution of the D" layer at the base of the mantle is of broad geophysical interest because the layer is expected to strongly modulate and possibly control the geophysical fields which originate in the Earth's core. The surface manifestations of core heat flow and the geodynamo provide essential information for our understanding of the deep interior and how it affects tectonic, igneous and possibly climatic processes. A significant fraction of core heat may be transported to the surface by mantle plumes that likely originate in D" and are responsible for flood basalts, oceanic plateaus, hot spots and their associated climatic effects [Duncan and Richards, 1990; Tarduno et al., 1991]. Possible metallic constituents in D" caused by chemical reaction between mantle and core may distort surface observations of the geomagnetic field, with important implications for models of fluid motion in the core and the geodynamo [Jeanloz, 1990]. Correlations between flood basalt activity and geomagnetic field reversal frequency [Courtilot and Besse, 1987] as well as arguments based on the secular variation of the magnetic field [Bloxham and Jackson, 1991] suggest that lateral and temporal variations in temperature in

D", caused by large scale mantle flow, the generation of plumes or the impingement of subducted oceanic crust, may influence the nature of flow in the core and the geodynamo.

The D" layer is a few hundred kilometer thick region of anomalous seismic properties. It is distinguished from the rest of the lower mantle by strong lateral heterogeneity, anomalous, often negative velocity gradients and a velocity discontinuity over much of its upper boundary [Young and Lay, 1987; Weber and Körnig, 1990]. The mineralogical or compositional changes responsible for these unusual seismic properties remain unknown.

(Mg,Fe)SiO<sub>3</sub> perovskite is generally accepted as the most abundant mineral in the lower mantle [Jeanloz and Knittle, 1989; Bukowski and Wolf, 1990]. Perovskite dominated assemblages readily account for the seismic properties of the bulk of the lower mantle and phase equilibrium studies over most of the mantle's pressure-temperature regime confirm its stability [Knittle and Jeanloz, 1987]. However, the stability of perovskite under D" pressure-temperature conditions has not yet been experimentally addressed.

A comparison of recently proposed geotherms [Williams and Jeanloz, 1990; Knittle and Jeanloz, 1991b] with predicted silicate melting curves in the deep mantle suggests that perovskite cannot be a constituent of D" [Stixrude and Bukowski, 1990]. Near the core-mantle boundary, its melting point falls more than a thousand degrees below estimated temperatures in the Earth. The proposed core-mantle boundary temperature is primarily determined by the measured melting temperature of iron under shock-loading [Williams et al., 1987] which is found to be significantly higher than model calculations of the iron hughoniot had

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predicted [Brown and McQueen, 1982]. If the shock wave measurements and the predicted perovskite melting curve are approximately correct, the constitution of D" must differ substantially from the rest of the mantle, since seismic observations rule out significant amounts of partial melt. If the region's bulk chemistry is dominated, like the rest of the mantle, by Mg, Fe and Si oxide components, perovskite must undergo a transformation to a high pressure phase or assemblage under D" conditions. A proposed breakdown of perovskite to its constituent oxides permits a solid D", since oxides are expected to be more refractory than their compounds, and provides a possible explanation of the region's distinctive seismic properties [Stixrude and Bukowinski, 1990].

In this report, we present a thermodynamic analysis of the stability of perovskite at high pressures and temperatures and show that the breakdown of perovskite to its constituent oxides is unlikely in the Earth. The reaction does occur at appropriate pressure-temperature conditions but only for iron contents so high that the products disagree severely with the density and bulk modulus of the lowermost mantle. We discuss other possible resolutions of the apparent discrepancy between the perovskite melting curve, the geotherm and seismic observations.

#### THERMODYNAMIC POTENTIALS

The Gibbs free energy per formula unit,  $G$ , of a phase consisting of a solid solution of  $N$  species as a function of pressure,  $P$ , and temperature,  $T$ , is:

$$G_i(P, T) = \sum_{j=1}^N x_{ij} [g_{ij}(P, T) + RT \ln a_{ij}] \quad (1)$$

where  $x_{ij}$ ,  $g_{ij}$  and  $a_{ij}$  are the mole fraction, Gibbs free energy and activity of species  $j$  in phase  $i$ . We assume symmetric regular solution behavior [e.g. Guggenheim, 1952]:

$$RT \ln a_{ij} = s_i [RT \ln x_{ij} + W_i (1 - x_{ij})^2] \quad (2)$$

where  $W_i$  is the interaction parameter and  $s_i$  is the stoichiometric coefficient of the mixing site. The Gibbs free energy is written as:  $g_{ij}(P, T) = F_{ij}(V_{ij}, T) + PV_{ij}$ , where  $V$  is the molar volume. We assume that the Helmholtz free energy,  $F_{ij}$ , can be divided into a reference term, a purely volume dependent part given by Birch-Murnaghan finite strain theory, a thermal part, given by Debye theory and an electronic contribution, given by free electron theory (subscripts  $i, j$  understood) [Stixrude and Bukowinski, 1990]:

$$F(V, T) = F_o + F_c(V, T_o) + [F_{th}(V, T) - F_{th}(V, T_o)] + [F_{el}(V, T) - F_{el}(V, T_o)] \quad (3)$$

where the subscript  $o$  refers to zero pressure and temperature  $T_o$ . The cold part is given by:

$$F_c(V) = 9K_o V_o \left[ \frac{1}{2} f^2 + \frac{1}{3} a_1 f^3 + \frac{1}{4} a_2 f^4 \dots \right] \quad (4a)$$

$$f = \frac{1}{2} [(V/V_o)^{-2/3} - 1] \quad (4b)$$

$$a_1 = \frac{3}{2} (K_o' - 4) \quad (4c)$$

$$a_2 = \frac{3}{2} [K_o K_o'' + K_o' (K_o' - 7) + 143/9] \quad (4d)$$

where  $f$  is the finite strain and  $K_o$  and  $K_o'$  are the isothermal bulk modulus and its first pressure derivative. Terms of 4<sup>th</sup> and higher order are unimportant for the phases and pressure ranges considered here and are neglected. In general, the thermal contribution contains both quasiharmonic terms, for which we assume the Debye model, and anharmonic terms [Wallace, 1972]:

$$F_{th}(V, T) = 9nRT(T/\theta)^3 \int_0^{\theta/T} \ln(1 - e^{-t}) t^2 dt + A_2 T^2 \quad (5)$$

