

Tuning dissociation using isoelectronically doped graphene and hexagonal boron nitride: Water and other small molecules

Yasmine S. Al-Hamdani, Dario Alfè, O. Anatole von Lilienfeld, and Angelos Michaelides

Citation: *The Journal of Chemical Physics* **144**, 154706 (2016); doi: 10.1063/1.4945783

View online: <http://dx.doi.org/10.1063/1.4945783>

View Table of Contents: <http://scitation.aip.org/content/aip/journal/jcp/144/15?ver=pdfcov>

Published by the [AIP Publishing](#)

Articles you may be interested in

[The different adsorption mechanism of methane molecule onto a boron nitride and a graphene flakes](#)

J. Appl. Phys. **116**, 153507 (2014); 10.1063/1.4898336

[Tuning the band structure, magnetic and transport properties of the zigzag graphene nanoribbons/hexagonal boron nitride heterostructures by transverse electric field](#)

J. Chem. Phys. **141**, 014708 (2014); 10.1063/1.4885857

[Interlayer coupling enhancement in graphene/hexagonal boron nitride heterostructures by intercalated defects or vacancies](#)

J. Chem. Phys. **140**, 134706 (2014); 10.1063/1.4870097

[Graphene quantum dots embedded in hexagonal boron nitride sheets](#)

Appl. Phys. Lett. **98**, 013105 (2011); 10.1063/1.3533804

[Defects-enhanced dissociation of H₂ on boron nitride nanotubes](#)

J. Chem. Phys. **124**, 054706 (2006); 10.1063/1.2162897



NEW Special Topic Sections

NOW ONLINE
Lithium Niobate Properties and Applications:
Reviews of Emerging Trends

AIP | Applied Physics
Reviews

Tuning dissociation using isoelectronically doped graphene and hexagonal boron nitride: Water and other small molecules

Yasmine S. Al-Hamdani,^{1,2} Dario Alfè,^{1,3} O. Anatole von Lilienfeld,⁴
 and Angelos Michaelides^{1,5,a)}

¹Thomas Young Centre and London Centre for Nanotechnology, 17–19 Gordon Street, London WC1H 0AH, United Kingdom

²Department of Chemistry, University College London, 20 Gordon Street, London WC1H 0AJ, United Kingdom

³Department of Earth Sciences, University College London, Gower Street, London WC1E 6BT, United Kingdom

⁴Institute of Physical Chemistry and National Center for Computational Design and Discovery of Novel Materials (MARVEL), Department of Chemistry, University of Basel, Klingelbergstrasse 80, CH-4056 Basel, Switzerland

⁵Department of Physics and Astronomy, University College London, Gower Street, London WC1E 6BT, United Kingdom

(Received 23 December 2015; accepted 23 March 2016; published online 21 April 2016)

Novel uses for 2-dimensional materials like graphene and hexagonal boron nitride (h-BN) are being frequently discovered especially for membrane and catalysis applications. Still however, a great deal remains to be understood about the interaction of environmentally and industrially relevant molecules such as water with these materials. Taking inspiration from advances in hybridising graphene and h-BN, we explore using density functional theory, the dissociation of water, hydrogen, methane, and methanol on graphene, h-BN, and their isoelectronic doped counterparts: BN doped graphene and C doped h-BN. We find that doped surfaces are considerably more reactive than their pristine counterparts and by comparing the reactivity of several small molecules, we develop a general framework for dissociative adsorption. From this a particularly attractive consequence of isoelectronic doping emerges: substrates can be doped to enhance their reactivity specifically towards either polar or non-polar adsorbates. As such, these substrates are potentially viable candidates for selective catalysts and membranes, with the implication that a range of tuneable materials can be designed. *Published by AIP Publishing.* [<http://dx.doi.org/10.1063/1.4945783>]

I. INTRODUCTION

Amongst the many materials being studied for chemical applications, 2-dimensional (2D) materials like graphene and hexagonal boron nitride (h-BN) are some of the most versatile and interesting, thanks to their novel properties and sustainable compositions. Properties of graphene and h-BN manifest themselves in various important applications such as desalination,¹ water purification,² energy storage,³ energy generation,^{4–8} and catalysis.^{9–11} For example, sizeable voltages have been measured from forming water salinity gradients across graphene sheets and nanotubes,^{4–7} and Siria *et al.* demonstrated that water flowing osmotically through a BN nanotube produces remarkably large electric currents.⁸ This was attributed to the possible dissociation and adsorption of water on the interior of the nanotube which influences the dynamics inside the nanotube.

Much of the work on graphene and h-BN is also motivated by the sustainability and the availability of the component elements — an aspect which can be difficult to meet using materials that contain transition or noble metals.¹² Already, hydrogenated h-BN is thought to be a potential photocatalyst as a material that is active under visible light and has a band gap roughly in line with the reduction and oxidation potentials

of water.⁹ Similar efforts are being made to develop graphene into a photocatalyst by modification of its band gap, and also as a support to other photocatalytic materials.^{10,11}

Despite the promising applications of h-BN and graphene as membranes and catalysts, there are still major gaps in our understanding of the interaction of molecules like water on clean graphene and h-BN surfaces on the atomic level, and even less is known about how doping in the materials alters their interaction with molecules. Indeed, experimental routes to produce hybrid composites of h-BN and graphene^{13,14} have emerged with high levels of control being reported on the nanometre scale, which is another reason to gain better atomic level understanding. Various theoretical studies on band gap engineering using h-BN and graphene mixtures,^{15–23} have revealed the tuneability of these materials through the mixture of atoms. Other studies have focused on exploiting this tuneability for catalysis of oxygen reduction reactions,^{24–33} water transport,³⁴ and H₂ adsorption.^{35–37}

An important aspect to consider, if using graphene and h-BN based materials as catalysts, is their degree of selectivity. A high degree of selectivity is an extremely desirable property for any catalyst and indeed, the rational design of metal-based heterogeneous catalysts is the focus of intense research (see, e.g., Refs. 38–44). However, even in these cases, the metal-based catalysts do not necessarily have very different selectivities, and although they can be doped or alloyed

^{a)}Electronic mail: angelos.michaelides@ucl.ac.uk

to vary their reactivity, the effect on reaction energies and barriers is often a constant shift with respect to different molecules.^{44–47} For instance, in the reaction pathways towards H₂ formation discussed by Cortright *et al.*, a metal catalyst is used throughout, which also catalyses H₂ consuming reactions instead.⁴⁸ Meanwhile, Guo *et al.* have shown that a more complex selective catalyst gives rise to a higher conversion rate of methane to H₂.⁴⁹

Here we investigate water and some other environmentally and industrially relevant small molecules with density functional theory (DFT). The particular focus of this study is to establish the thermodynamics of dissociative adsorption and how this is affected by doping. From this work we draw a number of conclusions. First, doping strongly affects the dissociation process, in some situations making dissociation more favourable by several electronvolts. Second, different surfaces have varying reactivity for the set of molecules considered, with some substrates significantly enhancing the reactivity of polar molecules and others enhancing the reactivity of non-polar adsorbates.