where  $n$  is the number of atoms in the formula unit and  $\theta$  is the Debye temperature, whose negative logarithmic volume derivative is  $\gamma$ , the Grüneisen parameter. Anharmonic terms are expected to be small [Hardy, 1980; Stixrude and Bukowinski, 1990] at the high pressures of interest here and are neglected ( $A_2=0$ ). The electronic term appears only for the high pressure form of wüstite (Wü II), which is known to be metallic:

$$F_{el} = -\frac{1}{2} \beta_o (V/V_o)^{2/3} T^2 \quad (6)$$

where  $\beta_o = (m^*/m) \beta_{FE}$ ,  $m^*/m$  is the free electron mass ratio, which we assume is the same as for Fe and  $\beta_{FE}$  is the free electron heat capacity coefficient, which depends only on zero pressure conduction electron density [e.g. Ashcroft and Mermin, 1976]. The structure of Wü II and its solution properties remain unknown. We assume that Wü II forms a complete solid solution with Pe II and that this phase (Mw II) contains an electronic contribution to its free energy even at very small FeO concentrations. These assumptions lead to a lower bound on the free energy of Mw II and thus a lower bound on the width of the perovskite stability field since the structure of Wü II may differ from that of Pe II, and, if a solid solution exists, it is likely insulating or semi-conducting at small FeO contents.

The parameters are either directly measured or constrained by experimental data or first principles calculations (Table 1). Following the approach of Bass *et al.* [1981], we adopt the directly measured value of  $K_o$ , where ultrasonic or brillouin determinations exist, and constrain  $K_o'$  by inverting compression data. The reference state Debye temperature,  $\theta_o$ , and Grüneisen parameter,  $\gamma_o$ , are determined by inverting thermal expansion and calorimetric data as described by Stixrude and Bukowinski [1990] (see Figures 1-3) [see also Ita and Stixrude, 1992; Hemley *et al.*, 1992]. Reference free energies and interaction parameters are

TABLE 1. Thermodynamic potential parameters.

Phase	Formula	$V_o$ (cm <sup>3</sup> /mol)	$K_o$ (GPa)		$K_o'$		$\theta_{\sigma}$ %		
			Value	Ref.	Value	Ref.	$\theta_o$	$\gamma_o$	Ref.
Olivine (Ol)	Mg <sub>2</sub> SiO <sub>4</sub>	43.76	128(2)	1,2	5.0(5)	17	924(10)	1.14(10)	25,26
	Fe <sub>2</sub> SiO <sub>4</sub>	46.27	127(2)	3	5.2(5)	18	688(10)	1.08(10)	27,28
Wadsleyite (Wa)	Mg <sub>2</sub> SiO <sub>4</sub>	40.52	174(2)	4	4.0(5)	19	974(10)	1.32(10)	25,29
	Fe <sub>2</sub> SiO <sub>4</sub>	43.22	174(2) <sup>a</sup>	5	4.0(5) <sup>a</sup>	5	771(50) <sup>c</sup>	1.32(20) <sup>a</sup>	
Spinel (Sp)	Mg <sub>2</sub> SiO <sub>4</sub>	39.65	183(3)	6	4.1(5)	20	1017(10)	1.21(10)	25,30
	Fe <sub>2</sub> SiO <sub>4</sub>	42.02	192(3)	7	4.1(5)	21	781(10)	1.21(10)	27,30
Enstatite (En)	MgSiO <sub>3</sub>	31.33	106(2)	8	5.0(5)	22,23	935(10)	0.97(10)	31,32
Majorite (Mj)	Mg <sub>4</sub> Si <sub>4</sub> O <sub>12</sub>	114.15	151(10)	9	4(1) <sup>b</sup>		949(50) <sup>d</sup>	1.24(20) <sup>b</sup>	
Ilmenite (Il)	MgSiO <sub>3</sub>	26.35	212(4)	10	4.3(5)	22	1026(10)	1.48(10)	33
Perovskite (Pv)	MgSiO <sub>3</sub>	24.46	263(7)	11,12	3.9(5)	11,12	1017(10)	1.96(10)	34,35
	FeSiO <sub>3</sub>	25.49	263(7) <sup>a</sup>	12	3.9(5) <sup>a</sup>	12	749(50) <sup>d</sup>	1.96(10) <sup>a</sup>	
Magnesio-wüstite (Mw)	MgO	11.25	160(2)	13	4.1(2)	13	776(10)	1.45(10)	26,36
	FeO	12.25	152(2)	14	4.9(2)	14	434(50) <sup>d</sup>	1.45(10)	37
Magnesio-wüstite II (Mw II)	MgO	10.79	161(10)	15	4.1(1.0)	15	698(50)	1.36(20)	15
	FeO	10.84	205(10)	14	6(1)	14	630(50)	1.45(20) <sup>e</sup>	38
Stishovite (St)	SiO <sub>2</sub>	14.01	314(8)	16	8.8(8)	24	1152(10)	1.35(10)	27,39
Liquid	Mg <sub>2</sub> SiO <sub>4</sub>	50.12	37.7(2.0)	41	6.3(5)	23	595(20)	0.99(20)	40,41
Liquid	MgSiO <sub>3</sub>	38.73	21.3(2.0)	41	6.4(5)	23	625(20)	0.36(20)	40,41

Estimated standard deviations are shown in parentheses.  $q=2.5(1.7)$  for perovskite [Hemley et al., 1992] and is assumed to be 1(2) for all other phases. References: 1, *Graham and Barsch* [1969]; 2, *Isaak et al.* [1989]; 3, *Graham et al.* [1988]; 4, *Sawamoto et al.* [1984]; 5, *Hazen et al.* [1990]; 6, *Weidner et al.* [1984]; 7, *Liebermann* [1975]; 8, *Weidner et al.* [1978]; 9, *Bass and Kanzaki* [1990]; 10, *Weidner and Ito* [1985]; 11, *Knittle and Jeanloz* [1987]; 12, *Mao et al.* [1991]; 13, *Jackson and Niesler* [1982]; 14, *Jackson et al.* [1990]; 15, *Bukowinski* [1985]; 16, *Weidner et al.* [1982]; 17, *Olinger* [1977]; 18, *Williams et al.* [1990]; 19, *Fei et al.* [1991b]; 20, *Sawamoto et al.* [1986]; 21, *Bass et al.* [1981]; 22, *Duffy and Anderson* [1989]; 23, *Stixrude and Bukowinski* [1990]; 24, *Tsuchida and Yagi* [1989]; 25, *Ashida et al.* [1987]; 26, *Suzuki* [1975a]; 27, *Watanabe* [1982]; 28, *Suzuki et al.* [1981]; 29, *Suzuki et al.* [1980]; 30, *Suzuki* [1979]; 31, *Krupka et al.* [1985]; 32, *Suzuki* [1975b]; 33, *Ashida et al.* [1988]; 34, *Ito and Takahashi* [1989]; 35, *Knittle et al.* [1986]; 36, *Krupka et al.* [1979]; 37, *Touloukian et al.* [1977]; 38, *Knittle and Jeanloz* [1991]; 39, *Ito et al.* [1974]; 40, *Lange and Carmichael* [1987]; 41, *Stebbins et al.* [1984]. Volumes are from *Jeanloz and Thompson* [1983] except those of Majorite [Ito and Stixrude, 1992] and the liquid phases [Lange and Carmichael, 1987]. a. Assumed to be the same as for spinel; b. Assumed to be the same as pyrope; c. Dependence on Mg/Fe ratio assumed to be the same as for spinel; d. Ratio of majorite to pyrope thermal debye temperatures assumed to be the same as the ratio of the corresponding elastic debye temperatures.