Below, we begin by describing our computational setup in Section II and present our DFT results for water adsorption in Section III A, followed by an overview regarding the relative adsorption of other molecules in Section III B. In Section IV we discuss the trends observed in adsorption sites and structures, and propose a general framework for dissociative adsorption before finally concluding, in Section V.

II. METHODS

The dissociative adsorption of a water monomer and other molecules on graphene, h-BN, and their doped counterparts was calculated using DFT and the Vienna *Ab Initio* Simulation Package (VASP) 5.3.2.^{50–53} VASP uses plane-wave basis sets and projector augmented wave (PAW) potentials^{54,55} to model the core region of atoms.

A. System setup

The graphene and h-BN substrates are modelled using (5 × 5) hexagonal unit cells containing 50 atoms, for which adsorption energies are converged to less than 10 meV with respect to (7 × 7) unit cells. After a series of convergence tests for the plane-wave cutoff energy, we chose to use a 400 eV energy cutoff, which gives dissociative adsorption energies converged to within 16 meV of a 600 eV energy cutoff. Γ -point sampling of reciprocal space for the (5 × 5) cell was used but \mathbf{k} -point densities up to (9 × 9 × 1) were tested. Adsorption energies using Γ -point sampling are within 50 meV (3%) of the converged adsorption energies for all substrates. Spin polarisation was applied since H pre-adsorption on the substrates gives rise to spin polarised states. A 10 Å separation in the z-direction between substrates without a dipole correction proved to be converged for dissociative adsorption energies of water compared to using a dipole correction or a 20 Å separation (<15 meV difference).⁵⁶

For the dissociative adsorption energies evaluated here (spanning a few eV) we have mostly used the Perdew-Burke-Ernzerhof (PBE)⁵⁷ generalised gradient approximation

exchange-correlation functional. However we have also verified that the key results obtained here are not particularly sensitive to the choice of exchange-correlation functional, as discussed in Section IV.

There are many different ways of isoelectronically doping graphene with BN and *vice versa* and as a first step we focus on low concentrations of doping: one pair of BN substituting two C atoms in a (5 × 5) unit cell of graphene which we refer to as boron nitride doped graphene (BNDG) and likewise, two C atoms substituting a BN pair in a (5 × 5) unit cell of h-BN, henceforth referred to as 2CBN (Fig. 1). Doped substrates are modelled by isoelectronically doping the pristine sheets and relaxing the unit cells using a plane-wave energy cutoff of 600 eV to remove any strain introduced by the mixture of B, N and C atoms. Relaxation effects are small: less than 1% of the relaxed lattice constant of the undoped system.⁵⁸

When water dissociates on a 2D substrate there are a number of possible adsorption scenarios. Here, we have focused on four possible outcomes. Schematic illustrations are given in Fig. 2 and in brief they involve the following: (i) An OH group on the surface and the release of (half) an H₂ molecule, referred to as “OH (1/2H₂ gas)”;

(ii) the adsorption of both OH and H components of water on the surface, with them both being on one side of the substrate, namely, “*cis*(OH–H).” We consider this configuration to be particularly important because 2D materials tend to be examined by supporting them on other materials, leaving only one side of the surface exposed;

(iii) the adsorption of both OH and H on the surface but this time on opposite sides of the substrate, referred to here as “*trans*(OH–H)”. This could arise from having the substrate suspended in a wet environment or from the H atoms diffusing through the sheet and there are indications that graphene and h-BN are permeable to protons.⁵⁹ However, since it is not clear how likely it is for molecules to dissociate on different sides of the substrates, we consider this *trans*(OH–H) configuration to be less relevant

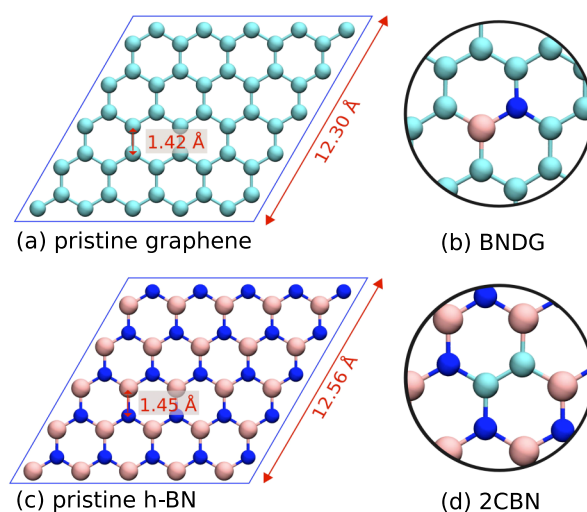


FIG. 1. The clean and doped graphene and h-BN surfaces considered in this study. (a) (5 × 5) unit cell of graphene. (b) BN doping in (5 × 5) unit cell of graphene which we refer to as BNDG. (c) (5 × 5) unit cell of h-BN. (d) C doping in (5 × 5) unit cell of h-BN, referred to as 2CBN. For clarity only a small portion of the (5 × 5) unit cell is shown in (b) and (d). C is coloured cyan, B is pink, and N is blue.

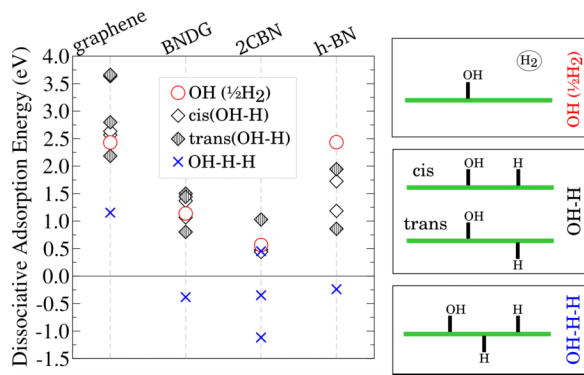


FIG. 2. The dissociative adsorption energy of water on graphene, BNDG, 2CBN, and h-BN is shown for different adsorption structures. Red circles indicate the adsorption of OH from water onto the substrate and the release of hydrogen gas. The black diamonds indicate the dissociative adsorption of a water molecule into OH and H on the substrate. The blue crosses correspond to the adsorption energies on a hydrogenated substrate. The categories of water dissociation on the substrate are illustrated on the right.

than *cis*(OH–H); (iv) lastly, “OH–H–H” which is again the adsorption of both OH and H, this time on a surface that has an H atom pre-adsorbed. We tested this particular setup in light of previous experimental and simulation work, where this is thought to cause water dissociation.⁸ Many adsorption sites are available for each category and we have calculated only a number of possibilities: *ortho*, *meta*, and *para*-positioning of the adsorbed components with respect to each other, as well as adsorption of the components far away from each other and the doping site in the substrate.