determined by inverting phase equilibria measurements. We find the set of  $W_i$ 's and  $F_{oij}$ 's which provide the best simultaneous solution to a set of reactions (Table 2) of the form:

$$\sum_{i=1}^N v_{ij} \mu_{ij}(P_{\text{exp}}, T_{\text{exp}}, x_{ij, \text{exp}}) = 0 \quad (7)$$

where  $\mu_{ij}$ , the chemical potential of species  $j$  in phase  $i$ :

$$\mu_{ij} = F_{ij} + PV_{ij} + s_i [RT \ln x_{ij} + W_i (1 - x_{ij})^2] \quad (8)$$

is evaluated at the experimentally observed pressure, temperature and mole fraction with the parameters of Table 1. The coefficients,  $v_{ij}$ , balance the reaction. Unary reactions are represented by single  $P, T$  points, while each binary reaction includes several observations of the coexisting phases. Because the set of equations (7) depends linearly on the parameters ( $F_{oij}, W_i$ ) the best least-squares solution is readily found, e.g. by singular value decomposition (Table 3).

The thermodynamic potential (1) is expressed in terms of its natural variables and thus contains all thermodynamic information about the model system; it is a fundamental thermodynamic relation in the terminology of *Callen*

[1985]. All thermodynamic properties of the model system are given by pressure and temperature derivatives of (1). The potential formulation has important advantages over alternative schemes which determine  $G$  by integrating empirical expressions for the volume and entropy in terms of their conjugate variables (polynomials in pressure and temperature or more complex forms) [see e.g. *Wood and Fraser*, 1977]. Unless self-consistency is explicitly imposed over the temperature and pressure regime of interest, they lead to path dependent results for  $G$  and its derivatives. Thermodynamic identities, such as those between mixed partial derivatives:

$$-\left(\frac{\partial S}{\partial P}\right)_T = \left(\frac{\partial V}{\partial T}\right)_P \quad (9)$$

are everywhere satisfied by our approach, while they are generally not by empirical integration schemes.

In addition to complete self-consistency, the thermodynamic potentials used here generally require only a small number of parameters since they are physically based (finite strain theory and the Debye model of the vibrational density of states), which allows us to constrain each parameter independently by a large number of experimental

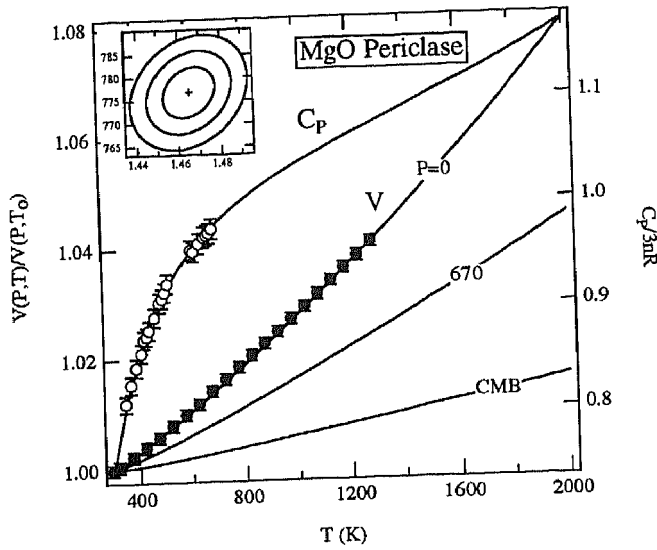


Fig. 1. Inversion of heat capacity and thermal expansion data for the Grüneisen parameter,  $\gamma_0$ , and Debye temperature,  $\theta_D$ , of Periclase. Lines are calculated using the best fitting parameters, indicated by the cross in the inset ( $\theta_D$  on the vertical axis and  $\gamma_0$  on the horizontal). The ellipses are 1-, 2- and 3- $\sigma$  confidence intervals. Thermal expansion at pressures corresponding to the upper mantle-lower mantle boundary (24 GPa), and the core-mantle boundary (136 GPa) is also shown. Sources of the data are given in Table 1.

measurements. The potentials describe a wide variety of thermodynamic properties of mantle minerals at high pressures and temperatures, including their equations of state, phase equilibria and melting behavior [Stixrude and Bukowinski, 1990]. In the form of their volume derivative - the Mie-Grüneisen equation of state - they have been extensively tested against shock wave and static compression

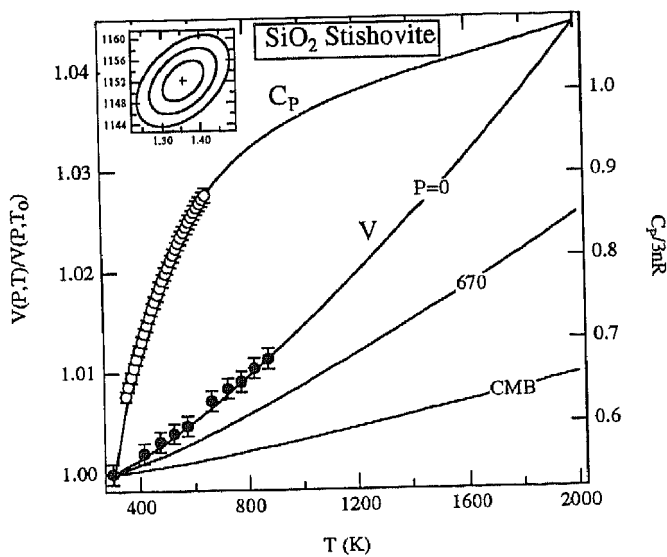


Fig. 2. Inversion of heat capacity and thermal expansion data for  $\gamma_0$  and  $\theta_D$  of stishovite. The format is the same as in Figure 1.