The absolute adsorption energy for dissociative adsorption, E_{ads} is defined as

$$E_{ads} = E_{ads/sub}^{tot} - E_{sub}^{tot} - E_{ads}^{tot}, \quad (1)$$

where $E_{ads/sub}^{tot}$ is the total energy of the adsorption system, E_{sub}^{tot} is the total energy of the relaxed substrate, and E_{ads}^{tot} is the energy of the intact molecule in the gas phase. Equation (1) is used for all but one adsorption state, that is OH (1/2H₂ gas). Here we also take into account the energy ($E_{H_2}^{tot}$) of the 1/2H₂ gas molecule that is formed,

$$E_{ads} = E_{ads/sub}^{tot} + 1/2E_{H_2}^{tot} - E_{sub}^{tot} - E_{ads}^{tot}. \quad (2)$$

Within these definitions, negative adsorption energies correspond to favourable (exothermic) adsorption processes. Bond strengths of hydrogen and hydroxyl to the surfaces are calculated with respect to a gas phase hydrogen atom or hydroxyl group instead of the whole molecule,

$$E_{bond} = E_{sub}^{tot} + E_{ads}^{tot} - E_{ads/sub}^{tot}. \quad (3)$$

III. RESULTS

We begin with the results for the dissociative adsorption of water on the pure substrates, graphene and h-BN, and on the doped substrates, BNDG and 2CBN. In general, we find that the dissociation of water is more facile on the doped substrates and is also strongly affected by the presence of a pre-adsorbed H atom, local electronic induction, and steric effects arising from rehybridisation of orbitals in the substrate atoms. We

use these insights to look at the adsorption of H₂, methane, and methanol on the same surfaces in Section III B. From our analysis, we see that different substrates favour the dissociation of different molecules, depending on their polarity, enabling us to make comparisons between the adsorption behaviour of polar and non-polar molecules and fragments.

A. Dissociative adsorption of water on graphene, h-BN, BNDG, and 2CBN

Fig. 2 reports results for the dissociative adsorption of water on the clean and doped substrates. It can be seen that the energetics of the dissociation process varies significantly for the various adsorption structures and substrates.

On pristine graphene we find that dissociation is strongly endothermic in agreement with previous work.^{60,61} In addition the energy of the dissociation process varies by as much as 2 eV depending on the final adsorption configuration. The lowest adsorption configuration on pristine graphene is *trans*(OH–H) (2.19 eV) with OH and H in *ortho* positions, in agreement with the work of Xu *et al.*⁶¹ The *cis*(OH–H) configuration shown in Fig. 3(a) on graphene has a dissociative adsorption energy of 2.57 eV and is thus ~0.4 eV less stable than *trans*(OH–H). Dissociative water adsorption is in general more thermodynamically favourable on h-BN than on graphene. For example, the *cis*(OH–H) state on pristine h-BN shown in Fig. 3(d) has E_{ads} of 1.19 eV and is 1.38 eV more favourable than the equivalent structure on graphene. Nonetheless, given just how thermodynamically unfavourable water dissociation is, it is unlikely that water monomers will dissociate on pristine graphene and h-BN.

Upon moving to the doped substrates, for which numerous configurations were considered, we find a significant lowering in the energy to adsorb water. From graphene to BNDG, and from h-BN to 2CBN, we gain ~1 eV in the adsorption of a water molecule. The *cis*(OH–H) state and lowest energy dissociation state for each doped surface is shown in Fig. 3. On both BNDG and 2CBN, B–OH and C–H bonds are formed. Note from Table I that the B–OH bond is ~1.3 eV stronger on BNDG than on h-BN (or C–OH on graphene). Hence, a marked activation of the B atom towards binding OH results from the mixture of N and C atoms surrounding it and in this

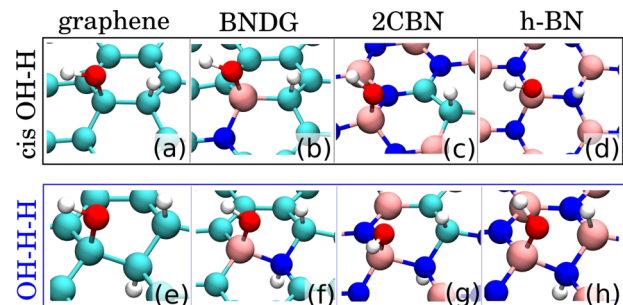


FIG. 3. The most stable *cis*(OH–H) (top panel) and most stable overall dissociative adsorption structures (lower panel) of water on graphene, h-BN, BNDG and 2CBN are shown. (a) and (e) are water on pristine and hydrogenated graphene, respectively. (b) and (c) show water adsorbed on BNDG and 2CBN, whilst (f) and (g) show water adsorption on the hydrogenated counterparts. (d) and (h) are on pristine and hydrogenated h-BN, respectively.

TABLE I. Bond strengths (in eV) for H and OH on graphene, h-BN, and BNDG sheets with respect to a gas phase hydrogen atom or OH molecule. Parentheses indicate neighbouring atoms in the substrate. Negative bond energies correspond to endothermic but metastable adsorption minima. No minimum was found for OH adsorbed on the N atom.

Bond	Bond strength (eV)
Graphene	
C-H	0.81
C-OH	0.67
h-BN	
N-H	-0.77
B-H	-0.01
B-OH	0.67
BNDG	
B-H	0.98
N-H	0.07
(B)C-H	1.15
(N)C-H	1.04
B-OH	1.96
(B)C-OH	0.84
(N)C-OH	1.03

way doping leads to a considerable lowering of the dissociative adsorption energy.

The presence of the pre-adsorbed H atom also significantly improves the thermodynamics of water adsorption by ~ 1 eV for each substrate. Favourable OH-H-H configurations are shown in Fig. 3, and from Fig. 2 it can be seen that water splitting is thermodynamically favourable on the hydrogenated h-BN (-0.24 eV), BNDG (-0.38 eV), and 2CBN (-1.12 eV) surfaces. Thus doping and hydrogenating both graphene and h-BN makes the thermodynamics of water dissociation considerably more favourable. The general conclusion that pre-adsorbed hydrogen facilitates water dissociation is in agreement with the work of Siria *et al.*⁸ Interestingly, the overall most favourable states for water dissociation on the doped surfaces contain a B-N-C construction in the surface where B-OH, N-H, and C-H bonds are formed. We considered if the increased reactivity at these sites is due to the pre-adsorbed H atom on a N site destabilising the surface and thus activating it towards water adsorption, but this is unlikely because the N-H bond is very weak (only 0.07 eV). The B-N-C construction in the surface of doped substrates is therefore central to making the dissociation energy more exothermic and exemplifies the use of isoelectronic doping to tune the dissociative adsorption energy of water. In all OH-H-H states, the OH and H components of the dissociated water are arranged in a hydrogen bonded fashion. The hydrogen bond on h-BN at 1.95 Å is shorter than the hydrogen bond on graphene (2.23 Å) despite the slightly smaller lattice constant of graphene. The hydrogen bonding distances are indicative of the more polarised binding of OH and H on h-BN, which culminates in a more negative oxygen atom in the OH group and hence a shorter hydrogen bond.