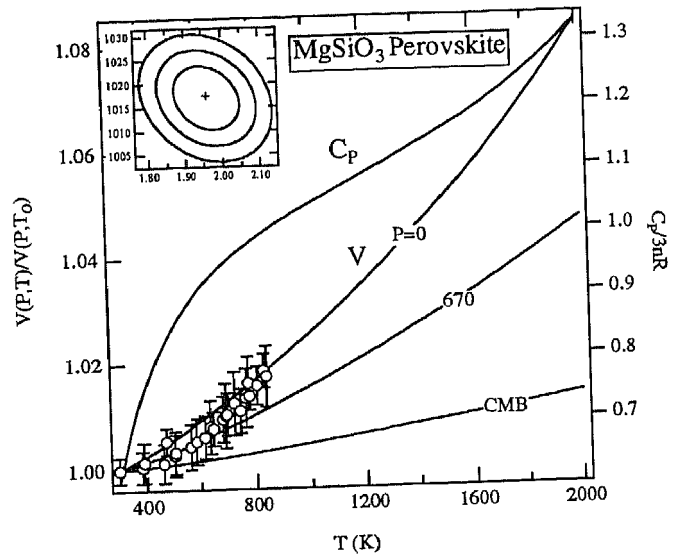


Fig. 3. Inversion of thermal expansion data and experimentally determined  $dT/dP$  slopes of the  $\text{Il}=\text{Pv}$  and  $\text{Sp}=\text{Pv}+\text{Mw}$  reactions (not shown) for  $\gamma_0$  and  $\theta_D$  of perovskite. The format is the same as in Figure 1.

data on a wide variety of solids including minerals to lower mantle pressures [Shapiro and Knopoff, 1969; Knopoff and Shapiro, 1969; McQueen et al., 1970; Jeanloz, 1989]. To the extent that anharmonic terms are negligible, the potentials remain sufficiently accurate for our purposes at the extreme

TABLE 2. Reactions used to constrain reference free energies and interaction parameters (Table 2).

Reaction	P (GPa)	T (K)	X <sub>Fe</sub>	Ref.
Ol=Wa	15.0	1873	0.0	1
Wa=Sp	20.5	1873	0.0	1
Sp=Pv+Mw	24.0	1600	0.0	2
2En=Wa+St	16.1	1573	0.0	3
En=Mj	16.8	1973	0.0	3
2Il=Sp+St	19.5	1273	0.0	3
Il=Pv	22.2	2123	0.0	3
Fo=Liquid	0.0	2163	0.0	4
En=Liquid	0.0	1830	0.0	4
Ol=Sp	6.0	1473	1.0	5
Ol=Sp	12.0	1473	Univ	1
Ol=Sp	13.0	1873	Univ	1
Wa=Sp	12.0	1473	Univ	1
Wa=Sp	13.0	1873	Univ	1
Sp=2Mw+St	22.5-24.5	1373	0.3-0.5	2
Sp=2Mw+St	22-23.1	1873	0.3-0.6	2
Pv=Mw+St	24.5-25.5	1373	0.3-0.4	2
Pv=Mw+St	23-25	1873	0.4-0.5	2

Columns show the pressure and temperature at which the reactions proceed for the indicated bulk iron content (X<sub>Fe</sub>). For binary reactions, observations over a range of pressure and bulk composition were used. Ol-Wa-Sp binary equilibria were constrained by the observed three phase univariant line. References: 1, Katsura and Ito [1989]; 2, Ito and Takahashi [1989]; 3, Gasparik [1990]; 4, Stebbins et al. [1984]; 5, Akimoto [1987].

TABLE 3. Thermodynamic potential parameters; reference free energies and interaction parameters. All quantities in kJ/mol.

Species	Global		High T		Low T	
	$F_o$	$W$	$F_o$	$W$	$F_o$	$W$
Forsterite	-127.7	6.1	-133.4	6.3	-124.0	6.5
Fayalite	-93.3	6.1	-96.6	6.3	-89.7	6.5
Mg-Wadsleyite	-99.6	3.9	-105.3	3.9	-95.9	3.9
Fe-Wadsleyite	-98.1	3.9	-100.9	3.9	-93.8	3.9
Mg-Spinel	-92.4	2.8	-98.0	3.6	-88.6	4.1
Fe-Spinel	-94.1	2.8	-97.6	3.6	-90.9	4.1
Mg-Enstatite	-91.0	0	-93.9	0	-89.2	0
Mg-Majorite	-205.4	0	-217.0	0	-198.1	0
Mg-Ilmenite	-35.9	0	-39.0	0	-33.7	0
Mg-Perovskite	3.4	0	-0.4	0	6.6	0
Fe-Perovskite	40.5	0	45.5	0	35.2	0
Periclase	0	13.1	0	13.5	0	12.6
Wüstite	0	13.1	0	13.5	0	12.6
Periclase II	139.7	13.1	139.7	13.1	139.7	13.1
Wüstite II	56.8	13.1	56.8	13.1	56.8	13.1
Stishovite	0	0	0	0	0	0
Mg <sub>2</sub> SiO <sub>4</sub> Liquid	-422.9	0	-428.6	0	-419.1	0
MgSiO <sub>3</sub> Liquid	-345.6	0	-348.5	0	-343.8	0

pressure-temperature conditions of interest here. The cold part remains well constrained by experimental measurements for most of the relevant phases, while the thermal part closely approaches its high temperature limiting behavior ( $T/\theta > 4$ ). In this limit, results are insensitive to the detailed form of the vibrational density of states which is approximately represented in the Debye model. The adequacy of the regular solution model and the assumed volume dependence of  $\gamma$  is more difficult to evaluate. We ensure that our results are insensitive to wide variations in  $W$  and  $q$ , the second logarithmic volume derivative of  $\theta$ .