Additional DFT calculations of water dissociation on the protonated (as opposed to hydrogenated) substrates were also performed. A homogeneous background charge is added

in the DFT calculations of the charged systems so that the electrostatic interactions do not diverge and can be computed under periodic boundary conditions. These reveal that protonation is slightly less effective than hydrogenation but still increases the tendency of water to dissociate by ~ 0.8 eV with respect to the non-protonated clean surfaces. Thus either hydrogen pre-adsorption or acidic conditions (pre-adsorbed protons) could be key elements in the activation of these sheets towards dissociative water adsorption.

Before moving on to discuss the other adsorbates, two additional features of these adsorption systems deserve comment. First, adsorption of the dissociated fragments on separate sides of the sheet (so-called *trans*-adsorption) is favoured in general. Specifically, *trans-ortho*(OH-H) adsorption is ~ 0.4 eV more stable than *cis-ortho*(OH-H) on graphene. This is consistent with previous work on graphene⁶²⁻⁶⁶ and demonstrates the stabilisation gained by adhering to a more tetrahedral structure around the sp^3 hybridised C atom. Likewise on h-BN and BNDG, the tetrahedral arrangements of *trans*(OH-H) and OH-H-H lead to lower dissociative adsorption energies (by about 0.3 eV). Note the 2CBN system is an exception and the most stable (OH-H) configuration on 2CBN has *cis-para*-positioning, shown in Fig. 3(c). The *trans-ortho*(OH-H) state on 2CBN is still close in energy and only 0.04 eV less stable than *cis-para*.⁶⁷ This can be explained by the difference in partial charges on the B atoms bonding to OH in each case. Electronegative N atom neighbours make B atoms more positive and subsequently form a stronger polar bond with OH. In the *trans-ortho* state, the B atom is surrounded by only two N atoms and hence is not as electrophilic as the B atom in the *cis-para* state which is bonded to three other N atoms. This example in 2CBN demonstrates that inductive effects from neighbouring atoms dominate over steric effects. Despite the advantage of satisfying the sp^3 hybridisation in *trans*-adsorption states, it is important to remember that in practice 2D materials are often suspended or grown over substrates⁶⁸⁻⁷⁶ (metals or silicon carbide) where *cis* configurations are more likely.

Second, inductive effects are also introduced by the adsorbed water fragments. This can be seen by comparing the co-adsorbed to the separately adsorbed OH and H fragments. Specifically, OH ($1/2H_2$ gas) adsorption on graphene and h-BN only differ by 5 meV and indeed the C-OH and B-OH bonds (as listed in Table I) in graphene and h-BN are almost identical. In contrast, C-H bonds in graphene are significantly stronger than N-H bonds in h-BN, implying that OH-H on graphene might be more stable, and yet water adsorption is more exothermic on h-BN. It follows that the binding of hydrogen atoms on the surface perturbs the local electronic structure and therefore, the bond strength of OH to the surface, such that the OH-H configuration is considerably more stable on h-BN than on graphene.

It is useful to explain these trends in terms of the physical properties of the surfaces and we have done this by looking at Bader charges,^{77,78} average electrostatic potentials at each atom, and Kohn-Sham orbitals of the dissociated states.⁷⁹ Comparison of the adsorption structures and Bader charges suggests the most stable adsorption states arise from the

following: (i) C–H in which the C site has the most negative partial charge across the surface; (ii) B–OH in which the B atom is positive and susceptible to nucleophilic attack; and (iii) N–H in which the N atom is the most negative and therefore the strongest nucleophile. A careful analysis reveals that the adsorption of water is affected by a combination of factors involving orbital overlap and electrostatic interactions. Graphene has weaker electrostatic interactions with water than h-BN, but better orbital overlap (evidenced by bond strengths in Table I). In contrast, hybrids of h-BN and graphene have stronger electrostatic interactions with water than graphene, and also stronger orbital overlap with water than h-BN. Due to these combined effects, doped graphene and h-BN are more suited for the adsorption of water. Evidently for a given substrate, electrostatic interactions with a molecule are determining the site of adsorption (e.g., in 2CBN the *cis-para* state of water is more stable than the *trans-ortho*).

To recap, isoelectronic doping has a significant impact on the thermodynamics of water dissociation of graphene and h-BN. The most thermodynamically favourable adsorption identified is the OH–H–H configuration on 2CBN with an adsorption energy of -1.12 eV. The strong adsorption energy on 2CBN can be attributed to the following: (i) the B–OH bond in which the B atom is more positive compared to B atoms in the other substrates; and (ii) a stronger C–H bond at 2CBN as opposed to a B–H bond at h-BN.

B. Dissociative adsorption of hydrogen, methane, and methanol

With the insight gained from water adsorption, we also calculated the dissociative adsorption of H₂, methane, and methanol. As before, various configurations were calculated for each system, and in Fig. 4(a) we report the most favourable dissociation energies found for the molecules on the same side (*cis* configurations) of the pure and doped substrates. The change in zero point energy (ZPE) upon dissociative adsorption for each system is also included in the energies in Fig. 4. ZPEs were calculated using the harmonic approximation and we find that the change in ZPE increases the dissociative adsorption energies by up to 0.3 eV, which is certainly not insignificant. In some cases, the adsorption energies of the *trans*-states are more favourable than *cis* but since it is more feasible for adsorbates to dissociate on the same side of the substrate, we show results only for *cis* configurations.

From these calculations with the other adsorbates, we learn two key things. First, doping of the pristine substrates helps the thermodynamics of dissociation for these molecules too. Second, the details are quite different, with methanol behaving in a similar manner to water by benefiting most from BN doping in graphene, whereas H₂ and methane benefit most from C doping in h-BN. Figs. 4(b) and 4(c) illustrate this latter point by showing the gain in dissociative adsorption energy for each molecule as a result of doping in the pristine substrates.