## RESULTS

Predicted pseudo-unary Mg<sub>2</sub>SiO<sub>4</sub> and MgSiO<sub>3</sub> phase diagrams agree well with the experimentally observed topology and the positions and slopes of the various reactions (Figures 4, 5). The predicted  $dT/dP$  slope of the Wa=Sp and Sp=Pv+Pe boundaries are somewhat smaller than observed. The calculated majorite field is smaller than observed. This may be caused by the assumed Debye temperature, which is based on elasticity, or because of the assumption of ideal mixing on the octahedral site. A lower Debye temperature, due perhaps to anomalous vibrational properties of the octahedral site or a favorable interaction between Mg and Si on that site may be responsible for the wider observed stability field. A wider majorite field would also reduce disagreement with the observed melting curve near 16 GPa, which may also be caused by a change in thermodynamic properties of enstatite across the ortho- to clino-enstatite transition which have been ignored here. Because experimental pressure and temperature calibration is

difficult under these conditions, differences between calculated and experimental melting curves between 15 and 30 GPa may not be significant. The calculations reproduce both lower and higher pressure measurements [which determine temperature directly, *Knittle and Jeanloz, 1989a*] to within experimental precision.

The predicted topology and temperature dependence of binary OI-Wa-Sp equilibria also agree well with observation and the position of the univariant lines agree to within .1 GPa (Figures 6, 7). Complete agreement is essentially impossible because the data appear to be mutually inconsistent. At 1473 K observations of coexisting OI and Wa appear directly above observations of pure Wa near 13.5 GPa, and the observation of Sp at 14.5 GPa is difficult to reconcile with observations of the Wa=Sp end-member reaction. The fact that such inconsistencies are much less frequent in the higher temperature measurements suggests that the lower temperature experiments may not have been sufficiently equilibrated.

Calculated pseudo-binary orthosilicate phase equilibria are compared with the low and high temperature experimental results of *Ito and Takahashi [1989]* in Figures 6 and 7. By

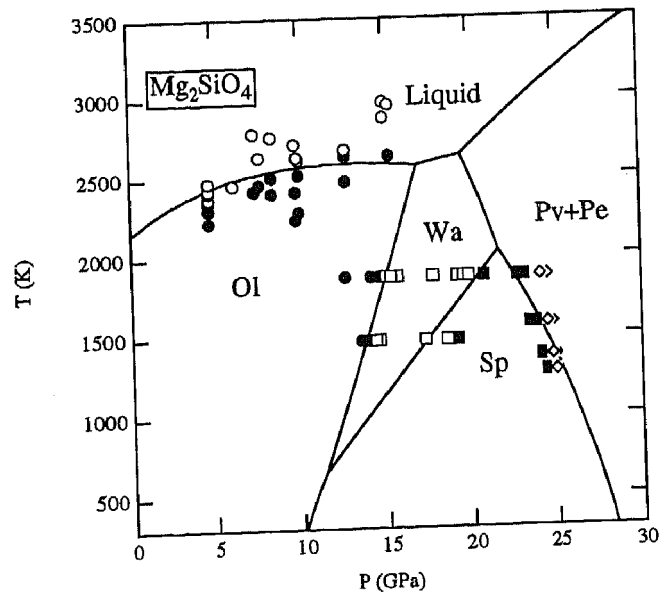


Fig. 4. Calculated pseudo-unary Mg<sub>2</sub>SiO<sub>4</sub> phase diagram compared with the experimental data of *Ohtani and Kumazawa [1981]* (OI=Liquid), *Katsura and Ito [1989]* (OI=Wa and Wa=Sp) and *Ito and Takahashi [1989]* (Sp=Pv+Pe). For the purposes of illustration we have assumed that the assemblage Pv+Pe melts congruently. Eutectic compositions are expected to lie between orthosilicate and metasilicate stoichiometries at high pressure [*Presnall and Gasparik, 1989*]. Accurate prediction of the eutectic composition requires a multi-component model for the liquid phase, which we do not consider here. Calculated equilibria are identical for the three parameter sets, except for the perovskite forming reactions which are shifted upward and downward by less than 0.5 GPa for the high and low temperature parameters, respectively. Variation of all other parameters within their estimated uncertainties (Table 1) have a smaller effect on calculated equilibria.

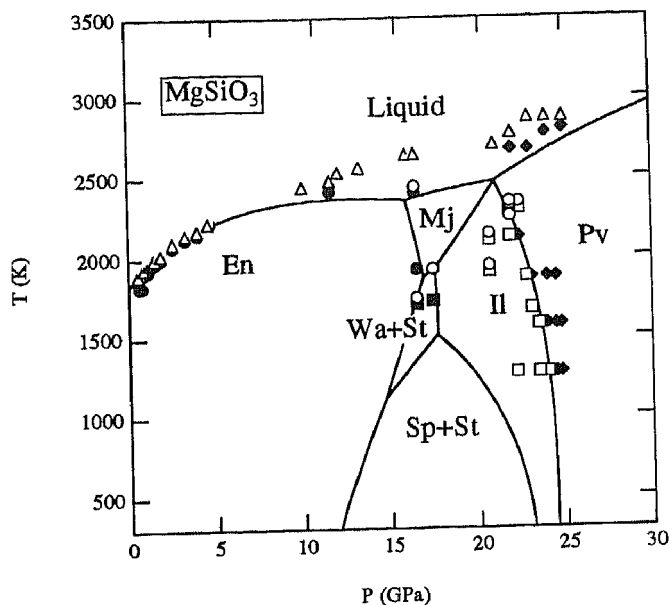


Fig. 5. Calculated pseudo-unary  $MgSiO_3$  phase diagram compared with the experimental data of *Boyd et al.* [1964] (melting below 5 GPa), *Presnall and Gasparik* [1990] (melting between 5 and 20 GPa), *Ito and Katsura* [1991] (melting above 20 GPa) *Ito and Takahashi* [1989] ( $Il=Pv$ ) and *Gasparik* [1990] (subsolidus reactions). Two distinct triple points ( $Mj=Il=Pv$ ,  $Mj=Pv=L$ ) appear coincident on this scale. Calculated equilibria are identical for the three parameter sets except for reactions involving *Il*, which are shifted upward and downward by less than 0.7 GPa for the low and high temperature parameters, respectively. Variation of all other parameters within their estimated uncertainties (Table 1) have a smaller effect on calculated equilibria.

inverting low and high temperature data separately for the reference free energies of the relevant species (Table 3), we obtained excellent fits to the observations. However, it proved impossible to reproduce both experiments with a single set of reference free energies. Parameters which give the best global fit reproduce the general topology but the calculated coexistence loops shift towards more iron rich compositions with decreasing temperature (dashed lines, Figures 6, 7) while the reverse is observed experimentally. The more complex thermodynamic model of *Fei et al.* [1991a] produces very similar results and is also unable to fit the data. The reason for this discrepancy is unclear. Barring severe deficiencies in two very different thermodynamic formulations, insufficient equilibration or an unquenchable phase transition in one or more of the relevant species may be responsible. Despite different predictions for the  $Pv=Mw+St$  reaction, all parameter sets produce a very narrow  $Sp+Pv+Mw$  coexistence field. This supports the conclusion of *Ito and Takahashi* [1989] that the spinel breakdown reaction can account for seismic reflections at 670 km depth even in a mantle of uniform composition.

Considering the discrepancies apparent in Figures 6 and 7 to be an additional source of uncertainty, the resulting errors

in our calculated phase diagrams at D'' pressure -temperature conditions are small and will not affect our basic conclusions. Differences between the predictions of high temperature and global parameter sets are reduced by increasing pressure (Figure 8). The stability field of perovskite is very wide at pressures greater than 60 GPa (1500 km depth), regardless of the parameter set used in the calculations; the low temperature parameters produce an even wider stability field of perovskite.

The predicted high pressure phase diagrams satisfy experimental constraints, including the transition of wüstite to a metallic form, observed near 70 GPa in diamond cell [ $T=4000$  K; *Knittle and Jeanloz*, 1986, 1991b] and shock wave experiments [ $T=1000$  K; *Jeanloz and Ahrens*, 1980] and perovskite syntheses up to 127 GPa and approximately 2500 K [*Knittle and Jeanloz*, 1987]. Our results closely resemble the predictions of *Jeanloz and Thompson* [1983] below 75 GPa. At higher pressures, our calculations show two pseudo-binary coexistence loops, as required by the phase rule, in place of their four-phase field.

The ternary  $MgO-FeO-SiO_2$  diagram summarizes the

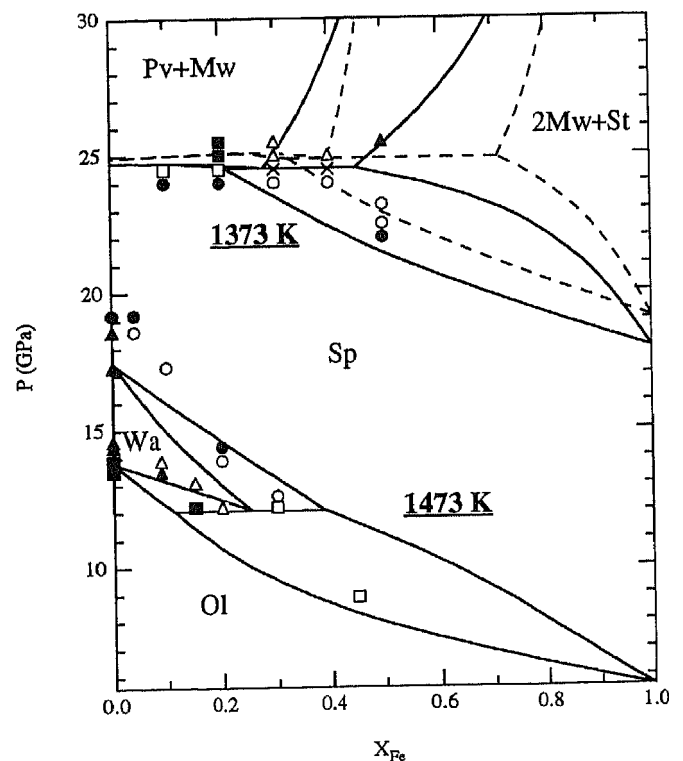


Fig. 6. Low temperature calculated pseudo-binary orthosilicate phase diagram compared with the experimental data of *Katsura and Ito* [1989] and *Ito and Takahashi* [1989]. Closed and open symbols and crosses represent observations of pseudo-unary, -binary, and -ternary coexistence. Solid lines are calculated with the low temperature parameter set, dashed lines with the global parameter set. The *Ol-Wa-Sp* equilibria differ by less than 0.02  $X_{Fe}$  units for the three parameter sets.

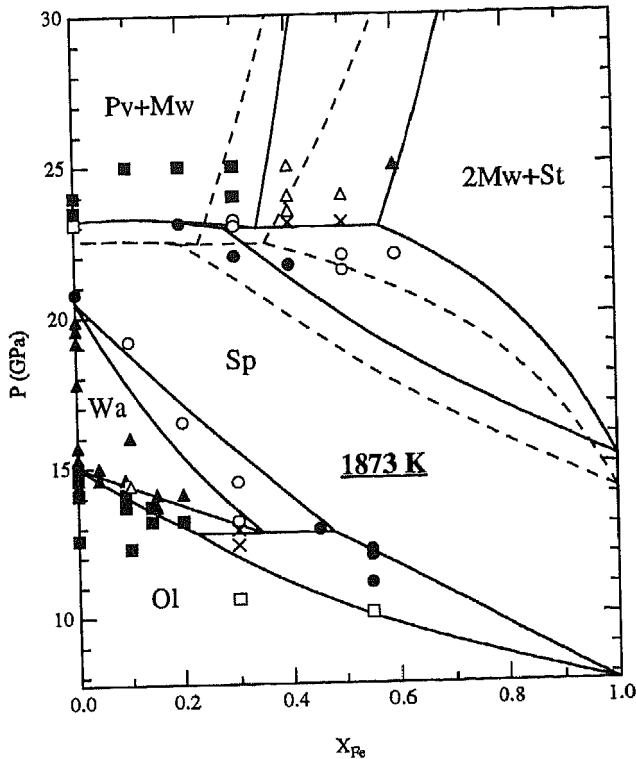


Fig. 7. High temperature calculated pseudo-binary orthosilicate phase diagram compared with the experimental data of *Katsura and Ito* [1989] and *Ito and Takahashi* [1989]. Closed and open symbols and crosses represent observations of pseudo-unary, -binary, and -ternary coexistence. Solid lines are calculated with the high temperature parameter set, dashed lines with the global parameter set. The Ol-Wa-Sp equilibria differ by less than 0.02  $X_{Fe}$  units for the three parameter sets. Variations in all other parameters within their estimated uncertainties (Table 1) have a smaller effect on calculated equilibria.

important results (Figure 9). To obtain a conservative estimate of the stability of perovskite with respect to its constituent oxides in D", we have chosen an extreme temperature [5000 K, higher than the melting point of perovskite at this pressure; *Stixrude and Bukowski*, 1990] since lower values widen the perovskite stability field. Superimposed on the phase diagram are contours of density and bulk modulus calculated with the same thermodynamic potential formalism and the same parameters used to determine the phase diagram. The intersection of the contours corresponding to the observed density and bulk modulus at the base of the mantle [*Dziewonski and Anderson*, 1981], which represent the location of the Earth at this pressure, lies well within the stability field of perovskite.