The adsorption of methanol varies in a similar way to water across the different substrates and favours the same adsorption sites (*ortho* on graphene, BNDG, and h-BN, and

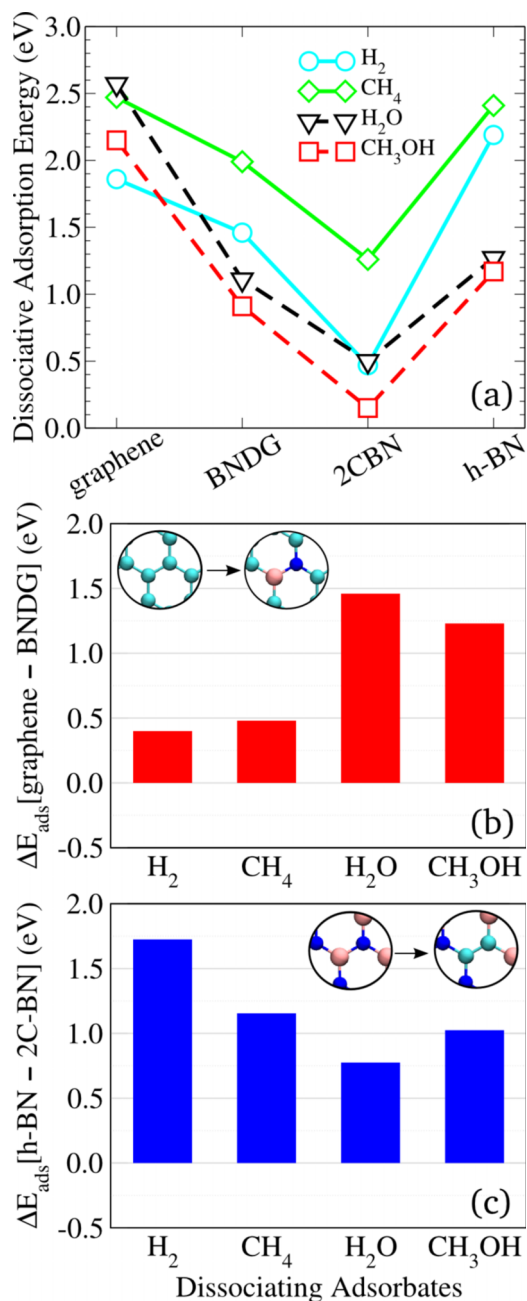


FIG. 4. (a) Dissociative adsorption energies including ZPE contributions of H₂, methane, water, and methanol on graphene, BNDG, 2CBN, and h-BN. H₂ in blue circles, methane in green diamonds, water in black triangles, and methanol in red squares. Results are given only for the most stable adsorption structure for each molecule and substrate with the fragments adsorbed on the same side of the substrate and without pre-adsorbed hydrogen. (b) Gain in adsorption energies from doping pristine graphene with BN (in eV) for different molecules, illustrating a marked increase in the reactivity with polar adsorbates. (c) Gain in adsorption energies from doping pristine h-BN with 2C (in eV) for different molecules and here the reactivity with non-polar adsorbates increases more significantly. The insets in (b) and (c) illustrate the doping.

para on 2CBN). From Fig. 4, we see that water and methanol adsorption energies both become more favourable by ~ 1.4 eV as the substrate is changed from graphene to BNDG. Having established that the C–OH to B–OH change in bond energy is the main contributor to the difference in adsorption energies for water on graphene and BNDG, we can deduce that the

same is true for methanol. Note that the adsorption of methanol is stronger than that of water on all substrates by 0.2–0.4 eV. On graphene, BNDG, and 2CBN the C–O bond of methanol is broken preferentially with the CH₃ fragment bonding to the substrate at the same sites as the H from water does. However on h-BN, the O–H bond is broken instead, resulting in N–H and B–OCH₃ bonds with the h-BN substrate.

Meanwhile the non-polar molecules, H₂ and methane, also benefit from doping of the pristine substrates but in particular from C doping in h-BN. This appears to be because the alkene-like bond between the two C atoms, which is susceptible to alkene addition reactions, is particularly effective at breaking weakly polarised bonds. Methane and H₂ follow exactly the same trend but H₂ is adsorbed around 0.6 eV more strongly overall.

By tracking the lowest adsorption states across the substrates in Fig. 4(a), we see that the preference for H₂ and water switch; H₂ adsorbs preferably on graphene and water is preferred on BNDG and pure h-BN. H₂ and water have almost the same dissociative adsorption energies on 2CBN (~0.5 eV). The different adsorption preferences that depend on the isoelectronic substrate doping is a significant outcome, especially given that these materials are composed of sustainable and abundant elements, making them desirable candidates for catalysis.

Finally, as with water adsorption, we also examined the effect of H pre-adsorption on the dissociative adsorption energy of these small molecules. We found in a similar manner to water that dissociative adsorption becomes more favourable by 0.7 – 1.5 eV on the hydrogenated surfaces, such that H₂, water, and methanol have exothermic dissociative adsorption energies on BNDG, 2CBN, and h-BN. Thus, as with water, doping and hydrogenation significantly improve the energetics of dissociative adsorption on graphene and h-BN.

IV. DISCUSSION AND GENERAL FRAMEWORK

Some important trends can be observed from the adsorption structures and energies of water and the other molecules studied here, which are likely to apply in general to polar and non-polar adsorbates on BNDG and C doped h-BN systems. Although we have studied water adsorption more extensively, the trends also hold for H₂, methanol, and methane. We summarise as follows:

- Isoelectronic doping of graphene with BN increases the reactivity with polar adsorbates (i.e., water and methanol) by ~1.4 eV but only changes the reactivity with non-polar adsorbates by ~0.5 eV. Conversely, isoelectronic doping of h-BN with C increases the reactivity most with H₂ and methane, by 1.2–1.8 eV.
- Hydrogen atom (or proton) pre-adsorption on the substrate significantly improves the thermodynamics of dissociation for the molecules considered by ~1 eV (~0.8 eV), resulting in exothermic dissociative adsorption, and suggesting that acidic conditions aid dissociation on the substrates.
- The most exothermic adsorption sites for polar adsorbates share the B–N–C construction in which

there is already a H atom pre-adsorbed on a N atom. Meanwhile, non-polar adsorbates favour C–C sites with localised electrons (as in 2CBN).

- Local electronic inductive effects dominate over steric effects. In other words, *para*-positioning of molecule fragments is possible (however *ortho* is generally favoured) if the atoms in the substrate have a larger electrostatic potential in the *para*-sites.
- Atoms in the substrate that change to sp³ hybridisation as a result of chemisorption prefer to be in a more tetrahedral arrangement, e.g., the *trans-ortho* configuration is ~0.3 eV more stable than *cis-ortho*.

Some comments related to these trends are appropriate. First, all the numbers given have been derived from the PBE exchange-correlation functional. It is well known that bond strengths and adsorption energies vary from one functional to the next^{80,81} and PBE in particular neglects van der Waals (vdW) dispersion forces and does not include exact exchange. Indeed, previous work on similar systems to those considered here, namely, the physisorption of water on h-BN⁸² and on BN doped benzene,⁸³ has shown that vdW interactions can be important. Here, however, we are concerned with strongly bonded chemisorption structures of the dissociated fragments of water and the other molecules involving an energy scale of several electronvolts. Nonetheless we have investigated the dissociative adsorption energies for all states in Fig. 4 using the vdW-inclusive optB86b-vdW functional.^{84–86} We find that the inclusion of vdW interactions makes the thermodynamics of dissociative adsorption energy more favourable by 0.2–0.5 eV. With this functional, some adsorption states are exothermic even in the absence of pre-adsorbed hydrogen. In contrast, when we look at the thermodynamics of water adsorption with B3LYP,^{87–90} that accounts for some exact exchange but not dispersion, dissociative adsorption is less favourable by *circa* 0.2–0.4 eV. It is clear therefore that the thermodynamics of dissociative adsorption is sensitive to the choice of exchange-correlation functional, with the PBE values presented here resting in the middle of three functionals considered. Importantly, the relative energies and trends across the surfaces remain unchanged whether or not dispersion interactions or exact exchange are accounted for.

Second, when probed experimentally 2D materials like graphene and h-BN are often adsorbed on a support material such as metals or silicon carbide. We have not included supporting materials in this study but the electronic properties of graphene and h-BN can be influenced by the choice of support.^{68–75} Metals, for instance, can hybridise the p_z-states in graphene and the N atoms in h-BN, and thus alter the reactivity of the surfaces.^{69,71,72} It is also known that differences in the lattice constants of the 2D material and support can lead to an undulating moiré structure in which different regions of the 2D overlayer interact differently with the substrate.^{68,70,73,75,76} It would be interesting in future work to explore how the presence of a substrate alters the trends observed here.

Third, we have seen that depending on the type of doping, the thermodynamics of dissociation of either polar or non-polar molecules can be enhanced. This would potentially be

exploited in heterogeneous catalysis where it is generally desirable to identify catalysts that can cleave specific bonds and as a result enhance the selectivity towards a particular reaction product. In future work, it would be interesting to explore this possibility through calculations of the kinetics of dissociation on the substrates considered here. However, since it is now well established that reaction barriers for chemical reactions at surfaces correlate well with the thermodynamics, it is not unreasonable to suggest that the thermodynamic trends identified here could lead to interesting catalytic behaviour.

V. CONCLUSION

To conclude, the dissociative adsorption of water, H₂, methane, and methanol has been studied on pristine graphene and h-BN, and on their doped counterparts (BNDG and 2CBN) using DFT. Most notably, isoelectronic doping of the pristine surfaces makes the dissociation process more favourable generally by at least 1 eV. Based on electronic structure analyses, we conclude that the increased reactivity of the surface is because B atoms (as a doping species) are more susceptible to nucleophilic attack, and in 2CBN, the C–C double bond is more susceptible to alkene addition-like reactions. These changes in the local electronic structure favour particular adsorption configurations. The OH component bonds strongly to the doping B atom, whilst H atoms bond preferentially to C compared to either B or N atoms. Hence, methanol behaves very similarly to water as a polar molecule, because of the OH group. In the same vein, H₂ and methane follow the same trend across the different surfaces, with both binding preferentially on 2CBN, where there is a high energy C–C double bond.

The results presented in this study also suggest that adsorption is exothermic in the presence of adsorbed H atoms (or protons) on the surface. Thus, there could be important implications for the transport properties and chemical reactions of water and other molecules across doped graphene and h-BN membranes, and conditions (acidic or basic) are likely to be useful gauges for altering the interaction with molecules.

Finally, we observe variations in the thermodynamics for the set of molecules considered depending on the surface. Again, we caution that the calculation of reaction barriers and even rates is an important next step, but these results suggest that one can vary the preference for H₂ dissociative adsorption to that of water or methanol for example, and consequently alter the course of reaction pathways in either H₂ or methanol formation processes. Consider, for example, the wasteful dehydration and methanation reactions in reaction pathways of Cortright *et al.* catalysed by a metal for H₂ production;⁴⁸ wherein H₂ is consumed by reacting with CO₂ at low temperatures to produce alkanes and water. This reaction can be avoided if methanol, methane, and water are split more readily than H₂. According to our findings this might be achievable for methanol and water by doping graphene with BN. Overall, our results indicate that isoelectronically doped graphene and h-BN could exhibit interesting chemical and catalytic activities which could potentially be exploited.

ACKNOWLEDGMENTS

We are grateful for support from the University College London and Argonne National Laboratory (ANL) through the Thomas Young Centre-ANL initiative. Some of the research leading to these results has received funding from the European Research Council under the European Union's Seventh Framework Programme (No. FP/2007-2013)/ERC Grant Agreement No. 616121 (HeteroIce project). A.M. is supported by the Royal Society through a Wolfson Research Merit Award. O.A.v.L. acknowledges funding from the Swiss National Science foundation (No. PPOOP2 138932). In addition, we are grateful for computing resources provided by the London Centre for Nanotechnology and University College London. We would also like to thank Michail Stamatakis for his very helpful insights and suggestions.