The uncertainties in seismic properties near the core mantle boundary are larger than those in the rest of the mantle but correspond to an area of the phase diagram much smaller than that of the perovskite stability field (a Backus-Gilbert resolution calculation using an improved free-oscillation data set indicates that the density and bulk modulus averaged over

the entire D" layer are uncertain by approximately 1 %; G. Masters, personal communication, 1992). Uncertainties in the calculations are also much too small to substantially alter the phase diagram. Variations in experimentally constrained parameters within their uncertainties (Table 1), and in  $W$ 's from 0 to three times their nominal values move the phase boundaries by less than 0.4  $X_{Fe}$  units. For all parameters except  $q$  of perovskite, the changes are less than 0.2  $X_{Fe}$  units. Extreme variations in the  $F_o$ 's, obtained by extrapolating their apparent temperature dependence (difference between low- $T$  and high- $T$  parameter sets) to 5000 K also fails to exclude the Earth from the perovskite stability field. A parametric phase diagram (Figure 10) shows that  $MgSiO_3$  perovskite is stable for a wide range of thermal parameters. The stability limit of perovskite in this diagram lies close to the values of  $\theta_o$  and  $\gamma_o$  adopted here, but outside their 1- $\sigma$  uncertainty limits. The diagram illustrates an extreme lower bound on the stability of perovskite since temperatures at the top of D" and the average temperature over this layer are almost certainly well below 5000 K, possibly by as much as 2000 K. A temperature of 3000 K produces perovskite stability fields in Figures 8-10 which are 25-60 % wider. Alternative values of  $K_o$  and  $K_o'$  for perovskite [246 GPa, 4.5; *Yeganeh-Haeri et al.*, 1989] or the transformation of stishovite to its recently discovered high pressure form [*Tsuchida and Yagi*, 1989] have a much smaller effect than

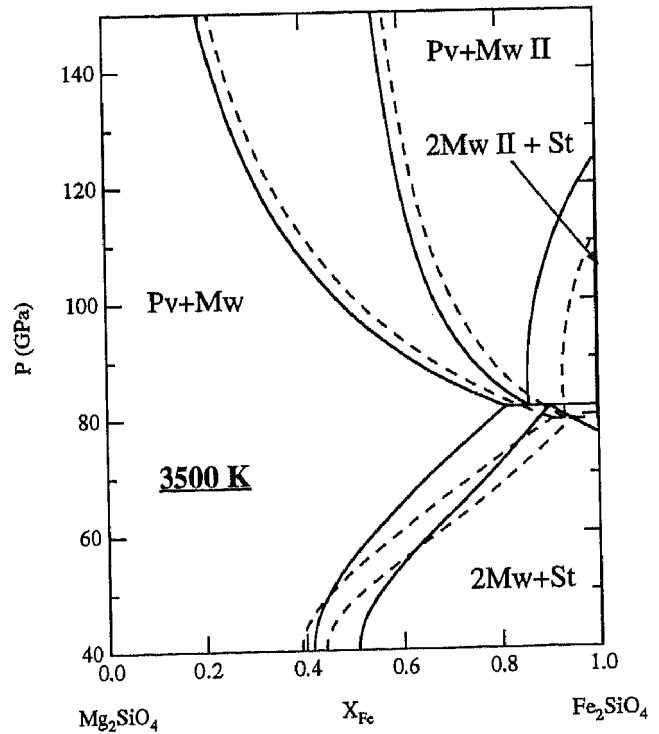


Fig. 8. High pressure orthosilicate phase diagram at 4000 K. Solid and dashed curves are calculated with the high temperature and global parameter sets respectively.

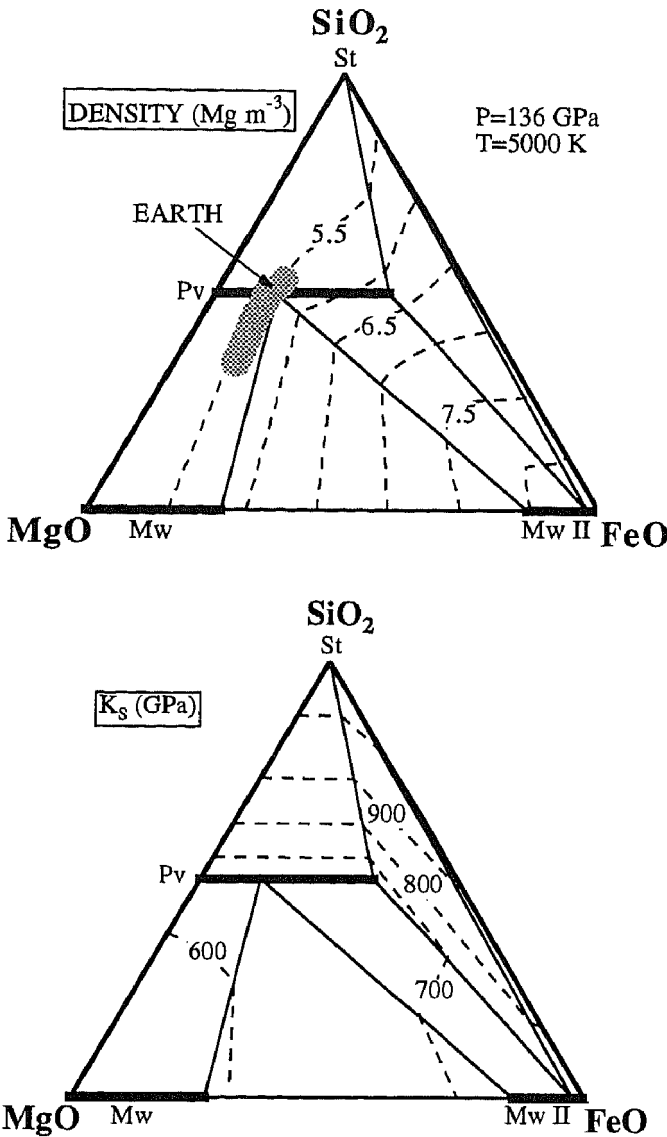


Fig. 9. Ternary MgO-FeO-SiO<sub>2</sub> phase diagram at 136 GPa and 5000 K calculated with the high temperature parameter set. Contours of density (upper figure) and adiabatic bulk modulus (lower figure) are superimposed. The shading, shown only in the upper figure for clarity, indicates the field of possible intersections of the contours corresponding to the observed density and adiabatic bulk modulus of the Earth allowed by uncertainties in calculated and seismically observed densities and bulk moduli. Adopting a value of 205 GPa for the room temperature phase transition in periclase eliminates Mw from the diagram and shifts the Pv+Mw II+St coexistence field towards lower iron contents ( $X_{Fe}(Pv)=0.58$ ,  $X_{Fe}(Mw II)=0.83$ ).