- ¹E. N. Wang and R. Karnik, *Nat. Nanotechnol.* **7**, 552 (2012).
- ²W. Lei, D. Portehault, D. Liu, S. Qin, and Y. Chen, *Nat. Commun.* **4**, 1777 (2013).
- ³R. Raccichini, A. Varzi, S. Passerini, and B. Scrosati, *Nat. Mater.* **14**, 271 (2015).
- ⁴B. B. Sales, M. Saakes, J. W. Post, C. J. N. Buisman, P. M. Biesheuvel, and H. V. M. Hamelers, *Environ. Sci. Technol.* **44**, 5661 (2010).
- ⁵P. Dhiman, F. Yavari, X. Mi, H. Gullapalli, Y. Shi, P. M. Ajayan, and N. Koratkar, *Nano Lett.* **11**, 3123 (2011).
- ⁶J. Yin, Z. Zhang, X. Li, J. Zhou, and W. Guo, *Nano Lett.* **12**, 1736 (2012).
- ⁷J. Yin, X. Li, J. Yu, Z. Zhang, J. Zhou, and W. Guo, *Nat. Nanotechnol.* **9**, 378 (2014).
- ⁸A. Siria, P. Poncharal, A.-L. Biance, R. Fulcrand, X. Blase, S. T. Purcell, and L. Bocquet, *Nature* **494**, 455 (2013).
- ⁹X. Li, J. Zhao, and J. Yang, *Sci. Rep.* **3**, 1 (2013).
- ¹⁰Y. Sun, Q. Wu, and G. Shi, *Energy Environ. Sci.* **4**, 1113 (2011).
- ¹¹Q. Li, B. Guo, J. Yu, J. Ran, B. Zhang, H. Yan, and J. R. Gong, *J. Am. Chem. Soc.* **133**, 10878 (2011).
- ¹²V. Fthenakis, *Renewable Sustainable Energy Rev.* **13**, 2746 (2009).
- ¹³L. Ci, L. Song, C. Jin, D. Jariwala, D. Wu, Y. Li, A. Srivastava, Z. Wang, K. Storr, L. Balicas *et al.*, *Nat. Mater.* **9**, 430 (2010).
- ¹⁴Z. Liu, L. Ma, G. Shi, W. Zhou, Y. Gong, S. Lei, X. Yang, J. Zhang, J. Yu, K. P. Hackenberg *et al.*, *Nat. Nanotechnol.* **8**, 119 (2013).
- ¹⁵X. Fan, Z. Shen, A. Q. Liu, and J.-L. Kuo, *Nanoscale* **4**, 2157 (2012).
- ¹⁶B. Xu, Y. H. Lu, Y. P. Feng, and J. Y. Lin, *J. Appl. Phys.* **108**, 073711 (2010).
- ¹⁷P. P. Shinde and V. Kumar, *Phys. Rev. B* **84**, 125401 (2011).
- ¹⁸Y. Ding, Y. Wang, and J. Ni, *Appl. Phys. Lett.* **95**, 123105 (2009).
- ¹⁹R. Zhao, J. Wang, M. Yang, Z. Liu, and Z. Liu, *J. Phys. Chem. C* **116**, 21098 (2012).
- ²⁰N. Kumar, K. Moses, K. Pramoda, S. N. Shirodkar, A. K. Mishra, U. V. Waghmare, A. Sundaresan, and C. N. R. Rao, *J. Mater. Chem. A* **1**, 5806 (2013).
- ²¹K. Moses, S. N. Shirodkar, U. V. Waghmare, and C. N. R. Rao, *Mater. Res. Express* **1**, 025603 (2014).
- ²²C.-K. Chang, S. Kataria, C.-C. Kuo, A. Ganguly, B.-Y. Wang, J.-Y. Hwang, K.-J. Huang, W.-H. Yang, S.-B. Wang, C.-H. Chuang *et al.*, *ACS Nano* **7**, 1333 (2013).
- ²³L. Ferrighi, M. I. Trioni, and C. Di Valentin, *J. Phys. Chem. C* **119**, 6056 (2015).
- ²⁴Y. Zhao, L. Yang, S. Chen, X. Wang, Y. Ma, Q. Wu, Y. Jiang, W. Qian, and Z. Hu, *J. Am. Chem. Soc.* **135**, 1201 (2013).
- ²⁵R. Baierle, T. Schmidt, and A. Fazzio, *Solid State Commun.* **142**, 49 (2007).
- ²⁶S. Kattel, P. Atanassov, and B. Kiefer, *J. Mater. Chem. A* **2**, 10273 (2014).
- ²⁷Y. Li, W. Zhou, H. Wang, L. Xie, Y. Liang, F. Wei, J.-C. Idrobo, S. J. Pennycook, and H. Dai, *Nat. Nanotechnol.* **7**, 394 (2012).
- ²⁸D. Sen, R. Thapa, and K. K. Chattopadhyay, *ChemPhysChem* **15**, 2542 (2014).
- ²⁹S. Wang, L. Zhang, Z. Xia, A. Roy, D. W. Chang, J.-B. Baek, and L. Dai, *Angew. Chem., Int. Ed.* **51**, 4209 (2012).
- ³⁰S. Zhong, L. Zhou, L. Wu, L. Tang, Q. He, and J. Ahmed, *J. Power Sources* **272**, 344 (2014).
- ³¹S. Sinthika, E. M. Kumar, and R. Thapa, *J. Mater. Chem. A* **2**, 12812 (2014).
- ³²S. Wang, E. Iyyamperumal, A. Roy, Y. Xue, D. Yu, and L. Dai, *Angew. Chem., Int. Ed.* **50**, 11756 (2011).