variations in thermal parameters on the location of the Earth and the position of phase boundaries on the diagram. Perhaps the largest uncertainty lies in the pressure of the assumed phase transition in periclase. The first principles calculations of *Bukowinski* [1985] provide a probable lower

bound on the transition pressure of 205 GPa at 300 K, much lower than the value adopted here from more recent results [500 GPa, *Mehl et al.*, 1988; this value is also consistent with the results of *Agnon and Bukowinski*, 1990; and *Zhang and Bukowinski*, 1991 which are based on modified electron gas theories]. Adopting the lower value expands the Mw II+St field somewhat but not enough to encompass the Earth.

CONCLUSIONS

Our results indicate that the breakdown of (Mg,Fe)SiO<sub>3</sub> perovskite to its constituent oxides is unlikely in the Earth. If core-mantle boundary temperatures are much lower than those derived from shock wave experiments [*Williams et al.*, 1987; *Jeanloz*, 1990], several alternative explanations for the seismic reflectors at the base of the mantle are possible. Small amounts of stishovite may be responsible, from silica enrichment relative to metasilicate stoichiometry (Figure 9). The phase transition in magnesiowüstite to its high pressure form may produce anomalous velocity gradients near the core-mantle boundary, although it appears much too diffuse to produce a seismic discontinuity (Figure 8). Still unobserved

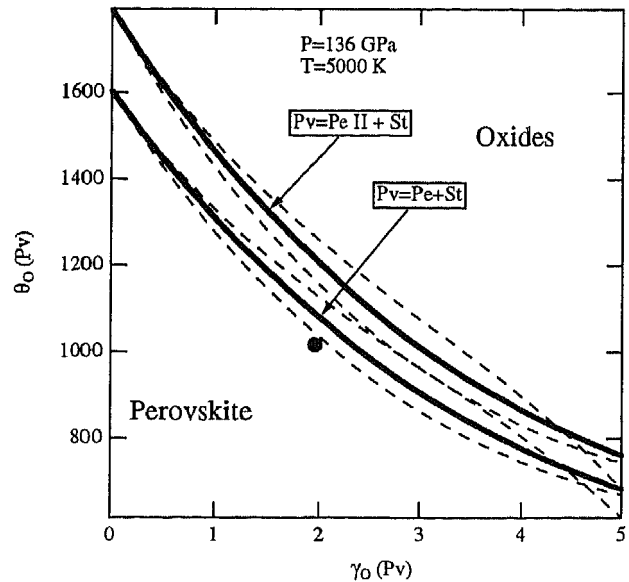


Fig. 10. Parametric phase diagram showing extreme lower bounds on the size of the stability field of MgSiO<sub>3</sub> perovskite relative to its constituent oxides at the pressure-temperature conditions of D<sup>''</sup>. The bold curves indicate the stability limit of perovskite for  $q(Pv)=2.5$ , while the lower and upper surrounding dashed lines show the limit for  $q(Pv)=1.5$  and 3.5 respectively. The circle represents the best fitting values adopted here (Figure 3; and Table 1). The size of the symbol is comparable to the 1- $\sigma$  uncertainties. Adopting different forms for the thermodynamic potentials (e.g. an Einstein model for the thermal energy) may lead to best fit values which fall outside our uncertainties. However, the position of the best fit values relative to the breakdown reactions is unlikely to change by more than a few percent in  $\theta_0$  or 10 % in  $\gamma_0$ . Lowering the temperature to 3000 K increases the stability of perovskite and shifts the curves towards higher values of  $\gamma_0$  by 0.5.



phase transitions in calcic, diopsidic or  $(\text{Mg,Fe})\text{SiO}_3$  perovskite, including the predicted orthorhombic to cubic transition in the latter [Wolf and Bukowinski, 1985; Bukowinski and Wolf, 1988; see also Hemley and Cohen, 1992], may produce velocity discontinuities at the appropriate pressures. Partial reaction of Mg-rich perovskite with iron alloys, similar to that observed at lower pressures [Knittle and Jeanloz, 1989a, 1991a], may also produce seismic reflectors.

All of these explanations rely on perovskite dominated assemblages and thus cannot reconcile geotherms based on shock wave experiments with the lack of partial melt in D". The melting point of iron under shock loading, and in static compression experiments at lower pressures is currently a matter of active debate [Ross et al., 1990; Boehler et al., 1990]. However, if the shock wave data and the predicted perovskite melting curve are approximately correct, they require either the absence of Mg-rich perovskite from D" or a substantial change in its properties, probably much greater than that associated with the predicted orthorhombic to cubic phase transformation. An exotic bulk composition in D", perhaps dominated by Ca and Al oxide components [Ruff and Anderson, 1980], may be required. Alternatively, a breakdown or transformation of  $(\text{Mg,Fe})\text{SiO}_3$  perovskite to a more diverse assemblage than those considered here remains a possible explanation of the seismic reflectors and the absence of melt in D". Consideration of other chemical components may lower the free energy of the non-perovskite assemblage, inducing a breakdown to a mixture of oxides and metallic alloys (Fe, FeSi). Though speculative, a metallic alloy-oxide layer may be heterogeneous on many length scales because of the disparate physical properties of the two components [Knittle and Jeanloz, 1991a]. This provides a possible explanation for observed heterogeneity in D" on length scales ranging from the smallest observable with seismic probes (100 km, roughly one seismic wavelength) to nearly the diameter of the core [Lavelly et al., 1986; Creager and Jordan, 1986; Morelli and Dziewonski, 1987; Young and Lay, 1987; Weber and Körnig, 1990; ].

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