- ³³H. Fei, R. Ye, G. Ye, Y. Gong, Z. Peng, X. Fan, E. L. G. Samuel, P. M. Ajayan, and J. M. Tour, *ACS Nano* **8**, 10837 (2014).
- ³⁴L. Joly, G. Tocci, S. Merabia, and A. Michaelides, *J. Phys. Chem. Lett.* **7**, 1381 (2016).
- ³⁵R. J. Baierle, P. Piquini, T. M. Schmidt, and A. Fazzio, *J. Phys. Chem. B* **110**, 21184 (2006).
- ³⁶M. Pizzochero, O. Leenaerts, B. Partoens, R. Martinazzo, and F. M. Peeters, *J. Phys.: Condens. Matter* **27**, 425502 (2015).
- ³⁷M. Chhetri, S. Maitra, H. Chakraborty, U. V. Waghmare, and C. Rao, *Energy Environ. Sci.* **9**, 95 (2016).
- ³⁸C. Tsai, K. Chan, J. K. Nørskov, and F. Abild-Pedersen, *Catal. Sci. Technol.* **5**, 246 (2015).
- ³⁹Y. H. Li, P. F. Liu, L. F. Pan, H. F. Wang, Z. Z. Yang, L. R. Zheng, P. Hu, H. J. Zhao, L. Gu, and H. G. Yang, *Nat. Commun.* **6**, 8064 (2015).
- ⁴⁰B. Yang, R. Burch, C. Hardacre, G. Headdock, and P. Hu, *ACS Catal.* **4**, 182 (2013).
- ⁴¹R. Subbaraman, D. Tripkovic, K.-C. Chang, D. Strmcnik, A. P. Paulikas, P. Hirunsi, M. Chan, J. Greeley, V. Stamenkovic, and N. M. Markovic, *Nat. Mater.* **11**, 550 (2012).
- ⁴²A. U. Nilekar, K. Sasaki, C. A. Farberow, R. R. Adzic, and M. Mavrikakis, *J. Am. Chem. Soc.* **133**, 18574 (2011).
- ⁴³L. C. Grabow and M. Mavrikakis, *ACS Catal.* **1**, 365 (2011).
- ⁴⁴A. Vojvodic and J. K. Nørskov, *Science* **334**, 1355 (2011).
- ⁴⁵C. Tsai, K. Chan, J. K. Nørskov, and F. Abild-Pedersen, *J. Phys. Chem. Lett.* **5**, 3884 (2014).
- ⁴⁶J. K. Nørskov, T. Bligaard, A. Logadottir, S. Bahn, L. B. Hansen, M. Bollinger, H. Bengaard, B. Hammer, Z. Slijvančanin, M. Mavrikakis *et al.*, *J. Catal.* **209**, 275 (2002).
- ⁴⁷A. Michaelides, Z.-P. Liu, C. Zhang, A. Alavi, D. A. King, and P. Hu, *J. Am. Chem. Soc.* **125**, 3704 (2003).
- ⁴⁸R. Cortright, R. Davda, and J. A. Dumesic, *Nature* **418**, 964 (2002).
- ⁴⁹X. Guo, G. Fang, G. Li, H. Ma, H. Fan, L. Yu, C. Ma, X. Wu, D. Deng, M. Wei *et al.*, *Science* **344**, 616 (2014).
- ⁵⁰G. Kresse and J. Hafner, *Phys. Rev. B* **47**, 558 (1993).
- ⁵¹G. Kresse and J. Hafner, *Phys. Rev. B* **49**, 14251 (1994).
- ⁵²G. Kresse and J. Furthmüller, *Comput. Mater. Sci.* **6**, 15 (1996).
- ⁵³G. Kresse and J. Furthmüller, *Phys. Rev. B* **54**, 11169 (1996).
- ⁵⁴P. E. Blöchl, *Phys. Rev. B* **50**, 17953 (1994).
- ⁵⁵G. Kresse and D. Joubert, *Phys. Rev. B* **59**, 1758 (1999).
- ⁵⁶For methanol the separation distance in the z-direction was increased to 20 Å to allow space for the larger adsorbed fragments.
- ⁵⁷J. P. Perdew, K. Burke, and M. Ernzerhof, *Phys. Rev. Lett.* **77**, 3865 (1996).
- ⁵⁸We verified the stability of the doped substrates by calculating their cohesive energies and we find good agreement with other work for similar arrangements of doping atoms¹⁷. Cohesive energies for the different substrates have been calculated as $E_{coh} = E_{sheet}^{tot} - N_C E_C^{tot} - N_B E_B^{tot} - N_N E_N^{tot}$ where E_{sheet}^{tot} , E_C^{tot} , E_B^{tot} and E_N^{tot} are the total energies of the sheet and gaseous C, B and N atoms in the unit cell, respectively, and N_C , N_B and N_N are the numbers of C, B and N atoms in the unit cell. The doped sheets in this study have cohesive energies between that of graphene and h-BN, and the four substrates range between -7.06 and -7.84 eV/atom.
- ⁵⁹S. Hu, M. Lozada-Hidalgo, F. Wang, A. Mishchenko, F. Schedin, R. Nair, E. Hill, D. Boukhvalov, M. Katsnelson, R. Dryfe *et al.*, *Nature* **516**, 227 (2014).
- ⁶⁰P. Cabrera-Sanfelix and G. R. Darling, *J. Phys. Chem. C* **111**, 18258 (2007).
- ⁶¹S. Xu, S. Irle, D. Musaev, and M. Lin, *J. Phys. Chem. A* **109**, 9563 (2005).
- ⁶²Y. Lin, F. Ding, and B. I. Yakobson, *Phys. Rev. B* **78**, 041402 (2008).
- ⁶³D. Boukhvalov, M. Katsnelson, and A. Lichtenstein, *Phys. Rev. B* **77**, 035427 (2008).
- ⁶⁴R. Balog, B. Jørgensen, L. Nilsson, M. Andersen, E. Rienks, M. Bianchi, M. Fanetti, E. Lægsgaard, A. Baraldi, S. Lizzit *et al.*, *Nat. Mater.* **9**, 315 (2010).
- ⁶⁵Ž. Šljivančanin, M. Andersen, L. Hornekær, and B. Hammer, *Phys. Rev. B* **83**, 205426 (2011).
- ⁶⁶P. Merino, M. Švec, J. I. Martnez, P. Mutombo, C. Gonzalez, J. A. Martn-Gago, P. L. de Andres, and P. Jelinek, *Langmuir* **31**, 233 (2015).
- ⁶⁷The 0.04 eV difference between *cis-para* and *trans-ortho* adsorption configurations remained the same using a denser $6 \times 6 \times 1$ k-point mesh.
- ⁶⁸Y. Ding, M. Iannuzzi, and J. Hutter, *J. Phys. Chem. C* **115**, 13685 (2011).
- ⁶⁹B. Wang and M.-L. Bocquet, *J. Phys. Chem. Lett.* **2**, 2341 (2011).
- ⁷⁰B. Wang, S. Günther, J. Wintterlin, and M.-L. Bocquet, *New J. Phys.* **12**, 043041 (2010).
- ⁷¹J. Wintterlin and M.-L. Bocquet, *Surf. Sci.* **603**, 1841 (2009).
- ⁷²J. G. Díaz, Y. Ding, R. Koitz, A. P. Seitsonen, M. Iannuzzi, and J. Hutter, *Theor. Chem. Acc.* **132**, 1350 (2013).
- ⁷³S. Altenburg, J. Kröger, B. Wang, M.-L. Bocquet, N. Lorente, and R. Berndt, *Phys. Rev. Lett.* **105**, 236101 (2010).
- ⁷⁴M. Lattalais and M.-L. Bocquet, *J. Phys. Chem. C* **119**, 9234 (2015).
- ⁷⁵X. Li, J. Feng, E. Wang, S. Meng, J. Klimeš, A. Michaelides *et al.*, *Phys. Rev. B* **85**, 085425 (2012).
- ⁷⁶S. Joshi, D. Eciya, R. Koitz, M. Iannuzzi, A. P. Seitsonen, J. Hutter, H. Sachdev, S. Vijayaraghavan, F. Bischoff, K. Seufert *et al.*, *Nano Lett.* **12**, 5821 (2012).
- ⁷⁷R. F. Bader, *Atoms in Molecules* (Wiley Online Library, 1990).
- ⁷⁸G. Henkelman, A. Arnaldsson, and H. Jónsson, *Comput. Mater. Sci.* **36**, 354 (2006).
- ⁷⁹Of course there are many ways to project charges onto atoms and Bader charges discussed here are simply used for pinpointing the relevant trends in the materials.
- ⁸⁰L. Schimka, J. Harl, A. Stroppa, A. Grüneis, M. Marsman, F. Mittendorfer, and G. Kresse, *Nat. Mater.* **9**, 741 (2010).
- ⁸¹J. Klimeš and A. Michaelides, *J. Chem. Phys.* **137**, 120901 (2012).
- ⁸²Y. S. Al-Hamdani, M. Ma, D. Alfè, O. A. von Lilienfeld, and A. Michaelides, *J. Chem. Phys.* **142**, 181101 (2015).
- ⁸³Y. S. Al-Hamdani, D. Alfè, O. A. von Lilienfeld, and A. Michaelides, *J. Chem. Phys.* **141**, 18C530 (2014).
- ⁸⁴M. Dion, H. Rydberg, E. Schröder, D. C. Langreth, and B. I. Lundqvist, *Phys. Rev. Lett.* **92**, 246401 (2004).
- ⁸⁵A. D. Becke, *J. Chem. Phys.* **84**, 4524 (1986).
- ⁸⁶J. Klimeš, D. R. Bowler, and A. Michaelides, *Phys. Rev. B* **83**, 195131 (2011).
- ⁸⁷A. D. Becke, *J. Chem. Phys.* **98**, 5648 (1993).
- ⁸⁸C. Lee, W. Yang, and R. G. Parr, *Phys. Rev. B* **37**, 785 (1988).
- ⁸⁹S. H. Vosko, L. Wilk, and M. Nusair, *Can. J. Phys.* **58**, 1200 (1980).
- ⁹⁰P. J. Stephens, F. J. Devlin, C. F. Chabalowski, and M. J. Frisch, *J. Phys. Chem.* **98**, 11623 (1994).